

Synthesis, Growth and Characterization of an **Organic Material: P-Chloroanilinium Ethanoate**

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How to cite this paper: Sakunthaladevi, R. and Jothi, L. (2020) Synthesis, Growth and Characterization of an Organic Material: P-Chloroanilinium Ethanoate. Journal of Minerals and Materials Characterization and Engineering, 8, 133-148. https://doi.org/10.4236/jmmce.2020.83009

Received: April 29, 2020 Accepted: May 24, 2020 Published: May 27, 2020

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Abstract

Aromatic organic material of P-Chloroanilinium ethanoate (CAE) has been synthesized by condensation process using methanol as a solvent. The synthesized product is allowed to evaporate slowly at room temperature and crystals of cell dimension $25 \times 29 \times 10 \text{ mm}^3$ were grown. The grown material was subjected to single crystal X-ray diffraction to obtain the cell parameters. The presence of functional groups in the grown material was confirmed by Fourier transform infrared and Fourier transform Raman spectral studies. The UV-VIS absorption spectrum reveals that the material has lower UV cut-off wavelength at 337 nm. Using the Kurtz Perry Powder method, the NLO activity was confirmed. Mechanical strength of the crystal was estimated by Vickers micro hardness test which shows that the material belongs to the soft category. The thermal behavior of the material was investigated by using thermo gravimetric and differential scanning calorimetric analyses.

Keywords

X-Ray Diffraction, Linear and Non-Linear Optical Studies, Mechanical Parameters

1. Introduction

In recent years, organic electron donor-acceptor or charge transfer complexes fascinated much attention owing to their high potential applications in the field of optical signal processing, color displays, frequency conversion, photonics, laser remote sensing, organic superconductors, molecular electronics, electro-optical amplitude modulation, high density optical data storage, ultra-compact lasers, optical switching, optical parametric generation and biophotonics [1] [2] [3]. Aniline and its related structures are widely studied because of its importance in many pharmaceutical, electro-optical and several other social applications like production of dves and pesticides [4]. P-chloroaniline is an organochloric aniline compound with molecular formula C₆H₄NH₂Cl. Already synthesis, growth and characterization studies of organic nonlinear optical material 4-chloroaniline single crystal were reported [5]. Molecular ionic simple complex crystals of an organic molecule should have large second order hyperpolarizability to exhibit good nonlinear optical properties and show unique to the crystal structure. The distinct features of molecular ionic crystal give empathizing correlation between the crystal packing and physical properties [6]. Ethanoic acid is one of the well-known carboxylic acids promoting molecular self-assembly by means of strong hydrogen bonding through its carboxylic acid group and the ring substituted amino group of p-chloroaniline. Strong hydrogen bonds occur due to the polarizable hydrogen atom covalently bonded to an electron with drawing donor oxygen atom and interact with partially negatively charged and comparatively less polarizable acceptor nitrogen atom, which will also increase the molecular hyperpolarizability [7] [8]. Even though many works using aldehydes, substituted aldehydes with p-chloroaniline had been already reported [9]-[15], best of our knowledge no work had been done with p-chloroaniline with ethanoic acid. However, in this paper, we report first time the synthesis, growth and characterization of p-chloroanilinium ethanoate (CAE) ionic crystals and investigated the linear and nonlinear, optical, mechanical, and thermal properties of the grown CAE crystal.

2. Experimental Procedure

2.1. Material Synthesis

The solubility studies in different solvents showed that the titled compound was insoluble in water and highly soluble in ethanol and methanol. P-Chloroanilinium ethanoate (CAE) were synthesized by condensation process using methanol as a solvent because the growth rate of the crystal was well it. High purity chemicals P-Chloroaniline and Ethanonic acid were purchased from Loba chemicals that were used for this synthesis. The equimolar ratio of 4-Chloroaniline and Ethanonic acid was refluxed by methanol solvent and condensed about 7 hours with the temperature of 80°C, to get the saturated solution of CAE. The solution was filtered in a beaker to remove any impurities present in it, using a Whatman filter paper.

2.2. Material Synthesis

Consecutively, the beaker was covered with perforated polythene paper and placed in an undisturbed position to ensure the slow solvent evaporation at room temperature. Seeds were obtained within 20 days due to spontaneous nucleation in supersaturated solution of CAE. The size of the crystals depends on the amount of material available in the solution which in turn is decided by the solubility of the material in solvent.

An electron is transferred from the carbonyl group of ethanoate to the amine group of aromatic amine forms the anilinium ethanoate ionic crystal. The reaction mechanism of the grown CAE was shown in **Figure 1**.

Transparent and good quality seeds were selected for growth. After 25 days optically good colorless single crystal of dimension $25 \times 29 \times 10 \text{ mm}^3$ was grown as shown in **Figure 2**. The grown CAE crystal has been subjected to various characterization techniques.

3. Result and Discussion

3.1. X-Ray Diffraction Studies

Suitable crystals of P-Chloroanilinium ethanoate were selected for single crystal X-ray diffraction analysis. The selected crystal was employed in Bruker D8 VENTURE SC-XRD with Mo and Cu radiation. Bruker allows the combination of two X-ray sources (Mo, Cu) with a highly accurate goniometer. The PHOTON



Figure 1. Reaction scheme of a grown CAE.



Figure 2. Photograph of a grown CAE single crystal.

100 CMOS detector is optimized for both Mo (K α) and Cu (K α) radiations. Accurate unit cell parameters were refined by the selection of 150 reflections out of 161 reflections. Data collection was performed by using the APEX3 software program. This CAE crystal belongs to orthorhombic symmetry with space group P₂₁. The cell dimensions are a = 7.42 Å, b = 8.68 Å, c = 9.32 Å, $\alpha = \beta = \gamma = 90^{\circ}$, V = 600 Å³, which exhibit interesting ferroelectric, piezoelectric and triboluminescence properties. The possible correlation between these properties of organic single crystals belonging to the orthorhombic system has already been investigated and was found to be structure sensitive [16]. Table 1 shows the crystallographic data of CAE.

3.2. FT-Infrared and FT-Raman Spectral Studies

A Perkin Elmer-Paragon Fourier Transform Infrared Spectrometer was used to record the FTIR spectrum of CAE. The spectrum was recorded by the KBr pellet technique between the range 4000 cm⁻¹ to 400 cm⁻¹ as shown in **Figure 3**. Fourier

Parameters	Values	
<i>a</i> (Å)	7.42 Å	
<i>b</i> (Å)	8.68 Å	
<i>c</i> (Å)	9.32 Å	
α(°)	90°	
β (°)	90°	
γ(°)	90°	
Volume	600 Å ³	
Crystal system	Orthorhombic	
Space group	P ₂₁	

Table 1. The crystallographic data of CAE.





Transform Raman spectrum of CAE was recorded in the region of 4000 - 500 cm⁻¹ by using Bruker RFS 27 FT-Raman spectrometer equipped with an FRA-106. FT-Raman accessory in the Stokes region using the 1064 nm line of an Nd: YAG laser for excitation operating at 200 mW power with a resolution up to 2 cm⁻¹. The FT-Raman spectrum of CAE is shown in Figure 4. The existence of one or more aromatic rings in a structure was normally readily determined from the C-H and C=C-C ring related vibrations. The C-H stretching occurs at 3472 cm⁻¹ and was typically exhibited as a multiplicity of weak to moderate bands compared with the aliphatic C-H stretch. In the infrared spectrum, the peak at 3472 cm⁻¹ and 3382 cm⁻¹ corresponds to C-H stretching vibration but in Raman spectra the peak at 3382 cm⁻¹ and 3059 cm⁻¹. The other most important set of bands are the aromatic ring vibrations cantered around 1600 cm⁻¹ and 1500 cm⁻¹, which usually appear as a pair of bands and in this work, the peaks appear at 1615 cm^{-1} and 1503 cm^{-1} for infrared and in Raman 1613 cm^{-1} and 1566 cm^{-1} confirms the ring formation. In imine group 1,4 substitution (para) of C-H deformation vibration obtained in the region 800 cm⁻¹ - 840 cm⁻¹ in this present work the peak obtained at 824 cm⁻¹ (infrared) and 817 cm⁻¹ (Raman). The peak at 636 cm⁻¹ was assigned to confirm mono chlorinated aromatic C-Cl stretching in infrared and 635 cm⁻¹ in Raman spectral studies. A sharp peak at 1087 cm⁻¹ in infrared and 1089 cm⁻¹ in Raman bands gave the vibrational frequency of C-N aromatic stretching. The measured infrared and Raman peak positions and their assignments of CAE are given in Table 2.

3.3. UV-VIS Transmission Studies

Internal transmittance of CAE material was recorded using Perkin Elmer Lambda 35 UV-VIS spectrometer in the wavelength range of 200 - 1100 nm. Optically polished crystal of 2 mm thickness was used for the study and the recorded spectrum was shown in **Figure 5**. The UV-VIS spectrum of CAE in the range of





$FTIR (cm^{-1})$	<i>FT-Raman</i> (cm ⁻¹)	Spectral Assignments
3382 2982 1615 1503 1494 1285 1179 1087 824 636	3382 2980 1613 1506 1491 1287 1181 1089 817 635	C-H stretching vibration C=C stretching C=O stretching mode COO [−] symmetric stretching COO [−] anti symmetric stretching CH ₃ group deformation NH ₃ rocking C-N aromatic stretching C-Cl stretching

Table 2. The vibrational characteristics peaks of CAE.



Figure 5. Transmission spectrum of CAE.

200 - 238 nm consists of two sharp falls. The first fall at 237 nm interpreted as benzene $\pi \rightarrow \pi^*$ transition modified by the amine and nitro substituents, whereas the second fall observed at 289 nm arises from the electronic transition involving aniline and methyl groups [17]. At 238 nm, the percentage of transmittance starts increasing, which shoots 99% at 337 nm. Thus, we infer that CAE crystal was transparent in the region between 330 nm and 1100 nm and it is the key requirement for materials using for optoelectronic applications [18]. The optical absorption coefficient (*a*) was calculated from transmission spectrum using the formula,

$$\alpha = 2.3026 \frac{1}{d} \log \left(\frac{1}{T} \right)$$

where T was the transmittance and d was the thickness of the sample. From optical band gap energy was determined from,

$$\alpha = \frac{A(hv - E_g)^{\frac{1}{2}}}{hv}$$

where E_g was the optical bandgap of the sample, *h* was the Plank's constant, *v* was the frequency of the incident photon and *A* was a constant. The optical band gap energy of the grown crystal was calculated by,

$$E_g = \frac{1240}{\lambda} (eV)$$

where λ is the lower cut off wavelength (337 nm). The band gap of the grown CAE crystal was found to be 3.67 eV. Also, the optical energy band gap of CAE was evaluated by a linear plot between hv and $(ahv)^{1/2}$ and the graph was shown in **Figure 6** and is also found to be 3.64 eV, it was the same as the calculated value.

3.4. Fluorescence Spectral Analysis

Fluorescence spectral analysis was one of the effective methods that provide information about the fluorescence behaviors of the grown optical materials. The fluorescence emission spectrum of CAE was recorded by using Perkin Elmer LS 45. Compounds containing aromatic functional groups with low energy $\pi \rightarrow \pi^*$ transition levels and highly conjugated double bond structures exhibit fluorescence. This property finds wide applications in the branches of biochemical, medical and chemical research fields for analyzing organic compounds [19]. The existence of defects shows the major impact on fluorescence materials. The grown crystals were excited at 344 and 694 nm to discern the photoluminescence spectrum. Figure 7 shows the emission spectra recorded at room temperature in the range of 310 nm to 900 nm. The strong emission peak for CAE was observed at 344 nm and a weak emission spectrum observed at 695 nm. Using the spectrum the bandgap energy of CAE was calculated as 3.61 eV. This result has coincided with the UV-VIS transmission spectrum. The emission peak energy values were found to be lower than the band gap energy of the material. Therefore, the obtained fluorescence emission might be radiative recombination of trapped electron and holes.







Figure 7. Fluorescence spectrum of CAE.

3.5. Second Harmonic Generation Efficiency

Generally, the high SHG efficiency was related to favorable molecular alignment facilitating nonlinearity. In Kurtz Perry method the crystal was ground to homogeneous powder and tightly packed in a micro capillary tube and mounted in the path of the Q-switched Nd:YAG laser beam emitting 1064 nm, 9 ns pulse width with a repetition rate of 10 Hz and 7 mJ powers. The second harmonic generation efficiency of the grown CAE crystal was about ~1.05 times that of KDP.

3.6. Micro Hardness Studies

The important mechanical parameters like Vickers hardness number, elastic stiffness constant, Fracture toughness, Brittle index and yield strength of each load were determined for the grown CAE crystals using the applied load P and diagonal length d of the tester. Indentations were made on the flat polished (001) face of the crystal at room temperature for loads 25, 50 and 100 g using Vickers hardness tester fitted with diamond intender attached to an incident light microscope to study the mechanical strength of the crystal. Applied load P was varied from 25 to 100 g with constant indention time 10 s. Beyond 100 g of the applied load cracks were observed and material chipping became significant and thus hardness test could not be carried out further. The Vickers hardness number (H_v) of each load was calculated by,

$$H_v = \frac{1.8544P}{d^2} \left(\text{kg/mm}^2 \right)$$

where 1.8544 was the constant value for diamond pyramid indenter, the applied load is P and average of the two diagonal lengths of the indentations is d. The graph between the variations of H_v with load P was plotted in Figure 8 and a plot of log P versus log d is shown in Figure 9. It was evident from Figure 8 that H_v increases with increasing with load P due to the reverse size indentation effect [20]. The Meyers index number "n" estimated from the slope of Figure 9 which indicates that CAE crystal belongs to the soft materials category [21]. The strength of the bending atom in connection with a neighbouring atom can be analyzed by Wooster's empirical formula of elastic stiffness constant and it was derived from the Vickers hardness number (H_v). The elastic stiffness constant (C_{11}) of each load was calculated by using the formula,

$$C_{11} = \left(H_{\nu}\right)^{\frac{7}{4}} \left(\text{GPa}\right)$$

The plot between load *P* and stiffness constant C_{11} for the CAE crystal was shown in **Figure 10**. The fracture toughness K_c is the resistance to fracture which



Figure 8. Vickers hardness plot of CAE.







Figure 10. Plot of P Vs C_{11} of CAE.

envisages the toughness of the material. It also determines the intensity of the fracture stress applied under uniform loading and was given by the relation,

$$K_c = \frac{P}{\beta_o \left(L\right)^{\frac{3}{2}}} \left(\text{kg} \cdot \text{m}^{-3/2} \right)$$

where β_o is the indenter constant equal to 7 for the Vickers diamond pyramid indenter and $L \ge d/2$ and the graph between *P* and K_c was shown in Figure 11. The relation between hardness number H_v and fracture toughness K_c was known as the brittleness index Bi and it was computed by,

$$Bi = \frac{H_v}{K_c} \left(\mathrm{m}^{1/2} \right)$$

and a plot of load *P* and brittleness index *Bi* was shown in **Figure 12**. From the hardness values, the Yield strength (σ_y) can be calculated. The yield strength was defined as the stress at which the material begins to deform plastically and it depends on Meyer's index number n. For n > 2, σ_y can be calculated using the expression,

$$\sigma_{y} = \frac{3-n}{2.9} \left(\frac{12.5(n-2)}{3-n} \right)^{n-2} H_{y} (Pa)$$

For n < 2, the yield strength is calculated using the relation,

$$\sigma_{y} = \frac{H_{v}}{3} (Pa)$$

The yield strength of each load was shown in **Figure 13**. The mechanical parameter values of each load of CAE were tabulated in **Table 3**. From this, the hardness number, elastic stiffness constant, fracture and the yield strength are increased due to the increased load but the Brittle index number was decreased.

3.7. Thermal Analysis

Thermo gravimetric analysis was an efficient tool to describe the decomposition







Figure 12. Plot of *P*Vs *Bi* of CAE.



Figure 13. Plot of *P*Vs σ_y of CAE.

and phase transition in the material with temperature. The TG/DTG and TG-DSC curve was recorded by a heating rate of 10° C min⁻¹ by using NETZSCH

STA 449 F3 Jupiter thermal analyzer for the temperature range from 30°C to 270°C for 6.085 mg of a sample. The TG/DTG curve of the CAE was shown in **Figure 14**. The TG curve shows that the material was stable up to 73°C and the decomposition step was observed from 73°C to 189°C with a drastic weight loss of about 90%. This weight loss could be attributed to the compound oxidation with the evaporation of H_2O , CO_2 gases [22]. Finally, the residue was observed at about 1.5% at 270°C. The sharpness of the DTG endothermic peak indicates good crystallinity of CAE.

The differential scanning calorimetry (DSC) measurement was carried out for CAE material by using a sample of 10 g with a thermal analyzer of Netzsch DSC 204. The DSC curve of the material was shown in **Figure 15**. The endothermic peaks at 73°C and 189°C are due to the decomposition process which coincided exactly with the TG/DTG curve. The shape of the peaks changing in such a way of height, breadth, and sharpness is due to the poor thermal contact of the sample and their quantity (mass) [23] [24]. The sharpness of the DSC peaks shows that the CAE sample has higher crystallinity and purity, which is useful for laser and NLO applications for temperature below 73°C.

The TG-DSC trace of the sample is shown in Figure 16. From the TG

Machanical Departmentance	Load P(g)		
Mechanical Falameters	25	50	100
Hardness Number H_v (kg/mm ²)	29.1	46.7	67.2
Elastic Stiffness constant C_{11} (GPa)	3.65	8.41	15.77
Fracture toughness K_c (10 ⁵ kgm ^{-3/2})	1.42	2.41	3.76
Brittleness Index Bi (m ^{1/2})	20.45	19.46	17.87
Yield Strength σ_y (10 ¹⁰ pa)	17.76	28.62	41.01









Figure 15. DSC trace of CAE.



Figure 16. TG-DSC trace of CAE.

thermo-gram there was a weight loss between 73°C with a maximum at 189°C. The total weight loss corresponds to 90%. The weight loss clearly illustrates the melting and its subsequent evaporation. The DSC trace for heating produced two nearly equal exothermic peaks at 73°C and 189°C due to chemical reaction. The area of the peak gave the amount of heat released if it is 128.1 J/g for the first and 127 J/g for the second peak. Since the CAE crystals were grown by a slow evaporation method, they have taken care of omitting the occluded water, if any, during the growth by annealing the samples fairly at higher temperatures. In this report, CAE shows a phase transition at 73°C, when the temperature was slowly

increased from room temperature.

4. Conclusion

An organic ionic crystal of CAE was synthesized by the condensation process and grown by slow evaporation solution growth method at room temperature. The unit cell parameters were confirmed by single crystal XRD studies which indicates that it belongs to the orthorhombic crystal system with space group P_{21} . Functional groups of the final product were analyzed by a comparative spectrum of both FTIR, FT Raman analyses. The lower cut off wavelength and transmission from the entire visible range was confirmed by the UV-VIS transmission spectrum and the band gap was found to be 3.67 eV. The powder second harmonic generation efficiency was 1.05 times that of KDP. Fluorescence spectral analysis provides information about the luminescence behavior of the grown optical materials. Emission peak energy values were found to be lower than the band gap energy of the material. From Vickers's hardness study the hardness number increases with increasing with load which indicates that CAE crystal belongs to soft material category. The sharpness of the DSC/DTG peaks shows that the CAE sample has higher crystallinity and purity, which was useful for laser and NLO applications below 73°C for its temperature.

Acknowledgements

The authors acknowledge Indian Institute of Science, Bangalore for NLO studies and Sophisticated Analytical Instrument Facility, IIT, Chennai for recording Single crystal data, FT-Raman and Thermal Analysis. The authors are thankful to Archbishop Casimir Instrumentation Centre, St. Joseph's College, Tiruchirappalli for UV, FT-IR, Fluorescence and Micro hardness tests.

Conflicts of Interest

The authors declare no conflicts of interest regarding the publication of this paper.

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