

# Role of Al(III) in Zn-Based Electrode Synthesis for Supercapacitor

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Supercapacitor is considered as one of the most promising energy storage systems because of its high power density, long life and low production cost. Electrode materials play important roles in the performance of Supercapacitor (SC). In this study, Zn-based hydrotalcite structure materials are prepared by hydrothermal method. The influence of Zn/Al ratio in precursors on electrochemical properties of electrode materials is investigated. The results show that Al(III) promotes the formation of relatively ordered active substances and participates in redox reaction on electrode surface. Specific capacitance of Zn-based electrode reaches  $2557 \text{ F} \cdot \text{g}^{-1}$  (1.0 A·g<sup>-1</sup>) at Zn/Al molar ratio of 1:1 in precursors. This method is simple and environmentally friendly. The electrode exhibits excellent electrochemical activity and stability, showing this material application prospect for supercapacitor.

# **Keywords**

Supercapacitor, Hydrotalcite, Zn-Based Electrode, Aluminium Ion Doping

# **1. Introduction**

The continuous progress of our society asks for green and sustainable development of economy, and also puts forward higher requirements for energy storage and conversion [1]. Typical energy storage devices include batteries and supercapacitors. Supercapacitor (or electrochemcial capacitors, SC) is novel energy storage system between traditional capacitor and rechargeable battery with high efficiency, practical, and environmental friendly [1] [2] [3] [4] [5]. Compared with traditional capacitors, SC has larger capacity, higher energy, wider operating temperature range and extremely long theoretical life. While compared with batteries, it has higher power density and almost no pollution. SC can be divided into double layer capacitors and pseudocapacitors. Charge storage in SC is mainly based on surface processes or surface reactions of electrode materials [2] [3]. Researches on SC mainly focus on how to increase energy density while maintain high power density in recent years. Carbon materials, conductive polymer materials and metal oxide/hydroxide materials are all applied for electrode materials to achieve above goals [4] [5].

Layered double metal hydroxides material (Layered double hydroxides, LDHs), including hydrotalcite and hydrotalcite-like compounds, is a kind of ideal SC electrode materials [5]. Large theoretical specific surface area of this material can provide certain double-layer capacitance. Mean while, active metal ions on surface of layers can act as active sites for electrochemical reactions, providing pseudocapacitance. Only a few atoms thickness of LDHs nanosheet makes this material not only have relatively high specific surface area, but also can fully expose electrochemical active sites. In LDHs structure, bivalent and trivalent metal ions (M(II) and M(III) ions) coordinate with OH groups located at the apex of octahedral and form nanosheet plane. Replacement of M(III) ions makes entire laminate positively charged, which must be balanced by anions with equal charge located interlayers. Since the proportion of M(II) and M(III) ions can be carried out according to the actual needs of electrochemical activity and specific conditions [5] [6].

In our preliminary study, Zn-based electrode materials were prepared which had mainly  $Zn_4(OH)_6CO_3 H_2O$  crystal structure and showed excellent electrochemical activity [7]. However, the role of Al(III) is not clear both in synthesis process and in electrochemical reaction. Whether can this layered active material still be obtained without Al(III)? Whether does Al(III) participate in synthesis process, as well as in the electrode activity and stability? These questions are all worthy of in-depth study to optimize electrode structure and clarify synthesis process and activity mechanism. In order to answer these questions, the materials with different Al(III) amount in precursors are prepared and their electrochemical activity, stability, composition and structure are analyzed.

#### 2. Experimental

#### 2.1. Preparation of Electrode

Zn-based electrode materials are prepared by hydrothermal method as literature [6] [7]. Typical synthesis steps are as follows. Zinc nitrate and aluminum nitrate are used as precursors and urea as precipitant. Moles of urea added are 5 times of that of metal ions. The above substances are dissolved in 36 ml deionized water and then transferred to a 50 ml hydrothermal kettle. Pretreated nickel foam  $(1 \times 1 \text{ cm})$  is added in. After 12 h hydrothermal reaction at 120°C, the product is washed to neutral, and dried at 60°C overnight. The sample is labeled as LHDs-a:b, with a:b of Zn(II) and Al(III) molar ratio in precursor.

Active carbon (AC) electrode on  $1 \times 1$  cm<sup>2</sup> nickel foam is prepared to assemble LDHs//AC supercapactior, in which mass proportion of AC (BP2000), ace-

tylene black and Poly tetra fluoroethylene (PTFE) is 8:1:1.

#### 2.2. Characterization

Field emission scanning electronic microscope (FESEM, JEOLJEM-6360LV, Japan) is used to analysize morphology of sample. The accelerating voltage is 20 -30 kV. The sample was processed with gold spraying before testing. X-ray diffraction (XRD) test is performed on X'pert proSuper ray diffractometer of PAN Analytical (Holland). Copper target is used as light source ( $\lambda = 1.5432$  nm) with graphite monochromator. The tube voltage is 40 kV, while tube current is 100 mA.

Three electrodes system is used to analyze electrode in 6 mol·L<sup>-1</sup> KOH electrolyte with as-prepared LDHs sample, Pt mess, and HgO/Hg electrode are used as working electrode, counter electrode, and reference electrode, respectively. Electrochemical performance of LDHs electrode and LDHs//AC was evaluated on CHI660E and LANHE electrochemical workstation (China).

#### 3. Results and Discussion

## 3.1. Role of Al(III) in Electrochemical Activities of Electrode

Influences of Zn/Al ratio on electrochemical properties of LDHs electrode are investigated, as shown in Figure 1(a). It can be seen that cyclic voltammetry curves (CV)  $(0.01 \text{ V} \cdot \text{s}^{-1})$  show obvious Redox peaks, indicating that all electrodes present typical Faraday pseudocapacitance characteristic [8]. Current response reaches maximum value for the sample with ratio of Zn/Al 1:1 in precursors, while the minimum of peak potential difference occurs on LDHs-5:1. As the ratio of Zn/Al decreased, the peak potential difference increased. The above results indicate that increase of Al(III) in precursor is conducive to obtaining higher current response value to some extent however peak potential difference increases. Therefore, in order to optimize both electrochemical activity and reversibility of electrode, it is necessary to analyze appropriate Al(III) amount in precursor.

**Figure 1(b)** shows galvanostatic charge and discharge curves of as-prepared LDHs materials at current density of 1 A· $g^{-1}$ . It can be seen that all curves present obvious charge-discharge platforms, corresponding to Redox peaks in CV test, indicating Faraday pseudocapacitance characteristics of electrode within investigated potential range [9].

Discharging time of electrode  $(1.0 \text{ A} \cdot \text{g}^{-1})$  increases with Al(III) ratio in precursors. Specific capacitance curves under different current densities are shown in **Figure 2(a)**. It can be seen that the maximum specific capacitance achieve about 2557 F·g<sup>-1</sup> on LDHs-1:1 electrode, which is much higher than that of other Zn/Al proportional electrode materials, and also higher than that of other hydrotalcite type materials reported [10] [11]. Energy density and power density of LDHs electrode are shown in **Figure 2(b)**. LDHs electrode material has higher energy and power performance when Zn/Al ratio is 1:1 in precursors. Energy density of LDHs-1:1 reaches 89.5 Wh·kg<sup>-1</sup> when power density is 2000 W·kg<sup>-1</sup>.



EIS results of LDHs electrode are shown in **Figure 3**. EIS spectrum present linear tendency with an approximately 90° angle to X-axis at low frequency region,

**Figure 1.** (a) CV curves at 0.01 V·s<sup>-1</sup> and (b) galvanostatic charge/discharge curves at 1 A·g<sup>-1</sup> of LDHs electrodes with different Zn/Al ratio in precursors.



**Figure 2.** (a) Specific capacitance of electrodes derived from galvanostatic charge/discharge curves and (b) Ragone plots of LDHs electrodes with different Zn/Al ratio in precursors.



Figure 3. Nyquist plots of LDHs electrodes with different Zn/Al ratio in precursors (Right: Local enlarged image).

indicating good capacity performance of electrodes [12]. Further analysis shows that the slope of the line in low frequency region decreases with increase of Al(III) amount in precursor. At the same time, equivalent series resistance increases at first and then decreases with Zn/Al ratio in precursors [6].

#### 3.2. Role of Al(III) in Composition and Structure of Materials

XRD patterns of products are shown in **Figure 4**. It can be seen that diffraction peaks strength increase slightly with Al(III) amount in precursors, however there is no significant change of peak position in XRD patterns. Most of these diffraction peaks correspond to characteristic diffraction peaks of  $Zn_4(OH)_6CO_3$ ·H<sub>2</sub>O (JCPDS No.011-0287). Some possible crystalline phases, such as AlO(OH) (JCPDS No.021-1307) and  $Zn(OH)_2$  (JCPDS No.089-0138), are not observed in XRD spectrum. This result shows that Al(III) in precursor promotes the formation of  $Zn_4(OH)_6CO_3$ ·H<sub>2</sub>O crystal phase.

Further comparison with  $Zn_4(OH)_6CO_3 H_2O$  standard spectrum shows that peak positions of as-products move to higher  $2\theta$  values slightly, accompanied by d values decrease simultaneously (**Table 1**). For example, the diffraction peak at  $2\theta = 12.829^\circ$  in  $Zn_4(OH)_6CO_3 H_2O$  standard spectrum has moved to  $2\theta =$  $12.965^\circ$  when Zn/Al ratio 5:1 in precursors. It can be speculated from these



Figure 4. XRD patterns of LDHs with different Zn/Al ratio in precursors.

Table 1. Calculations of d value of LDHs samples from XRD characteristics.

Sample	2 <i>0</i> (°)	d (nm)	Sample	2 <i>θ</i> (°)	d (nm)
LDHs-5:1	12.965	0.6829	LDHs-1h <sup>a</sup>	12.922	0.6851
LDHs-4:1	12.907	0.6859	LDHs-2h	12.926	0.6849
LDHs-3:1	12.891	0.6868	LDHs-3h	12.892	0.6867
LDHs-2:1	12.934	0.6845	LDHs-8h	12.788	0.6923
LDHs-1:1	12.931	0.6846	LDHs-12h	12.891	0.6868
			LDHs-20h	12.815	0.6908

a LDHs-1h: LDHs-3:1 product with 1 h hydrothermal process at 120°C.

results that a small amount of Al(III) is doped into structure by substituting Zn(II), which increases positive charge amount of laminate and promotes the attraction between laminate cations and inter-laminar anions. Substitution of Al(III) results in the formation of a bimetallic composite hydroxide structure in which Zn(II) ions act as the main cations and a small amount of Al(III) doped, that is, a hydrotalcite-like structure material. **Table 1** also shows that with increase of Al(III) ratio in precursors, the corresponding d value of the main diffraction peak does not change significantly, which indicates that Al(III) doping in Zn<sub>4</sub>(OH)<sub>6</sub>CO<sub>3</sub>·H<sub>2</sub>O structure is very limited.

Since Al(III) doping in  $Zn_4(OH)_6CO_3 H_2O$  structure is very limited, is it possible to synthesize this highly ordered product directly without Al(III) in precursors? In order to answer this question, similar synthesis process is carried with Zn(II) or Al(III) as single precursor, respectively. The morphology analyses of products are shown in **Figure 5**.

As we can see in Figure 5, as-products show thin layer on foam Ni surface



Figure 5. SEM images of as-synthesized products with different Zn/Al ratio of ((a), (b)) 1:0 and ((c), (d)) 0:1.

when there is only Zn(II) as precursor, along with some gathered dendrite flower-like clusters by aggregation and growth of nanoparticles. This morphology is completely different from that of LDHs. When only Al(III) present as precursor, the product is flower-like structure formed by self-assembly of 20 - 40 nm thickness nanosheets. This results proves that Al(III) acts as structure-directing agent guide and plays a key role and is vital to promote the formation of active layered structure product on foam Ni in hydrothermal synthesis process [13].

# 3.3. Contribution of Al(III) in Structure to LDHs Electrochemical Performance

In order to investigate whether Al(III) ions in structure would contribute to electrochemical activity of electrode, CV tests at 0.01 V·s<sup>-1</sup> are performed on samples with Zn/Al ratio of 0:1, 1:0, and 1:1 in precursors, respectively. As shown in **Figure 6(a)**, Redox peaks with high current response values are all observed. By comparison, it can be induced that Al(III) in structure contributes to electrode performance even if the doping is limited in composition. However, it must be noticed that the corresponding peak potential difference is larger for the sample with Al(III) as precursor only, indicating that Redox reaction on that shows lower reversibility. Therefore, limitation of Al(III) doping is practically beneficial to cycling stability of electrode and supercapacitor.

**Figure 6(b)** shows the cyclic stability test results of LDHs//AC asymmetric supercapacitors. As Al(III) amount increases in precursors, the cyclic stability of capacitor increases first and then decreases. After 5000 cycles of constant current test at current density of 0.5 g·A<sup>-1</sup>, supercapacitor assembled with LDHs-2:1 shows the best stability and remains 79.5% of initial specific capacitance. It can be concluded that it is possible to achieve higher cyclic stability by adjusting the composition and structure of Zn-based LDHs electrode materials with proper Al(III) present.



**Figure 6.** (a) CV curves at 0.01 V·s<sup>-1</sup> of LDHs electrodes and capacity retention after 5000 cycles for LDHs//AC supercapacitors with different Zn/Al ratio in precursors.

#### 4. Conclusion

Zn-based LDHs material with limited Al(III) substituted are synthesized by feasible hydrothermal method. The increase of Al(III) ratio in precursors is beneficial to obtain higher specific capacitance. Specific capacitance of LDHs-1:1 arrives about 2557  $F \cdot g^{-1}$  (1.0  $A \cdot g^{-1}$ ). When power density is 2000  $W \cdot kg^{-1}$ , energy density of the electrode reaches 89.5 Wh $\cdot kg^{-1}$ . Although Al(III) in LDHs structure is limited, it promotes relatively ordered electrode structure and participates in electrochemical reaction on electrode surface. The cyclic stability of electrode can also be improved with proper Al(III) doping. In further research, the actual doping amount of Al(III) in crystalline structure of LDHs material will be further refined, and the specific location of Al(III) in structure will be identified to optimize the electrode performance.

## **Conflicts of Interest**

The authors declare no conflicts of interest regarding the publication of this paper.

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