

Synthesis of Tenoltrifluoroacetone Composite Doped with Terbium, Dysprosium and Europium Encapsulated in Silica Oxide Matrix

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Abstract

The synthesis of the thenoyltrifluoroacetone compound doped with terbium, dysprosium and europium encapsulated in a silica matrix (TTA:Tb:Dy:Eu@SiO₂) were performed by the sol-gel method. The precursors to obtain the vitreous phase (SiO₂) were: Tetraethylorthosilicate (TEOS, C₈H₂₀O₄Si, 98%, Aldrich), and ethyl alcohol (CH₃CH₂OH, 99.5%, Meyer), distilled water and 0.05 ml of hydrochloric acid (HCl, Meyer). The sample with molar ratio 20:80 TTA:Tb:Dy:Eu@SiO₂ has the best emission intensity. Thermogravimetric analysis (TGA) shown that silica encapsulated samples decompose at lower temperatures than pure TTA:Tb:Dy:Eu luminescent material. Fourier Transform Infrared (IR-TF) shown the characteristic Si-O-Si bands that are presented at a wavelength of 1049, 853 and 440 cm⁻¹ confirming that the luminescent material is encapsulated in a silica matrix, finally X-ray diffraction (XRD) shown that TTA:Tb:Dy:Eu@SiO₂ composite is amorphous.

Keywords

Synthesis, Luminescence, Composite, Sol-Gel, Terbium, Dysprosium and Europium

1. Introduction

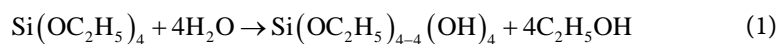
Luminescent materials are presented in a wide variety of materials such as inorganic crystals, glasses, ceramics and organic compounds [1] because the technological advance of luminescent materials has been growing since the applications that these have are diverse and are used in optoelectronics, medical appli-

cations, textiles, and others applications. In this way, the study of organic binders doped with rare earth (such as β -diketones) is growing rapidly since it is possible to increase considerably the intensity of emission of compounds due to the so-called antenna effect [2].

Organic compounds of lanthanides have been widely reported to have desirable properties that show intense narrow-band emissions through efficient intermolecular energy transfer from the binding states to the central metal ions. The luminescence spectra are greatly intensified by the completion of lanthanide ions with organic ligands. These lanthanide ions form a stable crystalline complex with heterocyclic ligands, such as bipyridyl (bipy) and phenanthroline (phen10) [3] [4] [5]. On the other hand some projects have been reported to obtain low-cost luminescent material, using a silica matrix. Orozco [6], reported the synthesis and characterization of the luminescent and structural properties of zinc sulphide (ZnS) powders, doped with the trivalent Europium ion (Eu^{3+}) and encapsulated in silicon dioxide (SiO_2), the luminescent properties of the powders were significantly improved when they were thermally treated, being the experiment nine with the ratio [1:2, 1:2, 8%, 800°C] the one with the best result. Reyes *et al.* [7] obtained nanometric powders of $\text{BaTiO}_3\text{:Eu}^{3+}$ composite materials encapsulated in a SiO_2 matrix they found that is possible to increase the intensity of light emission in the powders, when they are thermally treated at 200°C for 24 h, according to the photoluminescence tests. Finally Y. Li *et al.* [8], synthesized $\text{TNaGdF}_4\text{:Yb, Er}$ nanoparticles were encapsulated in SiO_2 from the precursor TEOS by the sol-gel method. The light emission intensity of the SiO_2 encapsulation was 5 times higher compared to the $\text{NaGdF}_4\text{:Yb}$ nanoparticles. Therefore, it is demonstrated that the coating of ceramic and organic particles in SiO_2 improves the optical properties of the composite materials by increasing the light emission [9] [10] [11]. In this work, we improve this methodology in TTA:Tb:Dy:Eu organic powders.

2. Methodology

For the samples of the glass-ceramic materials, powders of TTA:Tb:Dy:Eu were used as precursors; which were previously obtained. To obtain the vitreous phase, Tetraethylorthosilicate (TEOS, $\text{C}_8\text{H}_{20}\text{O}_4\text{Si}$, 98%, Aldrich), and ethyl alcohol ($\text{CH}_3\text{CH}_2\text{OH}$, 99.5%, Meyer), distilled water and 0.05 ml of hydrochloric acid (HCl, Meyer) were used. The vitreous solution was obtained by the hydrolysis and condensation of Tetraethylorthosilicate (TEOS) in ethanol and water as solvents, and adding acid as catalyst. The hydrolysis reaction occurs when Tetraethylorthosilicate (TEOS) and water are mixed in the ethyl alcohol, as it's shown in the following reaction:



SiOH groups are generated as an intermediate producer of the reaction. Complete hydrolysis of the alkoxide can result in $\text{Si}(\text{OH})_4$ silicic acid, but does not occur.

Synthesis of the composite TTA:Tb:Dy:Eu@SiO₂

Five samples were made with different weight relations of TTA:Tb:Dy:Eu with respect to SiO₂, as shown in **Table 1**.

To do this, 0.05 g of TTA:Tb:Dy:Eu were weighed and placed in an agate mortar, the volume of 2 ml SiO₂ was added until 8.90 ml SiO₂ was reached and it was manually mixed with the pistil until a paste was formed and finally a composite powder of TTA:Tb:Dy:Eu@SiO₂ was obtained. Once the TTA:Tb:Dy:Eu@SiO₂ composite is obtained, three samples with molar ratios of 5:95, 10:90 and 20:80 were placed in an oven at 88 °C for 24 hours.

3. Analysis of Results

3.1. Fourier Transform Infrared Spectroscopy (FT-IR)

The IR spectrograms for the pure compound TTA:Tb:Dy:Eu, as well as the compound TTA: Tb:Dy:Eu@SiO₂ were analyzed and the IR developed of the compound TTA:Tb:Dy:Eu@SiO₂ with a molar ratio of 5:95, 10:90 and 20:80 were studied.

Figure 1 shows the spectrum of the control sample, which is TTA:Tb:Dy:Eu. A band in the range 3600 - 2900 cm⁻¹ can be observed, which is attributed to the stretching vibrations hydroxyl group (-OH) because there may be moisture in the sample. The bands TTA:Tb:Dy:Eu, likewise the bands of C-CF₃ that indicate that the chelation of europium that predominates before the terbium and dysprosium was presented .

Figure 2 shows the IR spectrograms of the TTA:Tb:Dy:Eu@SiO₂ composite at different molar ratios, where the TTA:Tb:Dy:Eu nanoparticles are encapsulated silica matrix. The presence of the bands in 3500 and 1650 cm⁻¹ is observed, indicating that the silica contains a high amount of molecular water and hydroxyl. The band between 1610 - 1650 cm⁻¹, is assigned to the vibrations of molecular water deformation δOH and results from the angular deformation O-H in H₂O (H-O-H), so the band 1631 cm⁻¹ can be attributed to residual ethanol. The bands at 1070 cm⁻¹ are due to the asymmetric vas (Si-O-Si) stretching movement, in which the O atom moves back and forth along a line parallel to the axis joining the two Si atoms. The bands at 870 cm⁻¹ correspond to the symmetrical stretch

Table 1. Experimental design of TTA:Tb:Dy:Eu@SiO₂.

Exp.	%TTA:Tb:Dy:Eu	% SiO ₂	WTTA:Tb:Dy:Eu [g]	WSiO ₂ [g]	Vol. of solution SiO ₂ [ml]
1	5	95	0.05	0.95	8.9039
2	10	90	0.1	0.90	8.4353
3	15	85	0.15	0.85	7.9667
4	20	80	0.20	0.80	7.4980
5	25	75	0.25	0.75	7.0294

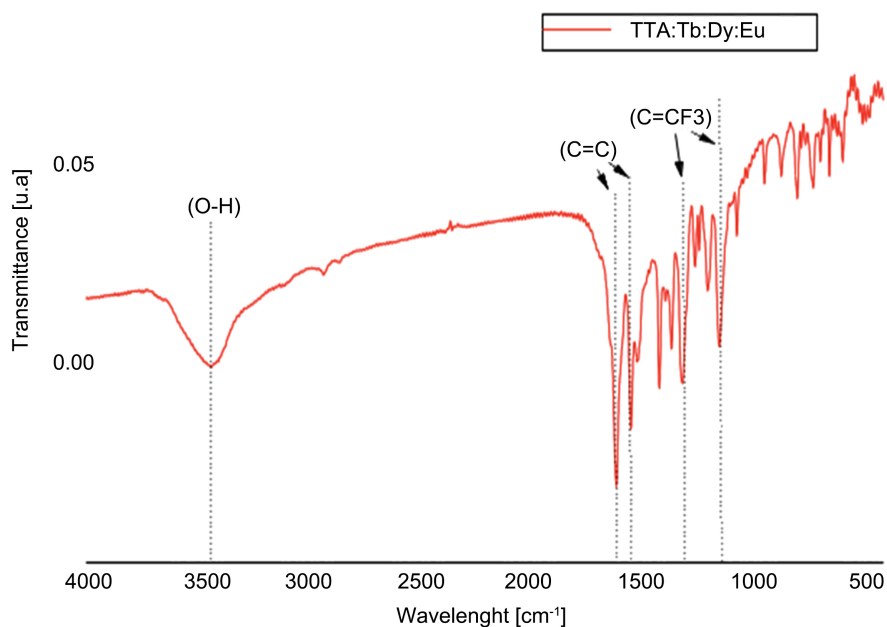


Figure 1. FT-IR Spectrum of TTA:Tb:Dy:Eu powder.

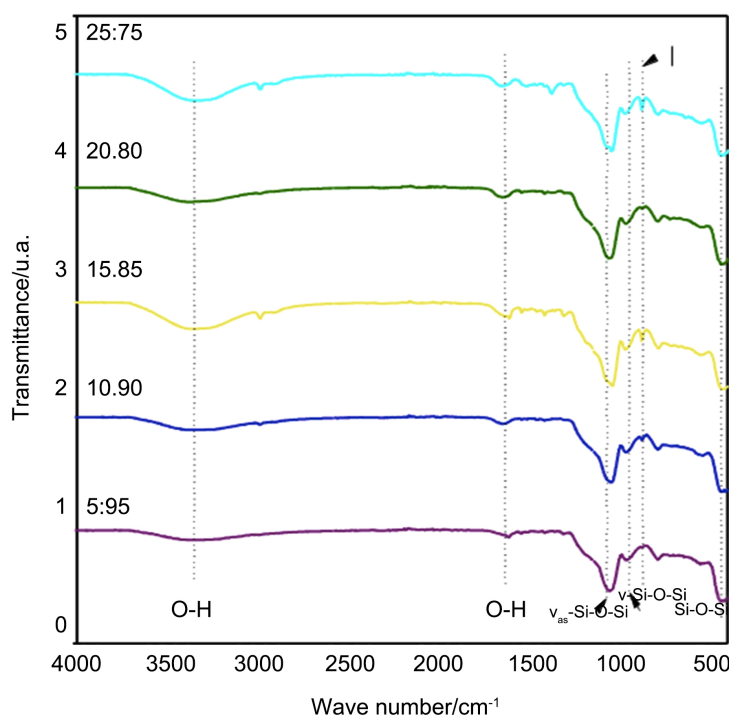


Figure 2. FTIR spectra of TTA:Tb:Dy:Eu@SiO₂ with different molar percent of SiO₂ without thermal treatment.

of the O atom along the line bisecting the axis formed by the two Si atoms in siloxane bonds vs (Si-O-Si). The bands at 438 cm⁻¹ are associated with rocking vibrations of the O atom around an axis that joins the two Si atoms in siloxane bonds (Si-O-Si). The band centered around 950 cm⁻¹ is assigned to the stretching vibration of ν-bonds (Si-O-Si). The spectra show that the amorphous silica

matrix effectively encapsulates the luminescent material TTA:Tb:Dy:Eu since the characteristic bands of TTA:Tb:Dy:Eu are not present. As can be seen in **Figure 2**, the structural bands between $1500 - 1550 \text{ cm}^{-1}$ that are due to the stretching vibrations of the C=C of the TTA disappear, only the characteristic spectrum of SiO_2 is observed since the samples have a higher concentration of silica.

The spectra of FTIR TTA:Tb:Dy:Eu@ SiO_2 of the samples 5:95, 10:90 and 20:80 with heat treatment 88°C for 24 hours are also shown, as it can be seen in **Figure 3**. Where the spectra with molar ratios 5:95, 10:90 and 20:80 with heat treatment can be seen that the sample with molar ratio 20:80 has the characteristic spectrum of silica, while the sample 5:95 and 10:90 have the C- CF_3 bonds since these bands are very defined. There are bands associated with the asymmetric vibration of Si-O-Si in the range of 1049 and 440 cm^{-1} these bands correspond to the silica matrix.

3.2. Photoluminescence

Excited emission spectra at 380 nm were performed, as well as the excitation spectra. First, the excitation spectrum is made to determine which is the best wavelength to excite the samples and obtain the highest emission intensity. The results that were obtained for the highest emission were in a length $\lambda = 380 \text{ nm}$, as can be seen in **Figure 4**. The compounds with respective molar ratios 5:95, 10:90, 15:82, 20:80 and 25:75 of TTA:Tb:Dy:Eu@ SiO_2 are observed in this figure is can be observed that the peak proportion of 5:95 is the highest.

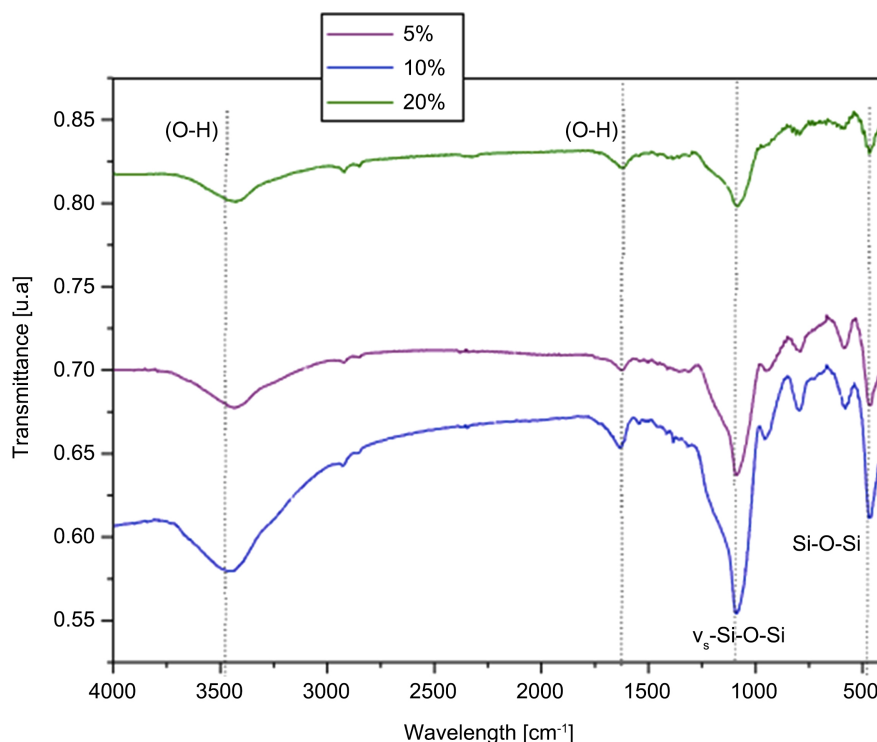


Figure 3. FT-IR spectra of samples 5TTA:Tb:Dy:Eu@95 SiO_2 , 10TTA:Tb:Dy:Eu@90 SiO_2 y 20TTA:Tb:Dy:Eu@80 SiO_2 with thermal treatment at 88°C with 24 hours.

In the case of the compound with molar ratio 20:80 purple color, two peaks are observed in $\lambda = 306$ nm and $\lambda = 380$ nm, despite the fact that in that sample the highest is $\lambda = 306$, the other samples do not present it therefore the peak of more intensity in the sample number five with the different molar ratios is the one found in 380 nm. The emission bands, on the other hand as is shown in **Figure 5** are characteristic of the Eu^{3+} transitions at 614.29, 594.87, 654.65 and 705.44 nm, with the highest emission correspond also to 5% of SiO_2 .

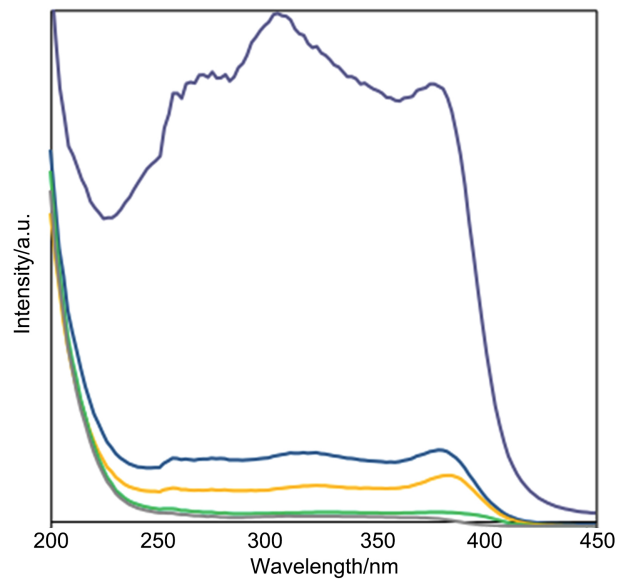


Figure 4. Excitation Spectrum of TTA:Tb:Dy:Eu@SiO₂ with the different percent of SiO₂ 5:95 (blue), 10:90 (yellow), 15:85 (green), 20:80 (purple), 25:75 (gray).

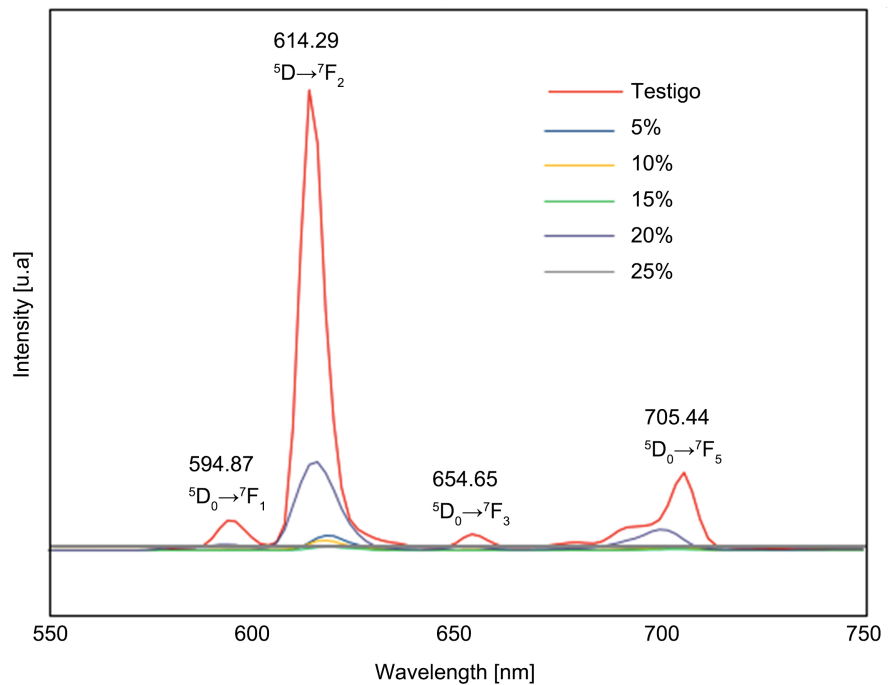


Figure 5. Spectra of emission TTA:Tb:Dy:Eu@SiO₂, for the different molar ratios.

3.3. Chromaticity Diagram

The chromaticity diagrams with their respective emission spectrum are shown below. The CIE chromatic coordinate of the compound TTA:Eu:Dy:Tb was a purity red-orange, this color is characteristic of the Eu^{3+} ion since it is the ion in the pure compound (TTA:Eu:Dy:Tb). The CIE chromaticity coordinate for composite TTA:Eu:Dy:Tb@ SiO_2 molar ratio 5:95, 10:90 and 20:80 in which you can note that for these two samples with 5% TTA:Eu:Dy:Tb, 10% TTA:Eu:Dy:Tb and 20% TTA:Eu:Dy:Tb are those that maintain a red-orange purity equal to the control sample. The sample containing 20% TTA:Eu:Dy:Tb is closer. This can be verified in the following diagrams. The CIE chromaticity coordinates for the control sample (TTA:Eu:Dy:Tb) are shown, as well as the silica doped samples with the different molar ratios of TTA:Eu:Dy:Tb@ SiO_2 . The displacement that the silica encapsulated samples have with respect to the pure sample is observed, where the molar ratio that is maintained in the purity of the red-orange region is the compound 20% TTA:Eu:Dy:Tb@80% SiO_2 , with these results it can be clearly seen that the samples that are in the red-orange region are the control (TTA:Eu:Dy:Tb), the compound that has 5%, 10% and 20% molar ratio TTA:Eu:Dy:Tb, while the samples with 15:85 and 25:75 molar ratio are displaced from the orange region. The values calculated CIE coordinates are shown in **Table 2**.

Figure 6 shows the comparison of the emission at the highest peaks of the TTA:Eu:Dy:Tb@ SiO_2 composite under an excitation of 380 nm. It can be seen that the intensity of the silica encapsulated material increase when the composite sample has a molar ratio of 20:80, and the sample with a molar ratio of 25:75. Therefore, this diagram showed the compound with the highest luminous intensity TTA:Eu:Dy:Tb.

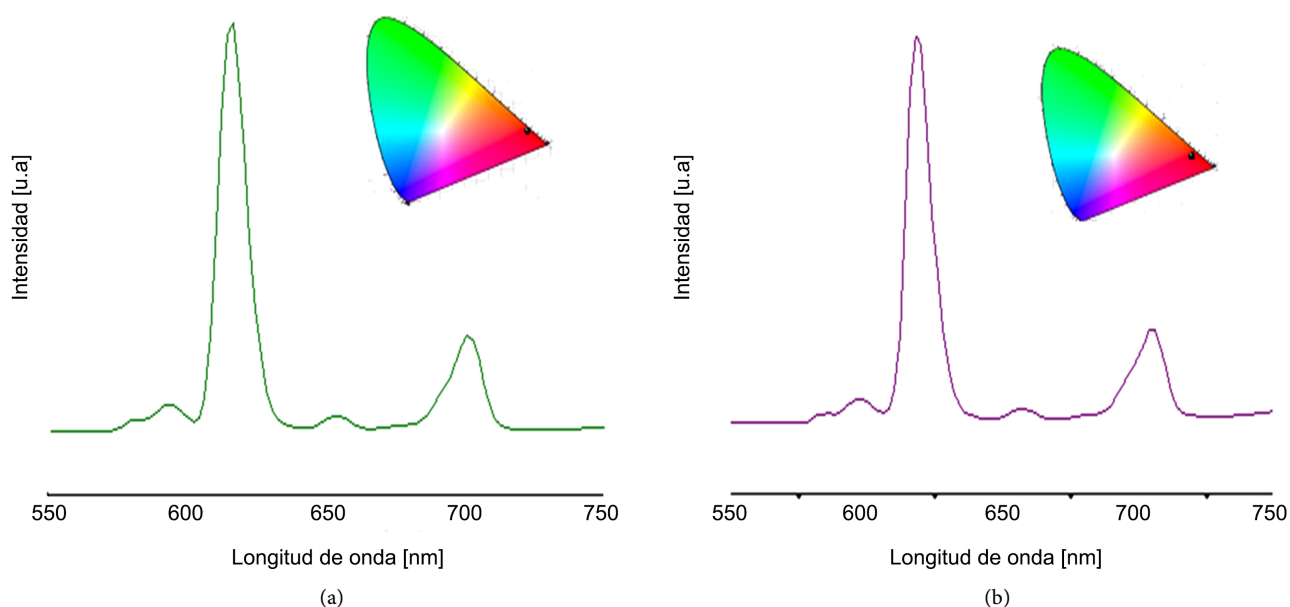


Figure 6. Spectra of emission TTA:Tb:Dy:Eu@ SiO_2 , and CIE coordinator for (a) 20:80 and (b) 25:75.

3.4. X-Ray Diffraction Spectroscopy (XRD)

The diffractograms presented in **Figure 7** show the amorphous behavior of the material TTA:Eu:Dy:Tb@SiO₂, as well as of TTA:Eu:Dy:Tb, with a crystal size in the order of nanometers. Likewise, a wide band can be observed that presents the composite Dy:Tb@SiO₂ in its different molar relations, where the shoulder grows with intensity in the samples that received thermal treatment at 88°C, since when the temperature increases, the change of position of the peaks increases due to a possible change of phase. The uniform frames on the surface suggest that a homogeneous molecular base material was obtained by the strong covalent bond in the Si-O-Si networks that belong to a complex and large molecular system [9].

Table 2. CIE X, Y of the samples with different percentage of SiO₂.

Muestra	x	y
TTA:Eu:Dy:Tb	0.66775392	0.32365108
TTA:Eu:Dy:Tb@SiO ₂ 5:95	0.63583116	0.32048406
TTA:Eu:Dy:Tb@SiO ₂ 10:90	0.63871604	0.326348
TTA:Eu:Dy:Tb@SiO ₂ 15:85	0.56552774	0.33406423
TTA:Eu:Dy:Tb@SiO ₂ 20:80	0.65539378	0.32666705
TTA:Eu:Dy:Tb@SiO ₂ 25:75	0.51643915	0.34470723

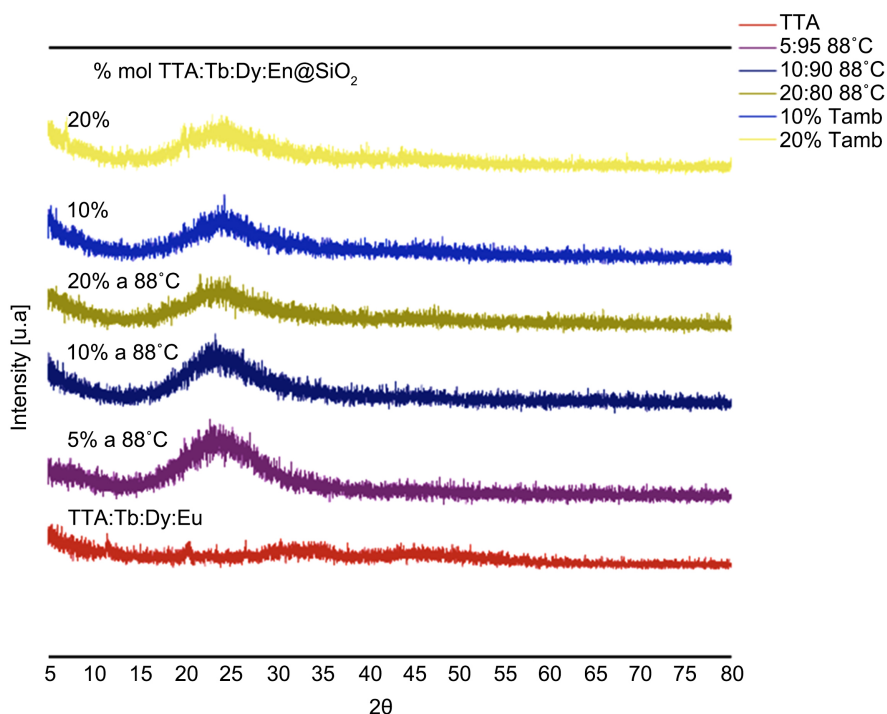


Figure 7. DRX spectra for different samples of TTA:Tb:Dy:Eu and TTA:Tb:Dy:Eu@SiO₂, with different percentage and heat treatment.

3.5 Thermogravimetric Analysis (TGA)

The thermogravimetric analysis of TTA:Eu:Dy:Tb shown in **Figure 8** can be divided into four stages as the temperature increases, from 30°C to 100°C, a peak decomposition center at 56°C and a weight loss of approximately 5% by weight is observed, which can be attributed to the release of surface moisture from TTA:Eu:Dy:Tb as TTA, you have another loss of 100°C to 231°C with 20% by weight. From 231°C to 400°C the weight loss was approximately 75% and a strong thermal absorption decomposition peak (center at 346°C). From 350°C, the weight loss was approximately 90% weight and a wide TTA:Eu:Dy:Tb was damaged.

Figure 9 shows the TGA curves for the composite TTA:Eu:Dy:Tb@SiO₂ with a molar ratio 5:95 at 1) room temperature and 2) with heat treatment, differences in decomposition temperatures and mass losses. The compound with molar ratio of 5:95 shows that the weight loss occurred at lower temperatures. Once the colloidal solution of TTA:Eu:Dy:Tb and the SiO₂ sun were mixed, you have a loss of 2% at a temperature of 31°C in this first stage is due to the losses of OH due to moisture and ethyl alcohol, you have a peak of decomposition at 310°C common loss of 83% weight. Meanwhile, the TGA curves for the composite TTA:Eu:Dy:Tb@SiO₂ with a molar ratio 20:80 at 3) room temperature and 4) with heat treatment. The TTA:Eu:Dy:Tb@SiO₂ composites at room temperature shows a higher weight loss at 367°C, while samples thermally treated at 88°C are the most weight loss are shown at 451°C, therefore, it can be observed that when the sample heat treatment, you have a lower percentage of weight loss than with samples at room temperature, as well as the decomposition temperature increases for heat-treated samples.

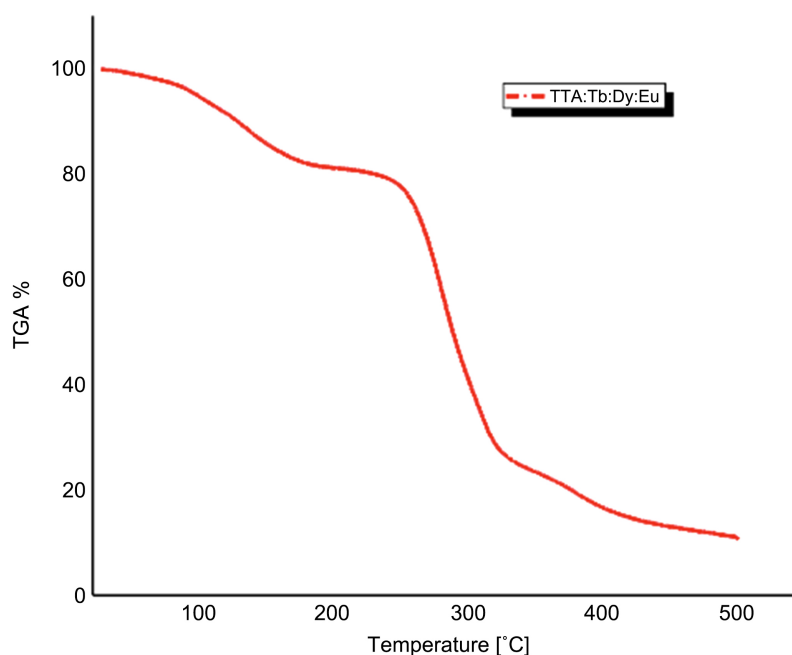


Figure 8. Thermogravimetric analysis of TTA:Eu:Dy:Tb sample.

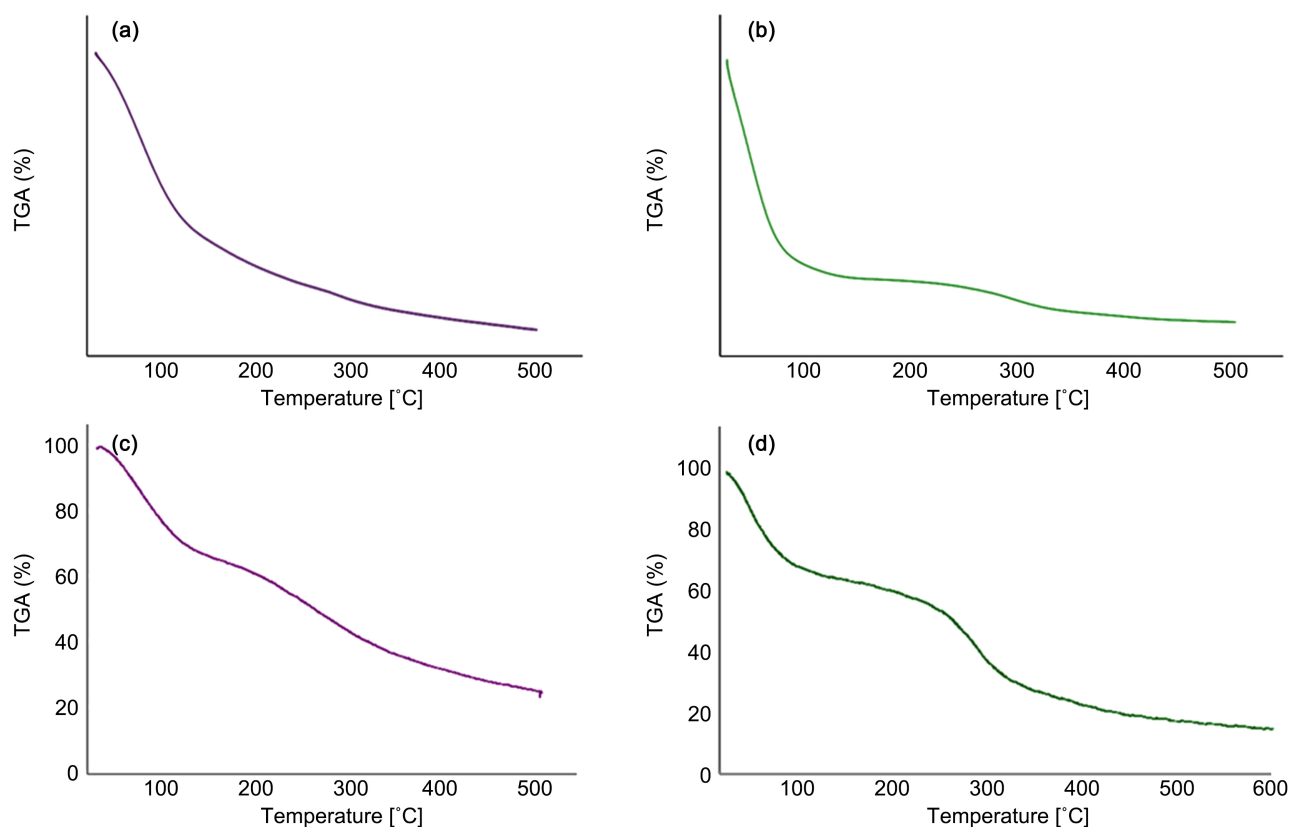


Figure 9. TGA studies for different percentage and heat treatment 5:95 at (a) room temperature (b) with heat treatment, and 20:80 (c) room temperature (d) with heat treatment.

4. Conclusions

The TTA:Tb:Dy:Eu@SiO₂ composite was synthesized by the sol-gel method, which presents a good luminescence. The amount of luminescent material at room temperature was increased from 0.7517 g of TTA:Tb:Dy:Eu to 9.8603 g of TTA:Tb:Dy:Eu@SiO₂ by encapsulating it in a silica matrix without losing the red luminescence. By means of XRD, FL, FT-IR, TGA characterization techniques, structural and luminescent properties were determined.

The structure of the TTA:Tb:Dy:Eu@SiO₂ composite was found to be amorphous at room temperature, as well as heat-treated to 88°C. In the samples a wide band (shoulder) that has the highest intensity which is at 22 on the scale 2θ are observed.

By analyzing FT-IR in the silica-free TTA:Tb:Dy:Eu material, the presence of characteristic bands of TTA-related C=CF₃ links were confirmed. In the TTA:Tb:Dy:Eu@SiO₂ composite the bands corresponding TTA disappear, keeping only the bands corresponding to Si-O-Si vibrations, corroborating the encapsulation of TTA: Tb:Dy:Eu in the SiO₂ matrix. In both compounds, there are OH- group due to the incorporation of the ethanol solvent.

The photoluminescence analysis shows that the composite TTA:Tb:Dy:Eu@SiO₂ with a molar ratio of 20% TTA:Tb:Dy:Eu and 80% SiO₂ obtains the highest luminescence than the rest of the composites due to the concentration of silica.

The highest intensity was found at $\lambda = 614$ nm where the characteristic transitions $^5D_0 \rightarrow ^7F_2$ of Eu^{3+} appeared, confirming the color it emits is red-orange. The TGA analysis confirms a red-orange emission that tends to decrease with the silica concentration because the $\text{TTA:Tb:Dy:Eu@SiO}_2$ composite decomposes at a lower temperature than when having the TTA:Tb:Dy:Eu material without silica.

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Conflicts of Interest

The authors declare no conflicts of interest regarding the publication of this paper.

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