

Optical Band Gap, Oxidation Polarizability, Optical Basicity and Electronegativity Measurements of Silicate Glasses Using Ellipsometer and Abbe Refractometer

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How to cite this paper: Hussain, Z. (2021) Optical Band Gap, Oxidation Polarizability, Optical Basicity and Electronegativity Measurements of Silicate Glasses Using Ellipsometer and Abbe Refractometer. *New Journal of Glass and Ceramics*, **11**, 1-33. https://doi.org/10.4236/njgc.2021.111001

Received: September 22, 2020 Accepted: January 5, 2021 Published: January 8, 2021

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Abstract

The values of refractive index (*n*) for silicate glasses (silica, soda lime and borosilicate 7059) are decreased from 1.5119 to 1.5111, 1.5086 to 1.5065 and 1.5296 to 1.5281, respectively; and the optical band gap (E_{e}) is increased from 9.8 to 9.81 eV, 9.845 to 9.88 eV and 9.56 to 9.58 eV, respectively over the temperature range 295 - 473 K using ellipsometer at wavelength 632.8 nm. While *n* is decreased from 1.5276 to 1.5274, 1.5074 to 1.5070 and from 1.5283 to 1.5281, respectively; and E_g is increased from 9.59 to 9.592 eV, 9.862 to 9.870 eV, and 9.574 to 9.58 eV, respectively over the temperature range 297 -322 K using Abbe refractometer at wavelength 589.3 nm. The values of oxide ion polarizability $[a_{02}, (n) \text{ and } a_{02}, (E_g)]$ regarding silica, soda lime and borosilicate 7059 glasses are decreased from 1.3427 to 1.3408, 1.6014 to 1.5941, 1.4329 to 1.4193, respectively over the temperature range 295 - 473 K using ellipsometer; and are decreased from 1.3786 to 1.3764, 1.5991 to 1.5969, 1.4297 to 1.4191, respectively over the temperature range 297 - 322 K using Abbe refractometer. Similarly, the values of optical basicity $[A(n) \text{ and } A(E_{z})]$ of silica, soda lime, and borosilicate 7059 glasses are decreased from 0.4272 to 0.4245, 0.6271 to 0.6224, 0.5045 to 0.4933, respectively over the temperature range 295 - 473 K using ellipsometer; and are decreased from 0.4586 to 0.4567, 0.6256 to 0.6242, 0.5018 to 0.4930, respectively over the temperature range 297 - 322 K using Abbe refractometer. Further, we have found that for silica, soda lime and borosilicate 7059, the values of electronegativity $(\xi_{Iav})\zeta_{1av}$) using Zahid numerical model [based on α_{02} . (*n*) and *A* (*n*)] are increased from 5.1035 to 5.5504, 4.0393 to 4.830, 4.8143 to 5.0111, respectively over the temperature range 295 - 473 K using ellipsometer; while these values are increased from 5.0657 to 5.2149, 5.0657 to 5.2149, 4.8357 to 5.0111, respectively over the temperature range 297 - 322 K using Abbe refractometer. It is very clear from this research report that both refractive index and optical band gap-based-oxide ion polarizability and optical basicity have the same decreasing trend as the temperature is increased, and this trend indicates that the reported glasses have a very small amount of electronic polarizability. Moreover, this decreasing trend occurs due to the decreasing amount of non-bridging oxygen (**NBO**) which in turn caused a decrease in refractive index within the silicate glass system at higher temperature. **Since the calculated values of electronegativity** are found to be in the range 4.0393 -5.5504 for the reported silicate glasses, so all these glasses have an ionic character. Moreover, low values of optical basicity and of oxide ion polarizability suggest that the silicate glasses are not novel glasses (optical functional glasses) for non-linear optical (NLO) devices or for three dimensional displays.

Keywords

Glass, Optical Properties, Thermal Properties, Physical Properties, Ellipsometry

1. Introduction

Among different classes of materials, glasses have acknowledged advantages because of their diverse technological and biological applications including solid state lasers, optical filters, water treatment and as biomaterials [1] [2] [3] [4]. Among the optical glasses, silicate glasses are of low cost and have high optical transmission in the visible and near-infrared spectral range which makes them commonly used as microscope slides [5] [6]. Inorganic glasses, glass-ceramic and glaze materials have widely been used in fields such as electrical engineering/electronics/sensors and solar energy [7]-[15]. Silicate glasses have also great use in optics/optical telecommunications [16] [17], structural mechanics [7] [18], medical [19] [20], nuclear technology [21] [22], superconductors [21] [23] and in microfluidics [24] [25]. Recently many researchers study glasses as switching and memory devices and as superior insulators and dielectrics [26] [27]. Silicate glass (silica, soda lime, borosilicate 7059) is an attractive host matrix for transition metal ions because of its excellent optical and mechanical properties, good chemical stability, low non-linear refractive index, large tensile fracture strength and due to good durability [4] [28] [29]. Oxide glasses take a considerable attention in view of their potential use in the areas of opto-electronics such as laser technology, optical fibers, non-linear optical devices and sensor systems [30] [31].

The soda lime silicate glass has been found to be a suitable optical material with high transparency, low melting point, high thermal stability and good rare-earth ions solubility [32] [33]. Silica glass, because of its favourable physical,

chemical, and optical characteristics, has been used in numerous applications such as laboratory glassware, as lenses or beam splitters, for lighting and IR heating, in telecommunications, in fiber optics and in mico and optoelectronics [34] [35]. Further, silicon oxide (SiO₂) has been used as substrates for electronic displays, optical fibers, optical disc, medical and dental implants and for radiation shielding [36] [37] [38]. Containers, windows, lighting, insulation, fibre, and other hand crafted art objects are typical of traditional uses of silicate glasses [39] [40].

In nuclear industry, borosilicate glass is mainly used as matrix for immobilizing the radioactive ions present in the waste generated from the nuclear reactors [41] [42] [43]. Due to these interesting physical properties, borosilicate glasses can be used as laser host matrices after doping with rare earth oxides [44] [45].

The optical research on rare earth (RE) doped glasses draw great consideration due to their broad application in optical areas like optical switches for laser and sensors and optical communications [46] [47]. These types of glasses can be implemented in high density optical memory applications such as coast-guard communication, colour display and for solar cells etc. [48] [49]. Glasses emerge as an important class of materials that exhibit very attractive properties and have shown great potential in variety of applications such as optical switches, solid oxide fuel cells, microelectronics, telecommunications and for medical [50] [51] [52].

In this paper, our main aim is to report temperature dependent linear optical and physical properties of silicate glasses (silica, soda lime, borosilicate 7059). In the first step, we have measured the temperature dependent refractive index (n) data on silicate glasses using single wavelength manual ellipsometry and Abbe refractometer over a temperature range 295 - 473 K. It should be noted that the absorption index (k) of silicate glasses is very small in the visible and near infrared spectral range and may be assumed to be zero for all practical purposes. In the second step, we have generated the data on molar refraction (R_m) , molar electronic polarizability (α_{me}), optical band gap (E_g), Oxide ion polarizability (a_{02}) and optical basicity (A) regarding silicate microscope glass slides. In the third step, we have generated temperature dependent electronegativity data on silicate glasses with the application of new models (Zahid models) along with the old models using the obtained values of oxide ion polarizability (a_{o2}) and optical basicity [A(n)], and this new data is found to be in a good agreement with the literature. The reported data has also been interpreted using different empirical models for technical applications.

2. Experimental Techniques

2.1. Sample Preparation

Silicate glasses mean silica, soda lime silica glass, and borosilicate glass. Silica is the dioxide form of silica, SiO₂, and occurs mostly as quartz sand, flint, and agate. It is formed when silicon is exposed to oxygen. The commercially supplied

SiO₂ glasses (microscope slides) employed in this study have mass density as 2.65 g/cm³ and molecular weight as **60.08 g/mol**. The other silicate glasses (soda lime and borosilicate microscope slides) are also commercially supplied. A typical composition of soda lime silicate glass is 73% SiO₂ + 15% Na₂O + 7% CaO + 4% MgO + 1% Al₂O₃, and its mass density is 2.5 g/cm³. Since the molecular weights of all these compositions are known in the literature and so they can be printed as 60.08 g/mol, 61.98 g/mol, 56.0774 g/mol, 40.3044 g/mol, and 101.96 g/mol, respectively. We have measured the molecular weight of soda lime as the average of sums of the % molecular weights of all constituents and measuring the molecular weight of soda lime using the mole % of all participating cations in the composition. As a result, the average molecular weight of soda lime silicate is measured as 59.34 g/mol.

The typical chemical composition of borosilicate 7059 glass is 80% SiO₂ + 13% B₂O₃ + 4% Na₂O + 3% Al₂O₃, and its mass density is 2.76 g/cm³. The molecular weights of all compositions relating to 7059 glass (as found in the literature) can be printed respectively as 60.08 g/mol, 69.62 g/mol, 61.98 g/mol and 101.96. Following the method as we did for soda lime glass, the average molecular weight of 7059 silicate glass is determined as 62.69 g/mol.

Next, molecular volume (V_m) of any solid = Molecular weight/Mass density = M/ρ = cc/mol. So, for silica glass, V_m = 60.08/2.65 = **22.672 cc/mol**; for soda lime, V_m = 60.08/2.65 = **23.74 cc/mol**; and for 7059 silicate glass, V_m = 60.08/2.65 = **22.714 cc/mol**.

These silicate glasses are available in many forms and sizes such as rod, tube, or microscopic slide. In this work, silicate flat glasses were obtained commercially as microscopic slides. Microscope slides of these glasses measure about 75 mm by 25 mm and are about 1 mm thick.

2.2. Substrate Cleaning Procedure

Silica and other glass slides were washed with "Micro" cleaning fluids (purchased from International Products Corporation) and rinsed in tap water and distilled water. Having been left to soak in freshly distilled water for several hrs, the substrates were flushed with both acetone and iso-propyl alcohol (IPA). The substrates were then subjected to ultrasonic distillation unit containing distilled water for 3 min, and finally were cycled for more than 20 min in a vapour degreasing unit (Sholet unit) in which iso-propyl alcohol was reflexed continuously, and at this stage the cleaning run was complete.

2.3. Measuring Techniques

The refractive indices were measured using both Abbe refractometer with a sodium vapour lamp as the light source emitting the light at a wavelength, λ , of 589.3 nm (D line) and ellipsometry using a laser with a wavelength at 632.8 nm.

2.3.1. Abbe Refractometer Measurements

The Abbe refractometer was used for measuring refractive indices of silicate

glasses over a limited operating temperature range. In principle, it consists of a pair of glass prisms with a substrate or a film of liquid between them. Light enters sample from the illuminating prism, get refracted at critical angle at the bottom surface of measuring prism, and then the telescope is used to measure position of the border between bright and light areas. A sample must be well cleaned, polished as flat as possible before placing on the prism surface. Secondly, as an instrument, Abbe refractometer only gives access to the real part of the refractive index.

The refractive indices of silicate glasses were measured by using an Abbe 60 refractometer with a sodium light as the light source at a wavelength λ of 589.3 nm and with temperature-controlled prisms having mono-bromona phthalence as the contact layer (with a refractive index of 1.51) between the silicate glass and the prism of the refractometer. Different Abbe data (scale readings) on silicate glasses was generated by varying the temperature of the glass slide (s) and the prism of the refractometer by circulating hot water through a temperature controlled system over the operating range 278 - 322 K, which was read from the integral digital display on the instrument. The whole temperature dependent Abbe data was then converted into the refractive index (*n*) data using Abbe utilities software. With good temperature control and use of the micrometer vernier scale, an average accuracy of ±0.0008 in the value of refractive index (*n*) was achieved across the specified temperature measuring range.

2.3.2. Ellipsometry and Optical Measurements

Ellipsometry is an optical technique that measures changes in polarization (phase, Δ , and amplitude, Ψ) of monochromatic light reflecting from matter, and these standard ellipsometric parameters Ψ and Δ are related to the complex ratio of reflection coefficient for light polarized parallel **p** and perpendicular **s** to the plane of incidence.

2.3.3. Basic Principle of Ellipsometry

The formulation for the manual null ellipsometry is based on Fresnel derived optical equations for any kind of monochromatic light reflecting from a material. Fresnel physical model associated with PCWSW'A version of the most common null ellipsometer configuration (Polarizer-Compensator-Window-Sample-Window-Analyzer) is shown in **Figure 1**, where Fresnel reflection (amplitude) coefficients for a dielectric substrate relating to s and p components can be expressed into a single relation [53] [54].

$$\rho = \frac{E'_{2p}}{E_{1p}} / \frac{E'_{2s}}{E_{1s}} = \left| \frac{r'_p}{r'_s} \right| e^{i\Delta_s} ,$$

 $\rho_s = \tan \psi_s \mathrm{e}^{\mathrm{i}\Delta_s},\tag{1}$

where Ψ_s and Δ_s are ellipsometric (measurable) angles for a bare substrate, *i* designates the imaginary unit $(-1)^{1/2}$, and the ratio, ρ_s is for the dielectric substrate.

or



Figure 1. Azimuths and amplitudes of plane-polarized light reflected from a sample (microscopic glass slide) at any angle of incidence.

Refractive index, n_s of the substrate is computed from the Expression [54] [55]

$$n_{s} = \tan \phi_{1} \left\{ 1 - \left(4\rho_{s} \sin^{2} \phi_{1} \right) \left(\rho_{s} + 1 \right)^{-2} \right\}^{1/2}.$$
 (2)

If the substrate is absorbing, its refractive index will, of course, be complex. The parallel and perpendicular components for Fresnel reflection coefficients, r_p and r_s were calculated using Fortran 77 version for a single substrate model [55] [56]. The details for the classical ellipsometry inversion formulas are described and reviewed elsewhere [55] [57].

2.3.4. Operation and Measurement of Ellipsometric Parameters Δ and ψ

High precision single wavelength (632.8 nm) manual ellipsometry at fixed angle $(\sim 60 \sim <^{\circ})$ was chosen for in situ heating experiments in order to generate a series of data on the optical constants of silicate glasses in the visible range of solar spectrum. The manual ellipsometer (LQ1PQWSW'AD) which was built up with some modifications in the above version, is shown in Figure 2. A beam of unpolarized monochromatic red light from a He-Ne laser after passing through a quarter-wave plate, Q1 (set at an azimuth~45°) was changed into a circularly polarized light. When this circularly polarized light was incident on a polarizer (P), (which is a Glan Thomson prism), only linearly polarized light was allowed to transmit due to total internal reflection occurring inside the prism. Another quarter wave plate, Q (also known as retarder or compensator, C, and was made out of birefringent mica like Q_1 and was fixed with its fast axis at $\pm 45^{\circ}$ to the plane of incidence) which was placed in between the polarizer (P) and the sample (S), changed the approaching polarized light into elliptically polarized light. This polarized light after falling normally (within 2° - 3°) on the fused silica optical window (W) was finally incident on the microscope glass slide at an angle $\sim 60^{\circ}$. It should be specified that the silicate glass slide with its back surface



Figure 2. A picture of the apparatus of manual ellipsometer built-up which composed of an aluminum alloy base plate to which ellipsometric components were fixed.

roughened was fixed in a horizontal (or vertical) position inside a vacuum cryostat (positioned on a spectrometer table) and was joined to a temperature controller through a alumel-chromel thermocouple. After reflection from the 7059 glass plate, the polarized light passed through another fused silica optical window (W') and was then transmitted through an analyzer (A) (another Glan Thomson prism), and finally reached photomultiplier tube (PM) connected to Keithly to monitor output signal. Since measurements performed with and without the fused silica windows revealed a lot of difference; thus, for the in situ experiments, all the measurements were made with the apparatus fitted with the fused silica optical windows under vacuum.

The precise manual ellipsometric measurements depend on the clear null point at the detector (PM). The angles (P) and (A) were adjusted for minimum output, and were measured as positive counter clockwise from the plane of incidence when looking into the beam. All the scales in the reported apparatus were illuminated to minimize eye strain. The polarizer (P) and the analyzer (A) scale readings were read on graduated circles, which were equipped with two diametrically opposite verniers to cancel the effects of eccentricity. The angle of the analyzer (A) determines the relative amplitude (Ψ°) of the reflected p and s waves; and the angle of polarizer (p), when it is on the extinction setting, measures the angle (Δ°). Once the ellipsometric measurements of Δ and Ψ are determined, the unknown parameters, real part of the refractive index (*n*) and index of absorption (*k*) of an optical substrate, can in principle be calculated using Fortran program.

3. Experimental Results and Error Analysis

The relations of (Ψ and Δ) to (P and A) in four zones at extinction settings are

given by equations [55] [57]

$$A_i = a_p \left(\text{or } a_s \right) = \pm \Psi_i \tag{3}$$

and

$$2p_i \pm \frac{\pi}{2} = \pm \left(-1\right)^i \Delta_i \tag{4}$$

where i = 1, 2, 3, 4. And a_p or a_s (depending upon zone) denotes the analyser azimuth angle and p is the polarizer azimuth angle. The quantities to be determined from the experimental data (Δ, Ψ) using ellipsometric equations are the real part of the refractive index (n), and the extinction coefficient (k) of the acquired glass slide. Ellipsometric Equation (1) was solved using Fortran 77 version. The (n, k) values of different glasses under different temperature conditions were computed directly from the experimental data $(\Delta$ and $\Psi)$. A comparison between manual and auto ellipsometric data is shown in **Table 1**, which indicates that the manual ellipsometric components were aligned to very high precision. The generated experimental data $(\Delta$ and $\Psi)$ of silicate glasses under different temperature conditions are tabulated in **Tables 2-4**.

For a clean substrate (silica, soda lime or 7059 glass), a change of $\pm 0.02^{\circ}$ in the measurement of the angle of incidence due to optical window strain gives a net change of $\pm 0.02^{\circ}$ in Ψ value, which gives an error in *n* of about $\delta n = \pm 0.00041$. Similarly, a change of $\pm 0.04^{\circ}$ in Δ due to same degree of strain brings a net change in the index of absorption of about $\delta k = \pm 0.0005$. After alignment, the maximum uncertainty in Δ and Ψ in an open air were found about $\pm 0.02^{\circ}$ and

 Table 1. Refractive index of 7059 glass (with back surface roughened by silicon carbide)

 measured with the data obtained using Abbe refractometer.

Type of instrument used	∆ (°)	ψ (°)	Π _e	ABBE Reading (°)	∏a
Manual ellipsometer	0.06	5.34	1.5296		
Automatic ellipsometer	1.96	5.04	1.5264		
Abbe refractometer				24.743	1.5283

Table 2. Temperature dependent ellipsometric study of transparent silica plate in comparison with the data obtained using Abbe refractometer.

	M	ABBE			
Temperature (K)	Δ (°)	ψ (°)	Пe	Reading (°)	Ω _s
295.0	-0.14	6.06	1.5119	24.672	1.5276
307.0	-0.14	6.06	1.5119	24.654	1.5275
321.0	-0.14	6.06	1.5119	24.654	1.5274
335.0	-0.14	6.07	1.5115		
373.0	-0.15	6.08	1.5111		

	Ma	ABBE				
Temperature (K)	∆ (°)	ψ (°)	Πe	Reading (°)	Пе	
297.0	-2.42	6.18	1.5086	22.321	1.5074	
309.0	-2.42	6.18	1.5086	22.309	1.5072	
321.0	-2.49	6.19	1.5082	22.288	1.5070	
335.0	-2.53	6.20	1.5079			
375.0	-2.48	6.22	1.5072			
427.0	-2.46	6.24	1.5065			

Table 3. Temperature dependent ellipsometric measurements on soda lime glass (with back face roughened by silica carbide) in comparison to data obtained using Abbe refractometer.

Table 4. Temperature dependent ellipsometric measurements on 7059 glass (with back face roughened by silica carbide) in comparison to data obtained using Abbe refractometer.

	М	anual ellipsome	ABBE		
Temperature (K)	Δ (°)	Ψ (*)	Ωe	Reading (°)	₿. Da
295.0	0.06	5.34	1.5296	24.747	1.5283
309.0	0.06	5.34	1.5296	24.735	1.5282
321.0	0.06	5.34	1.5296	24.719	1.5281
337.0	-0.07	5.35	1.5292		
373.0	0.08	5.36	1.5289		
425.0	0.05	5.37	1.5285		
473.0	-0.28	5.38	1.5281		

 $\pm 0.01^{\circ}$, respectively, and with an in situ manual ellipsometer fitted with fused quartz inlet and exit optical windows, the maximum calibration errors in Δ and Ψ were about $\pm 0.05^{\circ}$ and $\pm 0.02^{\circ}$ respectively. Nevertheless, four-zone ellipsometric measurements were always made in order to minimize systematic errors.

4. Results and Discussion

4.1. Molar Refraction (R_m), Molar Electronic Polarizability (α_{me}), Optical Band Gap (E_g) and Oxide Ion Polarizability [α_{02} . (n) and α_{02} . (E_g)]

In the first stage, the data on molar refraction (R_m), molar electron polarizability (a_{me}), and optical band gap (E_g) of silicate glasses (silica, soda lime, borosilicate 7059) were generated from Tables 2-4 using the following course of action.

Molar refraction (R_m) is related to molar volume (V_m) by the relation [58] [59]

$$R_m = \left(\frac{n^2 - 1}{n^2 + 2}\right) V_m \tag{5}$$

where

$$V_m = \frac{M}{\rho} = \frac{\text{molecular weight}}{\text{mass density}}$$
(6)

Molar electronic polarizability (a_{me}) can be related with the molar refractive index by an expression [60] [61]

$$\alpha_{me} = \frac{R_m}{2.52} \tag{7}$$

According to Duffy, there also exists a good correlation between molar refraction and the band gap energy (E_g) of the materials (or glasses) and that is given as [60] [61]

$$R_m = V_m \left[1 - \sqrt{\frac{E_g}{20}} \right] \tag{8}$$

The required data on the above optical parameters was obtained using Equations (5)-(8).

In the second stage, the following models were used to obtain other properties. The electronic oxide ion polarizabilities based on refractice index and optical band gap a_{02} . (*n*) and a_{02} . (*E*_g) for the silicate glasses were determined using their reported refractive indices (*n*) and the obtained optical band gaps (*E*_g). The oxide ion polarizability based on refractive index was calculated using relation as reported by Meen and Bhatia [62] [63]

$$\alpha_{02-}(n) = \left[\left(\frac{V_m}{2.52} \right) \left(\frac{n^2 - 1}{n^2 + 2} \right) - \sum \alpha_i \right] q^{-1}.$$
(9)

where V_m is molar volume; *n*, the refractive index; E_g expresses the optical band gap of the glass; $\sum \alpha_i$ denotes the sum of molar cation polarizabilities of all constituents, and q^{-1} is the inverse of total number of oxide ions in the chemical formula of the oxide.

In the case of silica glass, the physical parameters like V_m , $\sum \alpha_i$, and q^{-1} involved in the Equation (9) are given and/or measured as follows.

$$V_m = \frac{\text{Molecular weight}}{\text{Mass density}} = \frac{60.08}{2.65} = 22.672 \text{ mol}^{-1}/\text{cm}^{-3}; \quad \frac{V_m}{2.52} = 8.997;$$

the cation polarizability for SiO₂, $\sum \alpha_i = 0.033 \text{ Å}^3 \{\text{A41, A42}\}$; the inverse of number of oxide ions in SiO₂, $q^{-1} = 0.504 \times 10^{-24} \text{ cm}^3/\text{ions}$. It should be noted that the value of "q" is the number of moles of oxide ions in SiO₂ multiplied by Avogado numbers. Substituting all these values into Equation (9), the Equation (9) is reshaped as

$$\alpha_{02-}(n) = \left[8.997 \times \left(\frac{n^2 - 1}{n^2 + 2}\right) - 0.033\right] \times 0.504 \times 10^{-24}.$$
 (10)

The values of *n* and α_{o2} (*n*) for a silica glass [using Equation (10)] were measured from 1.5119 to 1.5111 and 1.3438 to 1.3420, respectively over the temperature range 295 - 373 K using ellipsometer at λ = 633.8 nm. The values of *n* and

 a_{o2} (*n*) relating to silica glass were respectively measured from 1.5276 to 1.5274 and 1.3786 to1.3782, respectively over the temperature range 297 - 322 K using Abbe refractometer at $\lambda = 589.3$ nm.

In the case of soda lime silicate glass, we have

$$V_m = \frac{\text{Molecular weight}}{\text{Mass density}} = \frac{59.34}{2.50} = 23.74 \text{ mol}^{-1}/\text{cm}^{-3}; \quad \frac{V_m}{2.52} = 9.42;$$

 $\sum \alpha_i = 0.1153 \text{ Å}^3$; $q^{-1} = 0.593 \times 10^{-24} \text{ cm}^3/\text{ions}$. Substituting all these values into Equation (9), we obtain new relation as:

$$\alpha_{02}(n) = \left[9.42 \times \left(\frac{n^2 - 1}{n^2 + 2}\right) - 0.1153\right] \times 0.593 \times 10^{-24}$$
(11)

The values of *n* and a_{o2} (*n*) for a soda lime glass [using Equation (11)] were measured from 1.5086 to 1.5065 and 1.6012 to 1.5954, respectively over the temperature range 295 - 373 K using ellipsometer at $\lambda = 633.8$ nm. The values of *n* and a_{o2} (*n*) relating to silica glass were respectively measured from 1.5074 to 1.5070 and 1.5983 to1.5969 over the temperature range 298 - 322 K using Abbe refractometer at $\lambda = 589.3$ nm.

In the case of borosilicate 7059 glass, we have

$$V_m = \frac{\text{Molecular weight}}{\text{Mass density}} = \frac{62.27}{2.76} = 22.562 \text{ mol}^{-1}/\text{cm}^{-3} ; \frac{V_m}{2.52} = 8.953 ;$$

 $\sum \alpha_i = 0.0448 \text{ Å}^3$; $q^{-1} = 0.524 \times 10^{-24} \text{ cm}^3/\text{ions}$. After substituting all these values into Equation (9), we have new equation as

$$\alpha_{02}(n) = \left[8.953 \times \left(\frac{n^2 - 1}{n^2 + 2}\right) - 0.0448\right] \times 0.524 \times 10^{-24}$$
(12)

The values of *n* and $(a_{02}$. (*n*)) for a borosilicate 7059 glass [using Equation (12)] were measured from 1.5296 to 1.5281 and 1.4329 to 1.4288, respectively over the temperature range 295 - 473 K using ellipsometer at $\lambda = 633.8$ nm. The values of *n* and $(a_{02}$. (*n*)) relating to borosilicate 7059 glass were respectively measured from 1.5283 to 1.5281 and 1.4297 to 1.4287 over the temperature range 297 - 322 K using Abbe refractometer at $\lambda = 589.3$ nm.

Similarly, optical band gap based oxide ion polarizability of any silicate glass can be calculated using the following relation [63] [64]

$$\alpha_{02}\left(E_{g}\right) = \left[\left(\frac{V_{m}}{2.52}\right)\left(1 - \sqrt{\frac{E_{g}}{20}}\right) - \sum \alpha_{i}\right]q^{-1}$$
(13)

Using all the values of the optical parameters as mentioned above, the respective relations regarding silica, soda lime, and borosilicate 7059 glasses are printed as:

$$\alpha_{02} \left(E_g \right) = \left[8.997 \times \left(1 - \sqrt{\frac{E_g}{20}} \right) - 0.033 \right] \times 0.504 \times 10^{-24}$$
(14)

$$\alpha_{02}\left(E_{g}\right) = \left[9.42 \times \left(1 - \sqrt{\frac{E_{g}}{20}}\right) - 0.1153\right] \times 0.593 \times 10^{-24}$$
(15)

$$\alpha_{02}\left(E_{g}\right) = \left[8.953 \times \left(1 - \sqrt{\frac{E_{g}}{20}}\right) - 0.0448\right] \times 0.524 \times 10^{-24}$$
(16)

In the case of silica glass, the optical band gap (E_g) using Equation (14) was measured as 9.8 eV at room temperature and it increased to 9.81 at 373 K. The oxide ion polarizability of silica glass a_{o2} . (E_g) using Equation (14) was measured from 1.3427 to 1.3408 over the temperature range 295 - 373 K using ellipsometer at $\lambda = 633.8$ nm. The values of optical band gap relating to silica glass were measured from 9.59 to 9.592 eV over the temperature range 297 - 322 K using Abbe refractometer at $\lambda = 589.3$ nm, and the values of a_{o2} . (E_g) relating to silica glass were measured from 1.3766 to 1.3764 using Abbe refractometer over the same temperature range and at the same wavelength range.

For soda glass, E_{g} was measured from 9.845 to 9.88 eV over the temperature range 295 - 427 K using ellipsometer at λ = 633.8 nm, and the oxide ion polarizability of soda lime silicate glass using Equation (15) was measured from 1.6014 to 1.5941 using ellipsometer over the same temperature range and at the same wavelength. The values of optical band gap and α_{o2} (E_g) relating to soda lime silicate glass were measured respectively from 9.862 to 9.870 eV and from 1.5991 to 1.5972 over the temperature range 298 - 322 K using Abbe refractometer at λ = 589.3 nm. Similarly, the values of optical band gap and α_{02} (E_g) relating to borosilicate 7059 glass [using Equations (8) and (16)] were measured respectively from 9.56 to 9.58 eV and from 1.4223 to 1.4193 over the temperature range 295 -473 K using elipsometer at λ = 633.8 nm. The values of optical band gap (E_g) and α_{o2} (E_g) relating to 7059 glass were measured respectively from 9.574 to 9.58 eV and from 1.4195 to 1.4191 over the temperature range 297 - 322 K using Abbe refractometer at $\lambda = 589.3$ nm. The complete generated temperature dependent data on the oxide ion polarizability regarding investigated silicate glasses are included in Tables 5-7, and are plotted in Figure 3, Figure 5 and Figure 7. It

Table 5. The various physical properties of silica glass: <i>n</i> , R_m , E_g , α_{me} , α_{O2} - (<i>n</i>) and α_{O2} -	$(E_g),$
$A(n)$ and $A(E_g)$ [all the symbols are defined in the text] using ellipsometer and Abbe	re-
fractometer.	

		Ellipsometric										
Temperature (K)	n	R _m	<i>E</i> g (eV)	<i>a_{me}</i> (×10 ⁻²⁴ cm ³)	a ₀₂₋ (n)	a ₀₂₋ (Eg)	A (n)	A (Eg)				
295.0	1.5119	6.802	9.80	2.6992	1.3438	1.3427	0.4272	0.4262				
307.0	1.5119	6.802	9.80	2.6992	1.3438	1.3427	0.4272	0.4262				
321.0	1.5119	6.802	9.80	2.6992	1.3438	1.3427	0.4272	0.4262				
335.0	1.5115	6.797	9.80	2.6972	1.3428	1.3427	0.4262	0.4259				
373.0	1.5111	6.793	9.81	2.6959	1.3420	1.3408	0.4255	0.4245				
				Abbe								
297.0	1.5276	6.976	9.590	2.7683	1.3786	1.3766	0.4586	0.4569				
308.0	1.5275	6.976	9.590	2.7683	1.3786	1.3766	0.4586	0.4569				
322.0	1.5274	6.974	9.592	2.7675	1.3782	1.3764	0.4582	0.4567				



Figure 3. Refractive index based oxide ion polarizability, $a_{02-}(n)$ and optical band gap based oxide ion polarizability, $a_{02-}(E_g)$ of silica glass against temperature using ellipsometry and Abbe refractometer.

Table 6. The various physical properties of soda lime glass: <i>n</i> , R_{m} , E_{g} , a_{me} , a_{O2-} (<i>n</i>) and a_{O2-}
(E_g) , $A(n)$ and $A(E_g)$ [all the symbols are defined in the text] using ellipsometer and
Abbe refractometer.

		Ellipsometric											
Temperature (K)	n	Rm	<i>E</i> g (eV)	ame (×10 ⁻²⁴ cm ³)	a ₀₂₋ (n)	a02- (Eg)	A (n)	A (Eg)					
295.0	1.5086	7.084	9.85	2.8110	1.6012	1.6014	0.6271	0.6269					
309.0	1.5086	7.084	9.85	2.8110	1.6012	1.6014	0.6271	0.6269					
321.0	1.5082	7.079	9.85	2.8090	1.6001	1.6003	0.6264	0.6261					
335.0	1.5079	7.075	9.86	2.8075	1.5992	1.5986	0.6257	0.6254					
375.0	1.5072	6.067	9.87	2.8044	1.5972	1.5962	0.6244	0.6237					
427.0	1.5065	7.058	9.88	2.8010	1.5954	1.5941	0.6232	0.6224					
				Abbe									
298.0	1.5074	7.070	9.862	2.8060	1.5983	1.5991	0.6251	0.6256					
308.0	1.5072	7.067	9.865	2.8040	1.5971	1.5974	0.6244	0.6246					
322.0	1.507	7.065	9.870	2.8036	1.5969	1.5972	0.6242	0.6243					

should be noted that the values of *n* and $(a_{02} (n))$ measured from both optical techniques (**Ellipsometry and Abbe 60 refractometer**) showed agreeable consistency. The interpretation of this data along with the data on optical basicity is given in the **next section**.

				Ellipsome	etric			
Temperature (K)	п	R _m	<i>E</i> g (eV)	<i>a_{me}</i> (×10 ⁻²⁴ cm ³)	a ₀₂₋ (n)	$a_{02-}(E_g)$	A (n)	A (Eg)
295.0	1.5296	7.012	9.560	2.7830	1.4329	1.4223	0.5045	0.4958
309.0	1.5296	7.012	9.560	2.7830	1.4329	1.4223	0.5045	0.4958
321.0	1.5296	7.012	9.560	2.7830	1.4329	1.4223	0.5045	0.4958
337.0	1.5296	7.007	9.563	2.7810	1.4318	1.4219	0.5037	0.4955
373.0	1.5089	7.005	9.570	2.7790	1.4308	1.4209	0.5028	0.4947
425.0	1.5085	7.001	9.572	2.7781	1.4303	1.4205	0.5023	0.4943
473.0	1.5281	6.994	9.580	2.7753	1.4288	1.4193	0.5012	0.4933
				Abbe				
297.0	1.5283	6.9982	9.574	2.7770	1.4297	1.4195	0.5018	0.4935
307.0	1.5282	6.9959	9.580	2.7762	1.4293	1.4191	0.5017	0.4932
322.0	1.5281	6.9936	9.580	2.7752	1.4288	1.4191	0.5010	0.4930

Table 7. The various physical properties of 7059 glass: *n*, R_m , E_g , a_{me} , a_{O2-} (*n*) and a_{O2-} (E_g), *A* (*n*) and *A* (E_g) [all the symbols are defined in the text] using ellipsometer and Abbe refractometer.

4.2. Optical Basicity

The theoretical optical basicity (A_{th}) for the materials under study can be measured using the relation [62] [63]

$$A_{th} = \sum x_i A_i, \qquad (17)$$

where A_i is the individual optical basicity of each of the constituting oxides, and x_i is the molar concentration of the respective constituent oxide.

The optical basicity can be alternatively calculated from the relationship of basicity and molar refractivity. Duffy [65] explained that there is an intrinsic relationship between oxide ion polarizability a_{02} and optical basicity of the glass medium by the following correlation.

This equation indicates that the optical basicity increases with increasing oxide ion polarizability. The newly generated data of optical basicity using Equation (18) is shown in **Tables 5-7** and plotted in **Figure 4**, **Figure 6** and **Figure 8**. The values of optical basicity can also be calculated by the data of oxide ion polarizability a_{O2-} (E_g) [generated by optical band gap (E_g) values] using the relation suggested by Duffy [65]

$$A(E_g) = 1.67 \left[1 - \frac{1}{\alpha_{O2-}(E_g)} \right].$$
 (19)

In the **first step**, the values of optical basicity (refractive index based) of all investigated silicate glasses are calculated theoretically and are printed as **0.48**, **0.5825**, and **0.5085** for the silica glass, soda lime glass, and borosilicate 7059 silicate glass, respectively. The values of optical basicity of all constituting oxides



Figure 4. Refractive index based optical basicity, A(n) and optical band gap based optical basicity, $A(E_g)$ of silica glass versus temperature using ellipsometry and Abbe refractometer.

have been taken from the references [62] [64]. In the **second step**, by the help of temperature dependent refractive index data (both ellipsometric and Abbe: **Tables 2-4**), we have measured the optical basicity values {A(n)} and { $A(E_g)$ } of investigated silicate glasses using the Equations (18) and (19), and the obtained results are included in **Tables 5-7** and are plotted in **Figure 4**, **Figure 6** and **Figure 8**. The measured optical basicity values of investigated glasses at room temperature are given as A(n) = 0.4272, 0.6271, 0.5045 for the silica, soda lime silicate, and borosilicate 7059 glasses, respectively (using ellipsometry), and A(n) = 0.4586, 0.6251, 0.5018 for the silica, soda lime silicate, and borosilicate basicity A(n) are very close to the theoretically calculated optical basicity (A_{th}) values regarding silica, soda lime, and borosilicate 7059 glasses.

As shown in **Tables 5-7**, refractive index (*n*) is decreased from 1.5119 to 1.5111 and optical band gap (E_g) of silica glass is increased from 9.8 to 9.81 eV over the temperature range 295 to 373 K using ellipsometer; and the n-value is decreased from 1.5276 to 1.5274 and E_g is increased from 9.59 to 9.592 eV over the temperature range 297 to 322 K using Abbe refractometer. In the case of so-da lime silicate glass, refractive index (*n*) is decreased from 1.5086 to 1.5065 and optical band gap (E_g) of soda lime glass is increased from 9.845 to 9.88 eV over the temperature range 295 to 427 K using ellipsometer; and the *n*-value is decreased from 1.5074 to 1.5070 and E_g is increased from 9.862 to 9.870 eV over

the temperature range 298 to 322 K using Abbe refractometer. The refractive index (*n*) associated with borosilicate 7059 glass is decreased from 1.5296 to 1.5281 and optical band gap (E_g) is increased from 9.56 to 9.88 eV over the temperature range 295 to 473 K using ellipsometer; and the n-value is decreased from 1.5283 to 1.5281 and E_g is increased from 9.574 to 9.58 eV over the temperature range 297 to 322 K using Abbe refractometer.

Figure 4, Figure 6 and **Figure 8** also show that refractive index based optical basicity is decreased from 0.4272 to 0.4255 and $A(E_g)$ is decreased from 0.4262 to 0.4245 (silica glass) over the temperature range 295 - 373 K using ellipsometer; A(n) is decreased from 0.4586 to 0.4582 and $A(E_g)$ is decreased from 0.4569 to 0.4567 over the temperature range 297 - 322 K using Abbe refractometer. In the case of soda lime silicate, A(n) is decreased from 0.6271 to 0.6232 and $A(E_g)$ is decreased from 0.6269 to 0.6224 over the temperature range 295 - 427 K using ellipsometer; A(n) is decreased from 0.6251 to 0.6242 and $A(E_g)$ is decreased from 0.6256 to 0.6243 over the temperature range 298 - 322 K using Abbe refractometer. The value of A(n) associated with borosilicate 7059 glass is decreased from 0.5045 to 0.5012 and $A(E_g)$ is decreased from 0.4958 to 0.4933 over the temperature range 295 - 473 K using ellipsometer; A(n) is decreased from 0.4935 to 0.4930 over the temperature range 297 - 322 K using Abbe refractometer.

Here, let me interpret the reported data on oxide ion polarizability and optical basicity regarding silicate glasses briefly as follows.

Figures 3-8 show that the optical band gap (E_g) value of silicate glasses is increased as temperature is increased. Such an increase can be explained by suggesting that the non-bridging oxygen (NBO) ions content decreases and leading to an increase in the value of E_{g} . Generally, the less polarizable cations in the glass matrix does not cause effective retardation of the light propagation through the vitreous network to lead to an increase in the refractive index (n). It is also very clear from **Figure 3**, **Figure 5** and **Figure 7** that both of refractive index and energy band gap based oxide ion polarizability have the same decreasing trend as the temperature increases. The decreasing trend occurs due to decreasing amount of non-bridging oxygen (**NBO**), and that trend is due to small single bond strength and small polarizability associated with the silicate glass system.

Next, the disagreement between A_{th} and experimental basicity value might occur as a result of significant structural changes such as change in coordination number. It can be clearly seen from the obtained values of optical basicity (**Figure 4**, **Figure 6** and **Figure 8**) that the optical basicity on the basis of *n* and E_g is decreased as temperature is increased. The decreasing trend in both optical basicity values indicates that the glasses prepared are acidic in nature and have very small amount of polarizability. The decrease in the optical basicity values (due to increase in temperature) is due to decreasing negative charge on the oxygen atoms that lead to decreasing covalency in the cation-oxygen bonding associated with the silicate glasses [66].



Figure 5. The plot of refractive index based oxide ion polarizability, a_{02} . (*n*) and optical band gap based oxide ion polarizability, a_{02} . (*E*_g) of soda lime glass against temperature using ellipsometry and Abbe refractometer.



Figure 6. The plot of refractive index based optical basicity, A(n) and optical band gap based optical basicity, $A(E_g)$ of soda lime glass versus temperature using ellipsometry and Abbe refractometer.



Figure 7. The plots of refractive index based oxide ion polarizability, $a_{02-}(n)$ and optical band gap based oxide ion polarizability, $a_{02-}(E_g)$ of 7059 glass against temperature using ellipsometer and Abbe refractometer.



Figure 8. The plots of refractive index based optical basicity, A(n) and optical band gap based optical basicity, $A(E_g)$ of 7059 glass against temperature using ellipsometry and Abbe refractometer.

So low values of optical basicity and of oxide ion polarizability [67] [68] [69] suggest that the silicate glasses are not novel glasses (optical functional glasses) for non-linear optical (NLO) devices or for three dimensional displays [70] [71] [72] [73] and biological systems [74]. However, any change of oxygen bonding in the glass network, for instance, the formation of non-bridging oxygen, can change the characteristic absorption edge leading to an increase in both the electronic and oxide ion polarizability in the network for NLO applications.

4.3. Oxide Ion Polarizability, Optical Basicity and Electronegativity

Since any methodology such as atomic force microscopy for the purpose to evaluate the electronegativity of any material (or any chemical compound) cannot take into account the real crystal structure of the material, *i.e.*, it does not estimate the real distances of the chemical bonds in the structure under consideration. In contrast oxide ion polarizability and optical basicity are based on the experimentally obtained materials constants such as refractive index or energy gap which closely are related to the real electronic structure of the oxides. So, we have measured the values of electronegativity of silicate glasses by using the experimental data on oxide ion polarizability and optical basicity being generated by the reported data on ellipsometer and Abbe refractometer over the temperature range 295 - 473 K.

There are many empirical models in the literature [75] [76] [77] [78] [79] which relate electronegativity to oxide ion polarizability or to optical basicity of a material, but the values of electronegativity, or of oxide ion polarizability [a_{02} . (*n*)] or of optical basicity of silicate glasses calculated by most of the models seem to be too large. So, we have followed the following approach for this problem.

According to Reddy *et al.* [76], the following empirical relation between oxide ion polarizability and average electronegativity is as follows:

$$\alpha_{\rm O2-}(n) = 4.624 - 0.7569\zeta_{av},\tag{20}$$

where, ζ_{av} is the average electronegativity of the material (or glass).

It is with great regret that this equation is not accurate to calculate electronegativity of a glass and so it needs some amendment. Accordingly a modified formula with a minor change in Equation (20) can be rewritten as the following:

$$\alpha_{02}(n) = 4.624 - 0.6430\zeta_{av} \tag{21}$$

This new Equation (21) measures the values of electronegativity of silicate glasses under study which are in agreement with the values of electronegativity as measured by other researchers. We have also developed our own model known as **Zahid model correlating oxide ion polarizability and electronega-tivity** and that model now exists as

$$0.92\zeta_{av} = 6.039 - \alpha_{O2}(n) \tag{22}$$

The values of electronegativity of silicate glasses as measured by Equation (21) are in good agreement with the values measured using Equation (22). All the values of electronegativity measured using Equations (21) and (22) are included in **Tables 8-10**, and are plotted in **Figures 9-14**.



Figure 9. The plot of refractive index based electronic oxide ion polarizability and optical basicity of silica glass as a function of temperature using ellipsometry and Abbe refractometer.



Figure 10. The plots of refractive index based electronic oxide ion polarizability and optical basicity of soda lime glass versus temperature using ellipsometry and Abbe refractometer.



Figure 11. The plots of refractive index based electronic oxide ion polarizability, a_{O2} . (*n*) and optical basicity, A(n) of 7059 glass using ellipsometry and Abbe refractometer.



Figure 12. The plot of electronegativity of silica glass measured from oxide ion polarizability (Zahid and Reddy models) against temperature using ellipsometry and Abbe refractometer.



Figure 13. The plot of electronegativity of silica glass measured from optical basicity, A(n) [Zahid and Reddy models] against temperature using ellipsometry and Abbe refractometer.



Figure 14. The changes in electronegativity of soda lime glass measured from oxide ion polarizability (Zahid and Reddy models) against temperature using ellipsometer and Abbe refractometer.

Table 8.	Temp	eratu	re depen	dent	data	of sil	ica	glass	regard	ling	n, A	(<i>n</i>),	$a_{\text{O2-}}$	(<i>n</i>),	ζ_{av} (A_n),
ζ_{av} { $a_{\mathrm{O2-}}$ ((<i>n</i>)} [a]	ll the	symbols	are	define	ed in	the	text]	using	ellij	psom	neter	and	abbe	e ref	rac-
tometer.																

		Ellipsometric									
Temperature (K)	л	$a_{02-}(n)$ (×10 ⁻²⁴ cm ³)	A(n)	ζ₂ν (A₂) Zahid model	ζ _{av} (A ₂) Reddy model	ζ _{sv} {a ₀₂₋ (n)} Zahid model	ζ _{#ν} { <i>α</i> ₀₂₋ (<i>n</i>)} Reddy model				
295.0	1.5119	1.3438	0.4272	5.5317	5.1022	5.1035	5.1014				
307.0	1.5119	1.3438	0.4272	5.5317	5.1022	5.1035	5.1014				
321.0	1.5119	1.3438	0.4272	5.5317	5.1022	5.1035	5.1014				
335.0	1.5115	1.3428	0.4262	5.5426	5.1066	5.1046	5.1030				
373.0	1.5111	1.3420	0.4255	5.5504	5.1097	5.1054	5.1042				
				Abt	De						
297.0	1.5276	1.3786	0.4586	5.2111	4.9645	5.0657	5.0473				
308.0	1.5275	1.3786	0.4586	5.2111	4.9645	5.0657	5.0473				
322.0	1.5274	1.3782	0.4582	5.2149	4.9662	5.0661	5.0479				

Table 9. Temperature dependent data of soda lime glass regarding *n*, *A* (*n*), a_{02-} (*n*), ζ_{av} (*A_n*), ζ_{av} { a_{02-} (*n*)} [all the symbols are defined in the text] using ellipsometer and abbe refractometer.

		Ellipsometric								
Temperature (K)	n	a ₀₂₋ (n) (×10 ⁻²⁴ cm ³)	A(n)	ζ", (A,) Zahid model	ζ _{av} (A _z) Reddy model	ζ _{4ν} { <i>a</i> ₀₂₋ (<i>n</i>)} Zahid model	ζ _{av} { <i>α</i> ₀₂₋ (<i>n</i>)} Reddy model			
297.0	1.5086	1.6012	0.6271	4.0393	4.2251	4.8237	4.7011			
309.0	1.5086	1.6012	0.6271	4.0393	4.2251	4.8237	4.7011			
321.0	1.5082	1.6001	0.6264	4.0429	4.2282	4.8249	4.7028			
335.0	1.5079	1.5992	0.6257	4.0469	4.2312	4.8259	4.7042			
375.0	1.5072	1.5972	0.6244	4.0531	4.2370	4.8280	4.7073			
427.0	1.5065	1.5954	0.6232	4.0592	4.2422	4.8300	4.7101			
				Ab	be					
298.0	1.5074	1.5983	0.6251	4.0495	4.2339	4.8268	4.7056			
308.0	1.5072	1.5971	0.6244	4.0531	4.2370	4.8282	4.7075			
322.0	1.5070	1.5969	0.6242	4.0541	4.2378	4.8284	4.7078			

Table 10. Temperature dependent data of 7059 glass regarding *n*, *A* (*n*), α_{O2-} (*n*), ζ_{av} (*A_n*) and ζ_{av} { α_{O2-} (*n*)} [all the symbols are defined in the text] using ellipsometer and abbe refractometer.

	Ellipsometric								
Temperature (K)	Д	$a_{02-}(n)$ (×10 ⁻²⁴ cm ³)	A (n)	ζ₄ν (A₂) Zahid model	ζ _{av} (A ₂) Reddy model	ζ _{av} {a ₀₂₋ (n)} Zahid model	ζ _{aν} { <i>a</i> ₀₂₋ (<i>n</i>)} Reddy model		
295.0	1.5296	1.4329	0.5045	4.8143	4.7631	5.0066	4.9628		

Continued											
309.0	1.5296	1.4329	0.5045	4.8143	4.7631	5.0066	4.9628				
321.0	1.5296	1.4329	0.5045	4.8143	4.7631	5.0066	4.9628				
337.0	1.5292	1.4318	0.5037	4.8206	4.7666	5.0078	4.9645				
373.0	1.5289	1.4308	0.5028	4.8277	4.7705	5.0089	4.9661				
425.0	1.5285	1.4303	0.5023	4.8317	4.7727	5.0095	4.9669				
473.0	1.5281	1.4288	0.5012	4.8404	4.7775	5.0111	4.9692				
	Abbe										
297.0	1.5283	1.4297	0.5018	4.8357	4.7749	5.0101	4.9678				
307.0	1.5282	1.4293	0.5017	4.8364	4.7753	5.0105	4.9684				
322.0	1.5281	1.4288	0.5017	4.8420	4.7784	5.0111	4.9692				

Similarly, in the case of optical basicity and electronegativity, we have only followed the model by Reddy [76] [77] who have derived the following empirical relationship for the refractive index based optical basicity, A(n) associated with the electronegativity of a material:

$$A(n) = 1.59 - 0.2279\zeta_{1av},$$
(23)

where ζ_{1av} is the average electronegativity of the material. This formulation is very suitable for the silicate glasses. Here, we have also established the empirical relation between optical basicity and average electronegativity known as **Zahid model and that** exists as:

$$\frac{2}{A(n)} = \zeta_{av} - 0.85 \tag{24}$$

The values of electronegativity using Zahid model [Equation (24)] are also in good agreement with the values measured using other models [Equation (23)]. All the values of electronegativity measured using Equation (23) and Equation (24) are tabulated in **Tables 8-10**, and are plotted in **Figures 9-17**.

Figures 9-17 show that for a silica glass, both the refractive index based optical basicity.

 a_{O2-} (*n*) and *A* (*n*) are decreased from 1.3438 to 1.3420 and from 0.4272 to 0.4255, respectively; while the values of (ζ_{1av}) calculated) using Zahid model and other models on the base of a_{O2-} (*n*) are increased from 5.1035 to 5.1054 and from 5.1014 to 5.1042, respectively; and are increased from 5.5317 to 5.5504 and from 5.1022 to 5.1097 on the base of *A* (*n*), respectively over the temperature range 295 - 373 K using ellipsometer. Both the refractive index based optical basicity a_{O2-} (*n*) and *A* (*n*) relating to silica glass are decreased from 1.3786 to 1.3782 and from 0.4586 to 0.4582, respectively; while the values of (ζ_{1av}) calculated) using Zahid model and other models on the base of a_{O2-} (*n*) are increased from 5.0657 to 5.0661 and from 5.0473 to 5.0479, respectively; and are increased from 5.2111 to 5.2149 and from 4.9645 to 4.9662 on the base of *A* (*n*), respectively over the temperature range 297 - 322 K using Abbe refractometer.



Figure 15. The changes in electronegativity of soda lime glass measured from optical basicity, A(n) [Zahid and Reddy models] against temperature using ellipsometry and Abbe refractometer.



Figure 16. The plots of electronegativity of 7059 glass measured from oxide ion polarizability {Zahid and Reddy models} versus temperature using ellipsometry and Abbe refractometer.



Figure 17. The plots of electronegativity of 7059 glass measured from optical basicity, *A* (*n*) {Zahid and Reddy models} versus temperature using ellipsometry and Abbe refractometer.

The values of $a_{O2-}(n)$ and A(n) relating to soda lime glass are decreased from 1.6012 to 1.5954 and from 0.6271 to 0.6232, respectively, while the calculated values of (ζ_{1av}) using Zahid model and other models on the base of $a_{O2-}(n)$ are increased from 4.8237 to 4.830 and from 4.7011 to 4.7101, respectively; and are increased from 4.0393 to 4.0592 and from 4.2251 to 4.2422 on the base of A(n), respectively over the temperature range 297 - 427 K using ellipsometer. Both the refractive index based optical basicity $a_{O2-}(n)$ and A(n) relating to soda lime glass are decreased from 1.3786 to 1.3782 and from 0.4586 to 0.4582, respectively; while the calculated values of (ζ_{1av}) using Zahid model and Reddy model on the base of $a_{O2-}(n)$ are increased from 5.0657 to 5.0661 and from 4.9645 to 4.9662 on the base of A(n), respectively over the temperature range decreased from 5.2111 to 5.2149 and from 4.9645 to 4.9662 on the base of A(n), respectively over the temperature.

In the case of borosilicate 7059 glass, both a_{02} . (*n*) and *A* (*n*) are decreased from 1.4329 to 1.4288 and from 0.5045 to 0.5012, respectively; while the values of electronegativity (ζ_{1av}) calculated on the base of a_{02} . (*n*) using Zahid model and other models, are increased from 5.0066 to 5.0111 and from 4.9628 to 4.9692, respectively; and are increased from 4.8143 to 4.8404 and from 4.7631 to 4.7775, respectively, measured on the base of *A* (*n*) over the temperature range 295 - 473 K using ellipsometer. Both the refractive index based optical basicity α_{02} . (*n*) and *A* (*n*) relating to borosilicate 7059 glass are decreased from 1.4297 to 1.4288 and from 0.5018 to 0.5017, respectively; while the calculated values of (ζ_{1av}) using Zahid model and Reddy model on the base of a_{O2-} (*n*), are increased from 5.0101 to 5.0111 and from 4.9678 to 4.9692, respectively, and are increased from 4.8357 to 4.8420 and from 4.7749 to 4.7784 on the base of *A* (*n*), respectively over the temperature range 297 - 322 K using Abbe refractometer.

Generally, as the difference in electronegativity between any two elements within a chemical compound approaches zero, the bond becomes less ionic and more covalent. As a difference of electronegativity exceeds zero and approaches 1.7, the bond becomes more polar covalent, and if the difference gets increasingly larger than 1.7, a bond between any two elements is more likely to be ionic [80] [81].

Since the calculated values of electronegativity are found in the range 4.0393 - 5.5504 for the silicate glasses (silica, soda lime, and borosilicate 7059), so from the above analysis all these glasses have an ionic character.

5. Summary and Conclusions

We have used two optical techniques: ellipsometry at a wavelength, λ of 632.8 nm and Abbe's refractometer at λ of 589.3 nm, to measure the optical and physical properties of silicate glasses over the temperature range from 295 to 473 K.

The values of refractive index (*n*) for silica, soda lime and borosilicate 7059 glasses are decreased but the optical band gap (E_g) values are increased over the temperature range 295 - 473 K using ellipsometer and Abbe refractometer. The values of oxide ion polarizability [a_{o2} . (*n*) and a_{o2} . (E_g)] regarding silicate glasses have got the decreasing trend over the temperature range 295 - 373 K using ellipsometer or Abbe refractometer. The differences between the values of a_{o2} . (E_g) and a_{o2} . (*n*) can be explained by the existence of the localized density states in band gap energy based on the theory of conduction of non-crystalline.

We have also measured refractive index and optical band gap based optical basicity $[A (n) \text{ and } A (E_g)]$ relating to silicate glasses. Experimental results show that A (n) and $A (E_g)$ of silica glasses are decreased over the temperature range 295 - 373 K using ellipsometer or Abbe refractometer. **Further,** we have found that for silicate glasses, the values of electronegativity (ζ_{1av}) measured using Zahid model and Reddy model on the base of a_{O2} . (*n*) or on the base of A (n) are increased over the temperature range 295 - 373 K using both the optical instruments.

From these findings, we conclude that the optical band gap (E_g) values of silicate glasses increase as temperature increases. Such an increase can be explained by suggesting that the non-bridging oxygen (NBO) ions content is decreased due to an increase in the value of E_g . It is also very clear that both oxide ion polarizability and optical basicity have the same decreasing trend as the temperature is increased. This trend occurs due to the decreasing amount of non-bridging oxygen (NBO), and that happens due to small single bond strength and a little change in the electronic polarizability associated with the silicate glass system. **Next**, the calculated values of electronegativity are found to be in the range 4.0393 - 5.5504 for the reported silicate glasses. And from this result we **conclude** that all these glasses have an ionic character.

Also, the decreasing trend in both optical basicity and of oxide ion polarizability values suggest that the silicate glasses are not novel glasses (optical functional glasses) for non-linear optical (NLO) devices or for three dimensional displays and for biological systems.

Acknowledgements

This research was done in the EEE department at Imperial College London by private fundings. The author would like to acknowledge the invaluable assistance provided by the technicians of the EEE laboratories.

Funding Information

Sampling and experimental process were financially supported by the author.

Ethics

This article is original and contains unpublished material. Author declares that there is no ethical issue and no conflict of interest that may arise after the publication of this manuscript.

Conflicts of Interest

The author declares no conflicts of interest regarding the publication of this paper.

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