

# Increases in Lorentz Factor with Dielectric Thickness

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# Abstract

For many years, a Lorentz factor of L = 1/3 has been used to describe the local electric field in thin amorphous dielectrics. However, the exact meaning of *thin* has been unclear. The local electric field  $E_{loc}$  modeling presented in this work indicates that L = 1/3 is indeed valid for very thin solid dielectrics ( $t_{diel} \le 20$  monolayers) but significant deviations from L = 1/3 start to occur for thicker dielectrics. For example,  $L \approx 2/3$  for dielectric thicknesses of  $t_{diel} = 50$  monolayers and increases to  $L \approx 1$  for dielectric thicknesses  $t_{diel} > 200$  monolayers. The increase in L with  $t_{diel}$  means that the local electric fields are significantly higher in thicker dielectrics and explains why the breakdown strength  $E_{bd}$  of solid polar dielectrics generally reduces with dielectric thickness  $t_{diel}$ . For example,  $E_{bd}$  for SiO<sub>2</sub> reduces from approximately  $E_{bd} \approx 25$  MV/cm at  $t_{diel} = 2$  nm to  $E_{bd} \approx 10$  MV/cm at  $t_{diel} = 50$  nm. However, while  $E_{bd}$  for SiO<sub>2</sub> reduces with  $t_{diel}$ , all SiO<sub>2</sub> thicknesses are found to breakdown at approximately the same local electric field ( $E_{loc}$ )<sub>bd</sub>  $\approx 40$  MV/cm. This corresponds to a coordination bond strength of 2.7 eV for the silicon-ion to transition from four-fold to three-fold coordination in the  $O - Si \equiv O_3$  tetrahedral structure.

# **Keywords**

Dielectrics, Dielectric Breakdown, Local Electric Field, Lorentz Factor, Time-Dependent Dielectric Breakdown, TDDB, Bond Breakage, Thermochemical E-Model

# **1. Introduction**

Early on it was recognized by Mossotti and Clausius that the *local electric field* played a very important role in the polarization/distortion of molecules in the dielectric [1] [2]. Somewhat later, H. A. Lorentz [3] showed that the local electric field could be expressed (in rationalized mks units) as:

$$E_{loc} = E_{diel} + L\left(\frac{P}{\varepsilon_0}\right),\tag{1}$$

where  $E_{diel} (=\Delta V_{diel}/t_{diel})$  is the average field in the dielectric,  $\Delta V_{diel}$  is the voltage drop in the dielectric, P is the dielectric polarization (dipole moment per unit volume),  $\varepsilon_0$  is the permittivity in free space, and L is now commonly referred to as the Lorentz Factor. The local electric field arises from the superposition of fields produced by the conduction charges (on the conductive plates of capacitor) plus the field contribution from all the dipoles in the dielectric material. The local electric field plays a critically important role not only in polarization/distortion of molecules in the dielectric but it also plays a critically important role in the dielectric breakdown strength  $E_{bd}$  and in time-dependent dielectric breakdown (TDDB) [4].

In linear, homogeneous and isotropic dielectrics, the polarization P can be written:

$$P = \chi \varepsilon_0 E_{diel} = (k - 1) \varepsilon_0 E_{diel} , \qquad (2)$$

where  $\chi$  is the electric susceptibility and k is the dielectric constant. Using Equation (1) and Equation (2), the Lorentz equation is produced for the local electric field:

$$E_{loc} = \left[ 1 + L(k-1) \right] E_{diel} \,. \tag{3}$$

To determine *L* for a dielectric, Lorentz imagined a spherical cavity of dipoles (around an arbitrary lattice site in the dielectric) and treated their contribution to the local electric field discretely (by summation). The radius *R* of the sphere was chosen to be much larger than the lattice constant. Lorentz showed that, for such a spherical collection of dipoles (in a crystal with cubic symmetry), the dipolar contribution to the local electric field was zero. To find the local field contribution for dipoles outside of the sphere, Lorentz placed a surface polarization charge density  $\sigma_{surface} = P \cos\theta$  on the sphere, as illustrated in Figure 1.

Integrating the surface polarization charge density over the Lorentz sphere, as illustrated in **Figure 1**, gives a contribution to the local electric field at the origin that is commonly referred to as the *Lorentz field*:

$$E_{\text{Lorentz Field}} = 2 \cdot \frac{1}{4\pi\varepsilon_0} \int_0^{\pi/2} \left[ \left( \frac{(P\cos\theta)\cos\theta}{R^2} \right) \right] R^2 \sin\theta d\theta \int_0^{2\pi} d\phi = \frac{1}{3} \left( \frac{P}{\varepsilon_0} \right).$$
(4)

Comparing Equations (1) and (4), we see that the Lorentz factor is: L = 1/3. Thus, L = 1/3 has long been used and assumed to be valid for dielectrics having either cubic crystalline structure or having an amorphous structure [5] [6].

For hyper-thin dielectrics, construction of a Lorentz sphere is difficult/impossible because the radius R of the sphere, as Lorentz suggested, should be chosen such that R is much larger than the lattice constant. For this reason, recently an alternative method to the standard Lorentz sphere approach is published for local electric field determination in hyper-thin dielectrics where the dielectric thickness may only be a few monolayers [7]. Fortunately, while this new method is used in local-field determination for hyper-thin dielectrics, the method is general enough that it can be easily extended to thicker dielectrics.



**Figure 1.** Lorentz sphere with a surface polarization charge density of  $\sigma_{\text{surface}} = P \cos \theta$ .

## 2. Approach for Local Electric Field Determination

The Lorentz sphere approach is apparently impossible to apply for hyper-thin dielectrics where the dielectrics may only be a few monolayers thick. Thus, an alternative approach is needed to determine the local electric field. To properly determine the local electric field, care must be taken to account for all charges that are acting at an arbitrary site of interest. As illustrated in **Figure 2**, the local electric field is the cumulative field (at the site of interest) due to all the other charges acting. For a parallel plate capacitor (the greatest interest here) the charges include: *external conduction charge density of*  $\sigma_q$  on electrode surfaces, *depolarization charge density of*  $-\sigma_P$  on surface of dielectric, *polarization charge density of*  $\sigma_P$  near to the surface of dielectric, plus the impact of the all the other dipoles in the bulk of the dielectric referred to here as *bulk dipoles*.

First, we consider the average/macroscopic electric field in the dielectric  $E_{diel}$  due to the *external charge density*  $\sigma_q$  on electrode plates plus the *dielectric surface depolarization charge density*  $-\sigma_P$ . Using Gauss' law for infinite parallel plate capacitors one obtains (in rationalized mks units):

$$E_{diel} = \frac{\sigma_q - \sigma_P}{\varepsilon_0} = \frac{\sigma_q - P}{\varepsilon_0} = \frac{\sigma_q - \varepsilon_0 \chi E_{diel}}{\varepsilon_0},$$
(5)

where  $\chi$  is the electric susceptibility. Using the fact that the dielectric constant k is given by  $k = 1 + \chi$ , and solving Equation (5) for  $E_{diel}$ , one obtains

$$E_{diel} = \frac{\sigma_q}{k\varepsilon_0} \,. \tag{6}$$

 $E_{diel}$  is the macroscopic/average-field in the dielectric. Integrating  $dV = -E_{diel} \cdot dx$  from x = 0 to  $x = t_{diel}$  gives  $E_{diel} = \Delta V_{diel}/t_{diel}$ , where  $\Delta V_{diel}$  is the voltage drop in the dielectric. It is important to emphasize that  $E_{diel}$  is the macroscopic electric field that must be used in Maxwell's equations for dielectric materials and is often referred to as the Maxwell field [Note that if electric displacement *D* is defined as  $D = \varepsilon E_{diel} = k\varepsilon_0 E_{diel} = \sigma_q$ , then the lines of *D* start and terminate only on conduction charges]. Since the dielectric surface depolarization charge density  $-\sigma_P$  is already incorporated into  $E_{diel}$ , then next we must account for the sheet of *near-surface polarization charge density*  $\sigma_P$  as well as the field contribution from the *bulk dipoles*.

As illustrated in Figure 2, the total local electric field  $E_{loc}$  at an arbitrary site of interest in the hyper-thin dielectric becomes:

$$E_{loc} = E_{\text{Conduction Charges}} + E_{\text{Dipoles}}$$

$$= E_{\text{Conduction Charges}} + E_{\text{Surface Dipole Layers}} + E_{\text{Bulk Dipoles}}$$

$$= E_{\sigma_q} + E_{-\sigma_P} + E_{\sigma_P} + E_{\text{Bulk Dipoles}}$$

$$= E_{diel} + E_{\sigma_P} + E_{\text{Bulk Dipoles}}.$$
(7)



Figure 2. This figure shows various charges that contribute to the local electric field. The physical thickness ( $t_{diel}$ ) of the hyper-thin dielectric illustrated is three monolayers.

The *surface dipole layers* (shown in **Figure 2**) have been treated in Equation (7) as closely spaced parallel sheets of uniform charge density  $\sigma_P$ , but with opposite polarity. The *conduction charge layer*  $\sigma_q$  plus the dielectric surface *depolarization charge layer*  $-\sigma_P$  have been incorporated to produce  $E_{diel}$ . The remaining charge layer  $\sigma_P$  will be treated separately and will be referred to as the *near surface polarization charge layer*  $\sigma_P$ . For parallel- plate capacitors, the field due to the sheet of near-surface polarization charge  $\sigma_P$  can be written as:

$$E_{\sigma_P} = \alpha \left(\frac{\sigma_P}{\varepsilon_0}\right) = \alpha \left(\frac{P}{\varepsilon_0}\right). \tag{8}$$

In Equation (8),  $\alpha$  is a dimensionless parameter that is purely device-structure dependent (not a function of polarization or field).  $\alpha$  can be determined from Coulomb's Law by integrating/summing over the uniform sheet of near-surface polarization charge  $\sigma_P$  [7]. For an infinite parallel plate capacitor structures,  $\alpha = 1$ .

Next we consider the bulk dipole contribution to the local electric field. As Lorentz orginially suggested, the dipolar contribution to the local electric field will be assumed to be proportional to the dielectric polarization P and can be written as,

$$E_{\text{Bulk Dipoles}} = \beta \left( \frac{P}{\varepsilon_0} \right), \tag{9}$$

where  $\beta$  is a dimensionless structure-dependent parameter. Thus, using Equations (7)-(9), we obtain

$$E_{loc} = E_{diel} + E_{\sigma_P} + E_{\text{Bulk Dipoles}}$$

$$= E_{diel} + (1 + \beta) \left(\frac{P}{\varepsilon_0}\right)$$

$$= E_{diel} + L \left(\frac{P}{\varepsilon_0}\right).$$
(10)

From Equation (10), we see that the Lorentz factor for dielectrics, in an infinite parallel plate capacitor arrangement, is given by:  $L = 1 + \beta$ .

#### 3. Lorentz Factor L Determination

The dielectric polarization P is the number of dipoles per unit volume, and for cubic symmetry we have:

$$P = \frac{\# \text{dipoles}}{\text{Volume}} = \frac{p_o}{abc} = \frac{q\delta}{abc},$$
(11)

where  $p_o = q\delta$  is molecular dipole moment (that is field induced) and the lattice parameters a = b = c for a crystal with cubic symmetry. Using Equations (9) and (11), we obtain for  $\beta$ :

$$\beta = \frac{\varepsilon_0 E_{\text{Bulk Dipoles}}}{P} = \frac{\varepsilon_0 abc}{q\delta} E_{\text{Bulk Dipoles}} , \qquad (12)$$

The method used for summing the bulk dipole contribution to the local electric field is illustrated in Figure 3.

Referring to Figure 3, the electrostatic potential at  $r_{,}$  due to the bulk dipoles located at r' in the bulk dipole layer, can be determined by:

$$V_{\text{Bulk Dipoles}}\left(r\right) = \frac{q}{4\pi\varepsilon_{0}} \sum_{\langle l,m,n \rangle} \left[ \frac{1}{\left| \mathbf{r}'(l,m,n) + \hat{z}(\delta/2) - \mathbf{r} \right|} - \frac{1}{\left| \mathbf{r}'(l,m,n) + \hat{z}(-\delta/2) - \mathbf{r} \right|} \right],\tag{13}$$

where:

$$\frac{|\mathbf{r}'(l,m,n) + \hat{z}(\delta/2) - \mathbf{r}|}{= \sqrt{[la - x]^2 + [mb - y]^2 + [nc + \delta/2 - z]^2}},$$
(14)

and



**Figure 3.** This figure shows the location of the bulk dipoles (in this cubic arrangement of dipoles). The dipole moment at each lattice site is  $p_0 = q\delta$  where the displacement  $\delta$  is induced by the electric field  $E_{diel}$ .

$$\left| \mathbf{r}'(l,m,n) + \hat{z}(-\delta/2) - \mathbf{r} \right| = \sqrt{\left[ la - x \right]^2 + \left[ mb - y \right]^2 + \left[ nc - \delta/2 - z \right]^2} .$$
(15)

In the above dipole summation, the bracketed lattice sites  $\langle l, m, n \rangle$  means that the summation is conducted over all lattice sites in the bulk dielectric, but with the origin excluded. Taking the gradient of the above dipolar electrostatic potential and considering only the *z* component of the dipolar field at origin, one obtains:

$$E_{\text{Bulk Dipoles}}^{z}(x, y, z)_{x=y=z=0} = -\left[\partial V_{\text{Bulk Dipoles}}(x, y, z)/\partial z\right]_{x=y=z=0}$$

$$= \frac{q}{4\pi\varepsilon_{0}} \sum_{\langle l,m,n \rangle} \left[ -\frac{nc + \delta/2}{\left[\left(la\right)^{2} + \left(mb\right)^{2} + \left(nc + \delta/2\right)^{2}\right]^{3/2}} + \frac{nc - \delta/2}{\left[\left(la\right)^{2} + \left(mb\right)^{2} + \left(nc - \delta/2\right)^{2}\right]^{3/2}} \right]^{3/2} \right]^{3/2}$$
(16)

Using Equation (12) and Equation (16), we obtain the structural parameter  $\beta$  for the bulk dipoles:

$$\beta = \frac{abc}{4\pi\delta} \sum_{\langle l,m,n \rangle} \left[ \frac{nc - \delta/2}{\left[ \left[ \left( la \right)^2 + \left( mb \right)^2 + \left( nc - \delta/2 \right)^2 \right]^{3/2} \right]^{3/2}} - \frac{nc + \delta/2}{\left[ \left( la \right)^2 + \left( mb \right)^2 + \left( nc + \delta/2 \right)^2 \right]^{3/2}} \right].$$
(17)

## 4. Layered Bulk-Dipole Summations

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It is found to be most convenient (in order to gain relatively rapid convergence) to do bulk layered-dipole summations as originally suggested by Nijboer and De Wette [8]. By layered summation, it is meant that for each integer value of *n* [representing each monolayer of added bulk-dipole dielectric thickness] the field contribution to the local electric field  $E_{loc}$  (at the origin) is fully comprehended for all dipoles in the *x*-*y* plane. This is accomplished by allowing *l* and *m* to range as far as needed such that dipoles contributions outside this range make a negligible contribution to  $E_{loc}$ . Thus, in the layered dipole summation approach,  $\beta$  is given by:

$$\beta = \sum_{-n}^{n} \beta_n , \qquad (18)$$

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where  $\beta_n$  is the full dipolar contribution from the n<sup>th</sup> layer of bulk dipoles. This layered-dipole summation approach for an elementary cube of dipoles is illustrated in **Figure 4**.

The dipole sum for the elementary cube shown in Figure 4 is  $\beta = 0$ . However, it is very important to emphasize that the net dipole sum is  $\beta = 0$  for all cubes, independent of cube size. This is shown in Figure 5 where a



**Figure 4.** Layered bulk-dipole summation approach illustrated for an elementary cube. For the n = 0 reference plane, the contribution from dipoles in this plane  $\beta_0$  is opposite (negative) to the macroscopic field in the dielectric  $E_{diel}$ . The contribution from dipoles above n = 0 plane ( $\beta_1$ ) and below the reference plane ( $\beta_{-1}$ ) are positive such that net contribution to the local electric field is zero ( $\beta = \beta_{-1} + \beta_0 + \beta_1 = 0$ ) for the elementary cube. For calculation purposes, a = 2.5 Å was used.



**Figure 5.** Monolayer-by-monolayer bulk-dipole contributions to  $\beta$ . For demonstration purposes, the planar bulk-dipole summations were conducted with 1 and m varying from -25 to +25. Each monolayer of dielectric thickness (assumed to be a = b = c = 2.5 Å) is indexed by *n* so that the thickness impact on  $\beta$  can be studied in detail. The reference plane (n = 0) produces the greatest single-layer bulk-dipole contribution to  $\beta$ . The negative sign means that the local field contribution of the  $\beta_0$  reference plane is actually opposite to  $E_{diel}$ . Note that each of the other monolayers of dielectric make a much smaller but positive contribution to  $\beta$ . Note also that when 1, m, and n all range from -25 to +25, the macroscopic sample is in the shape of a perfect cube and the dipole sum, as expected, produces  $\beta = 0$ .

more extensive layered dipole summation is illustrated with l, m, and n all extending from -25 to +25 lattice sites. Note that, even for this much larger cube of dipoles, the net dipole summation  $\beta$  is still zero.

**Figure 5** and the inserted summation table are very useful. For the n = 0 reference plane of bulk dipoles, that contains our origin, the reference-plane bulk-dipole summations (with *l* and *m* ranging from -25 to +25 lattice sites) produce  $\beta = \beta_0 = -0.7012$ . The negative sign means that the contribution of the dipoles to the local electric field in the reference plane is opposite to  $E_{diel}$ . Let us now add a monolayer of bulk dipoles above and below the reference plane. The dipolar sum now becomes:  $\beta = \beta_0 + (\beta_{+1} + \beta_{-1}) = -0.7012 + (0.0437 + 0.0437) = -0.6138$ . If we add two monolayers of bulk dipoles above and below the reference plane, then the planar dipolar sum becomes:  $\beta = \beta_0 + (\beta_{+1} + \beta_{-1}) = -0.7012 + (0.0176 + 0.0176) = -0.5786$ . Note that, as we add more monolayers, above and below the reference plane,  $\beta$  becomes more positive. Finally, if we continue to add more and more monolayers, finally reaching a total of 25 monolayers above the reference plane and 25 monolayers below the reference plane, then the bulk planar dipolar sum will become:  $\beta = -0.7012 + (0.3506 + 0.3506) = 0$ . This is expected since, with *l*, *m*, and *n* all ranging from -25 to +25, the macroscopic dielectric is in the shape of a perfect cube and, in agreement with Lorentz, the bulk dipole sum for a perfect cube of dipoles is  $\beta = 0$ .

It is very important to emphasize that the n = 0 reference plane (that contains our origin) tends to dominate all the other individual dipole planes and its value is  $\beta_0 \approx -0.7$ . The reason for this is illustrated in **Figure 6**. All the dipoles in the n = 0 reference plane make a constructive contribution to the local electric field and the all these contributions are opposite to the macroscopic/Maxwell field  $E_{diel}$  in the dielectric.

The net contribution to the local electric field, from each of the other dipolar planes  $\beta_{n\neq 0}$  is in the same direction as  $E_{diel}$  but the individual-plane contributions are much smaller than  $\beta_0$ . Thus, the value of the Lorentz factor L for hyper-thin dielectrics (a few monolayers of dielectric in an infinite parallel-plate capacitor structure) will be reasonably close to  $L \approx 1 + \beta_0 \approx 1/3$  and this is found to be in good agreement with the usual Lorentz factor assumption. However, for very thick dielectrics we expect  $\beta$  to asymptotically approach 0; thus, we expect  $L = 1 + \beta$  to asymptotically approach 1 for very thick dielectrics.

Before leaving this section, a question should be addressed: what must be the extent of the planar dipole summations (*l* and *m*) to insure that the contribution of the dipoles in each plane is fully comprehended? First, we note that, in the layered dipole sum approach, the dielectric thickness  $t_{diel}$  fixes the number of *n* monolayers (in the z-direction) that must be included in the  $\beta$  summation. However, the extent of the planar x-y dipole summations is still an open question.

Since the n = 0 reference plane is the dominant plane, let us see how it is impacted as we extend the *l*, *m* summation range. The extended dipole summation results for the n = 0 reference plane are shown in Figure 7. It can be seen that planar dipole summation approach leads to a relatively rapid convergence for  $\beta_0$ . This is because



**Figure 6.** All bulk dipoles in the n = 0 reference plane make a constructive contribution to the local electric field and the local electric field contribution from this plane is opposite to macroscopic/Maxwell field  $E_{diel}$ .



**Figure 7.** For the reference plane (n = 0), the x-y planar dipole summations were conducted over the indicated l and m ranges. Planar dipole summation convergence occurs relatively quickly with a summation error of only 2% when l and m extend from -25 to +25 lattice sites. A summation error of <1% results when l and m extend from -50 to +50 lattice sites. The asymptotic value for  $\beta_0$  is -0.7166.

dipole fields drop off relatively sharply as  $(1/r^3)$ . A summation error of <1% occurs in  $\beta_0$  determination if the planar dipole summations (1 and m) are extended from -50 to +50 lattice sites. The infinite-plane asymptotic summation value for  $\beta_0$  (=-0.7166), shown in **Figure 7**, was obtained using a geometrical series approximation for the converging summation series [7]. This asymptotic value for  $\beta_0$  is consistent with values previously reported for the n = 0 reference plane: -0.7189 found in [9] and -0.72 found in [10] [So as to avoid confusion for reader, it should be noted that the actual value reported in [9] is -9.033622 but this value has to be divided by  $4\pi$  to be consistent with the rationalized mks system of units used in this paper]. Therefore, planar dipole summations with 1 and m ranging from -50 to +50 lattice sites tend to produce an error of <1% in  $\beta_0$  determination. Thus, the range of *l* and *m* for planar summations will be from -50 to +50 lattice sites in the rest of this paper.

## 5. Bulk Layered-Dipole Summations for Thicker Dielectrics

A detailed example of planar dipole sums, for a moderately thick dielectric of 9 monolayers, will be now be conducted and then the method will be generalized to very thick dielectrics. Shown in **Table 1** are the summation results for a dielectric that has a physical thickness of 9 monolayers thick (7 bulk dipole monolayers are summed). To minimize error (<1%) in the construction of **Table 1**, the x-y planar dipole summations were conducted with both *l* and *m* ranging from -50 to +50 lattice sites.

Since the local field contribution from the dielectric surface dipole-layers are already incorporated in the model, the bulk-dipole layer summations (shown in **Table 1**) are over 7 bulk monolayers of dipoles. As indicated in **Table 1**, the variation in L (with position within the dielectric thickness) was determined by shifting the reference plane from the bottom of the dielectric to the top. The results are illustrated in **Figure 8**.

These planar dipole summation results indicate that some *L* variation does exist within thickness the dielectric, but generally the variations are small. Also, we note that the average value of the Lorentz factor does increase with thickness, but the average value  $\langle L \rangle = 0.39$  is still consistent with the common practice of using L = 1/3 for hyper-thin dielectrics. The results from Table 1 are shown plotted in Figure 8.

Using the summation approach [used for the construction of **Table 1**], we show the results for much thicker dielectrics. For each added monolayer of thickness [assumed to be 2.5 Å and indexed by *n*], all planar summations (*l* and *m*) were extended from -50 to +50 lattice sites. When n also extends over the range -50 to +50, the dipole sum becomes  $\beta = 0$ , thus  $L = 1 + \beta = 1$ .

Table 1. Lorentz factor across dielectric thickness.							
Plane Identification Number n	Ref Plane at Top of Dielectric Thickness			Ref Plane in Middle of Dielectric Thickness			Ref Plane at Bottom of Dielectric Thickness
6							0.00876
5						0.00881	0.00881
4					0.00884	0.00884	0.00884
3				0.00887	0.00887	0.00887	0.00887
2			0.00894	0.00894	0.00894	0.00894	0.00894
1		0.03497	0.03497	0.03497	0.03497	0.03497	0.03497
0	-0.70994	-0.70994	-0.70994	-0.70994	-0.70994	0.70994	-0.70994
-1	0.03497	0.03497	0.03497	0.03497	0.03497	0.03497	
-2	0.00894	0.00894	0.00894	0.00894	0.00894		
-3	0.00887	0.00887	0.00887	0.00887			
-4	0.00884	0.00884	0.00884				
-5	0.00881	0.00881					
-6	0.00876						
Dipole Sum: $\beta =$	-0.63075	-0.60453	-0.60440	-0.60437	-0.60440	-0.60453	-0.63075

0.39560

0.39547

0.39563

0.39560

0.39547

0.36925



Figure 8. Lorentz factor *L* determination for a dielectric of physical thickness of 9 monolayers. Only 7 monolayers are actually considered in the bulk dipole sum because the contribution of the surface dipole layers, to the local electric field, has already been included in model. The *L* variation within the thickness of hyper-thin dielectrics is found to be relatively small. Location points for the bulk dipole summations are indicated. The average value of L (*<L>* = 0.39) is in reasonably good agreement with the usual assumed Lorentz value of L = 1/3 for thin dielectrics.

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 $L = 1 + \beta =$ 

<L>=

0.36925

0.38804

**Figure 9** needs some discussion. The *flat-top region* in **Figure 9(d)** occurs because the full cube (with *l*, *m*, and *n* all ranging from -50 to 50 lattice sites) can fully fit within the dielectric thickness; in these locations  $\beta = 0$ , thus  $L = 1 + \beta = 1$ . A very thick dielectric (1001 monolayers) is illustrated in **Figure 10**. In most regions of the thick dielectric *l*, *m*, *n* can range unrestricted from -50 to +50 lattice sites so  $\beta = 0$  and  $L = 1 + \beta = 1$  over most



**Figure 9.** Shows Lorentz factor *L* as a function of position in dielectric thickness. The average value of the Lorentz factor  $\langle L \rangle$  is also shown. (a)  $t_{diel} = 21$  monolayers. (b)  $t_{diel} = 51$  monolayers. (c)  $t_{diel} = 101$  monolayers. (d)  $t_{diel} = 201$  monolayers. More variation about the mean value  $\langle L \rangle$  occurs in thicker dielectrics because layered bulk-dipole summation index *n* becomes truncated near the top and bottom portions of the dielectric. Near the top and bottom regions of the dielectric L = 2/3 while  $L \approx 1$  in the other regions of a thick dielectric.



**Figure 10.** Note that if the dielectric is very thick, then bulk-dipole cubes (with *l*, *m*, and *n* all ranging from -50 to +50) can easily fit within the thickness of the dielectric and  $\beta = 0$  for these cubes. Therefore, the average value of the Lorentz factor  $\langle L \rangle = 1 + \beta$  will asymptotically approach 1 for very thick dielectrics. At top and bottom surfaces of the thick dielectric the summation over n becomes truncated such that  $L = 1 - 0.36 \approx 2/3$ .

of the very thick dielectric. However, as one approaches the top and bottom of the thick dielectric, the summation over *n* [Equation (18)] becomes truncated such that at the top and bottom surfaces of the thick dielectric layer:  $L = 1 - 0.36 \approx 2/3$ . The dipole layer truncation, as one approaches the top and bottom dielectric surfaces, will cause more variation in *L* (from 2/3 to 1) within the thickness of thicker dielectrics.

The above results are summarized into a single curve (see Figure 11) where the average Lorentz factor  $\langle L \rangle$  is plotted versus dielectric  $t_{diel}$ . Note that for very thin films,  $t_{diel} \langle 20 \text{ monolayers}$ , the standard Lorentz value of L = 1/3 is reasonable. For dielectric thicknesses  $20 \langle t_{diel} \rangle \langle 50 \text{ monolayers}$ , the average Lorentz values are much closer to 3/5. For dielectric thicknesses  $50 \langle t_{diel} \rangle \langle 100 \text{ monolayers}$ , the average Lorentz value increases to approximately 4/5. For dielectric thicknesses  $100 \langle t_{diel} \rangle \langle 200 \text{ monolayers}$ , the average Lorentz value is much closer to 9/10. For dielectric thicknesses  $t_{diel} \rangle \langle 200 \text{ monolayers}$ , the average Lorentz value is much proaches 1.

### 6. Impact of Increases in L on Dielectric Breakdown Strength

One of the great mysteries in time-dependent dielectric breakdown (TDDB) physics has been: why does the dielectric breakdown strength  $E_{bd}$  reduce with increases in dielectric thickness  $t_{diel}$ ? Until now, only *extrinsic arguments* have been given: thicker dielectrics are hypothesized to have greater pre-existing defect densities thus making it easier for percolation path development resulting in lower breakdown strength.

The fact that  $E_{bd}$  reduces with  $t_{diel}$  is well established. Early investigations by Klein and Gafni showed this to be true [11]. They found that, by testing silicon oxides deposited on glass,  $E_{bd}$  reduces with increases in  $t_{diel}$ . While the quality of these early-produced deposited-oxides can be certainly questioned, extensive work done much later by Dumin [12] showed clearly that, for thermally-grown semiconductor-quality SiO<sub>2</sub> [see Figure 12],  $E_{bd}$  reduces with increases in  $t_{diel}$ . Figure 12 indicates that, for SiO<sub>2</sub>,  $E_{bd} \sim (1/t_{diel})^s$  with s = 0.22. The reduction in  $E_{bd}$  with increases in  $t_{diel}$  is not just a SiO<sub>2</sub> characteristic but seems to be a general trend for all polar solid dielectrics [13] [14].

The reduction in  $E_{bd}$  with  $t_{diel}$  is also reflected in time-dependent dielectric breakdown (TDDB) data. For constant field, thicker dielectrics fail faster, or equivalently, for a constant time-to-failure TF (e.g., TF = 10 sec), the *Failure Field* reduces with  $t_{diel}$ . The Suehle TDDB data [15], reproduced in **Figure 13(a)**, are in good agreement with the Dumin breakdown data. The Wu TDDB data [16] for hyper-thin SiO<sub>2</sub> [when stressing voltages are converted to fields ( $E_{ox} = \Delta V/t_{ox}$ )] are shown in **Figure 13(b**). While there is more dispersion in the Wu TDDB data, a power-law dependence (s = 0.18) is still evident.

The fact that thicker dielectrics breakdown at lower fields (and have lower TDDB performance) has been a great mystery and challenge for TDDB modeling. For *current-based TDDB models*, why would thinner SiO<sub>2</sub>



**Figure 11.** Average Lorentz factor as a function of dielectric thickness. For a very thin dielectrics ( $t_{diel} < 20$  monolayers),  $< L > \approx 1/3$ . For very thick dielectrics, <L> increases monotonically, asymptotically approaching 1.



Figure 12. Dumin breakdown data [12] for SiO<sub>2</sub>. Shows  $E_{bd}$  versus dielectric thickness  $t_{ox}$ . The breakdown data shows an inverse-thickness power-law dependence with s = 0.22. The breakdown strength  $E_{bd}$  reduces sharply initially with dielectric thickness but then a more gradual reduction with dielectric thickness.



**Figure 13.** (a) Re-plotting of the Suehle TDDB data [15], using Failure Field @ TF = 10 sec, reveals a power-law dependence (s = 0.22) reduction in failure-field with  $t_{ox}$  varying from 65 Å to 220 Å. (b) When converted to field ( $E_{ox} = \Delta V/t_{ox}$ ), re-plotting of the Wu TDDB data [16], using Failure Field @ TF = 10 sec, also indicates a power-law dependence (s = 0.18) with  $t_{ox}$  varying from 16.5 Å to 26.7 Å.

dielectrics have better breakdown strength and better TDDB performance when they tend to have higher leakages? As for *field-based models*, why isn't the breakdown strength independent of thickness? To circumvent these fundamental reliability physics questions, TDDB models have tended to use the *extrinsic hypothesis*: *thicker dielectrics must possess a greater pre-existing defect density*. The extrinsic hypothesis is illustrated in **Figure 14**.

The *extrinsic hypothesis* (of a higher density of pre-existing defects in thicker dielectrics) has been widely embraced, without serious objections. The extrinsic hypothesis has gained wide acceptance because the thickness dependence can be easily incorporated into all commonly used TDDB models: *E*, 1/*E*, Root-*E*, or  $V^{-n}$  [19]. All of these TDDB models have an adjustable pre-factor term into which the empirical thickness dependence  $(1/t_{diel})^s$  can be easily incorporated (as is commonly done for area scaling). Thus, all TDDB models tend to circumvent the thickness-dependence physics by simply hypothesizing that thicker dielectrics must have a much higher density of pre-existing defects.

After 50 years of intensive and extensive SiO<sub>2</sub> dielectric research (by hundreds of investigators), if higher defect density was the fundamental reason for  $E_{bd}$  reduction with  $t_{diel}$ , then surely someone would have found (as is allowed by the Poisson distribution) a single thick-oxide capacitor having a breakdown strength of >20 MV/cm.



Figure 14. (a) Pre-existing defects can exist in dielectrics at time zero. Defects can be also generated during TDDB testing. (b) Percolation path develops and breakdown occurs when a combination of pre-existing defects and generated defects form a continuous conduction path. [17] [18].

However, apparently, these thick-oxide/high-breakdown-field unicorns do not exist. Thus, the extrinsic hypothesis will be rejected and an effort is undertaken to find a more fundamental physics-based model for describing why thicker dielectrics have lower breakdown strength and lower TDDB performance. This has great relevance for low-k interconnect dielectrics [20] and high-voltage capacitors [21] where the dielectrics can be relatively thick.

The extrinsic hypothesis (of higher defect density) as for why  $E_{bd}$  reduces with  $t_{diel}$  has never been seriously challenged because, until now, there has never been an intrinsic/fundamental-physics reason presented for this. However, the increase in *L* with dielectric thickness (shown in **Figure 11**) is just the fundamental physics that is required to explain why the  $E_{bd}$  reduces with  $t_{diel}$ . Using Equation (3), we see that the  $E_{bd}$  can be written as:

$$E_{bd} = \frac{(E_{loc})_{bd}}{1 + L(k-1)}.$$
(19)

It is clear from Equation (19) that the breakdown strength of the dielectric is predicted to reduce when the dielectric constant k increases [22] and also this work shows that  $E_{bd}$  reduces with increases in L (which increases with thickness).

Using Equation (19), with the thickness dependence of  $\langle L \rangle$  shown in Figure 17, Figure 15 shows the model fit to the Dumin data. A constant value  $(E_{loc})_{bd} = 40$  MV/cm was used in the modeling and the relatively good fit to the breakdown data suggests that all SiO<sub>2</sub> thicknesses tend to breakdown at approximately same local electric field.

A local electric field of 40 MV/cm corresponds to a bond strength of 2.7 eV [19]. This is in good agreement with the energy required to break bond-coordination [19] [23], when the Si-ion goes from its normal 4-fold to 3-fold coordination in the  $O-Si \equiv O_3$  tetrahedral structure.

#### 7. Impact of Local Electric Field on Bond Breakage

The local electric field  $E_{loc}$ , that distorts polar bonds (by stretching, compressing, bending, etc.) and brings about the polarization, also serves to weaken polar molecular bonds. The local electric field thus reduces the bond strength  $\Delta H$  (activation energy required for bond breakage). Weakening of the molecular bonds is very important for any bond-breakage process, regardless of the actual breakage mechanism (standard Boltzmann processes and/or current driven processes). In previous work [23], this author has shown that the reduction bond strength can be described by:

$$\Delta H = \Delta H_0 - p(i, j) \cdot E_{loc} = \Delta H_0 - p(i, j) \cdot \left[1 + L(k-1)\right] E_{diel} , \qquad (20)$$

where  $\Delta H_0$  is the energy required to break the bond in the absence of field and p(i,j) is the dipole moment associated



**Figure 15.** Comparison of model with Dumin breakdown data. The model shows clearly that  $E_{bd}$  reduces with dielectric thickness as  $(1/t_{ax})^s$ . The relatively good model fit to Dumin data suggests that the local electric field at breakdown is a constant  $(E_{lac})_{bd} \approx 40$  MV/cm for all SiO<sub>2</sub> thicknesses.

with coordination bond breakage. The parameters *i* and *j* are the repulsive and attractive exponents, respectively, associated with the Mie-Gruneisen bonding potential. For strong polar/ionic bonding, the power-law molecular repulsive component "*i*" and the attractive ionic component "*j*" are typically given by i = 9 and j = 1, respectively. These polar molecular components produce a dipole moment for bond breakage of [23]

$$p(9,1) = (z^* e) r_0 [\eta(9,1)]^{-1} = (z^* e) r_0 (1.67).$$
(21)

For SiO<sub>2</sub>, with tetrahedral bonding, the Si cat-ion has a charge state of  $z^*e = 4f^* = 4(0.6)e = 2.4e$  (because the Si-O bond is about 60% polar). With a Si-O bond distance of  $r_0 = 1.7$  Å, then  $p(9,1) \approx 6.8e$  Å. Thus, Equation (20) becomes:

$$\Delta H = \Delta H_0 - (6.8e \text{ Å}) E_{loc} = \Delta H_0 - (6.8e \text{ Å}) [1 + L(k-1)] E_{diel}, \qquad (22)$$

where k (=3.9) is the dielectric constant for SiO<sub>2</sub>. With a bond strength of  $\Delta H_0 = 2.7$  eV, then the ultimate breakdown strength ( $\Delta H \rightarrow 0$ ) for very thin SiO<sub>2</sub> (with L = 1/3) becomes:  $(E_{diel})_{bd} \approx 20$  MV/cm whereas the ultimate breakdown strength for very thick SiO<sub>2</sub> (with L = 1) is  $(E_{diel})_{bd} \approx 10$  MV/cm.

## 8. Impact of Local Electric Field on TDDB

Time-dependent dielectric breakdown (TDDB) is a very important failure mechanism for metal oxide field effect transistors (MOSFETs) and capacitors [24]-[29]. As for the thermochemical TDDB model [4] for SiO<sub>2</sub>, bond strength  $\Delta H$  reduction occurs because of the local electric field and bond breakage occurs due to Boltzmann/thermal processes:

$$TF = A_0 \exp\left[\frac{\Delta H}{K_B T}\right] = A_0 \exp\left[-\frac{\left(6.8e\ \text{\AA}\right)\left[1 + L(k-1)\right]E_{diel}}{K_B T} + \frac{\Delta H_0}{K_B T}\right]$$
  
$$= A_0 \exp\left[-\gamma E_{diel} + \frac{\Delta H_0}{K_B T}\right],$$
(23)

where the field acceleration parameter  $\gamma$  is given by :

$$\gamma = \frac{p_{eff}}{K_B T} = \frac{\left(6.8e \text{ Å}\right)\left[1 + L\left(k - 1\right)\right]}{K_B T}.$$
(24)

For very thin SiO<sub>2</sub> dielectric (with  $L \approx 1/3$ ) the effective dipole moment for bond breakage is  $p_{eff} = 13.4e$  Å and this value is widely observed in the TDDB data for thin SiO<sub>2</sub> [23]. However, for very thick SiO<sub>2</sub> (with  $L \approx 1$ ), Equation (24) predicts that  $p_{eff}$  should approximately double for very thick dielectrics. Therefore, while the breakdown strength  $E_{bd}$  is expected to decrease with thickness, the field acceleration parameter  $\gamma$  is expected to increase with dielectric thickness (because L increases with thickness)

#### 9. Discussion

The local electric field  $E_{loc}$  (the field that polarizes, distorts, and weakens polar molecular bonds) is given by  $E_{loc} = [1 + L(k-1)]E_{diel}$  where L is the Lorentz factor, k is the dielectric constant, and  $E_{diel}$  is the macroscopic/average/Maxwell field in the dielectric. It has been shown that the Lorentz factor can be expressed as  $L = 1 + \beta$ , where  $\beta$  is determined purely from bulk dipole summation.

It has been determined that the n = 0 reference bulk-dipole plane dominates all the other individual planes above and below the reference plane. All the bulk dipoles in this n = 0 reference plane make a constructive contribution to the local electric field that is opposite to the macroscopic field  $E_{diel}$  in the dielectric. Therefore, the bulk-dipole sum for this n = 0 reference plane is negative and for an infinite reference plane of bulk dipoles:  $\beta_0 = -0.72$ .

As for the layered bulk-dipole summations, it has been shown in this work that beyond 50 lattice constants the planar bulk-dipole contributions to the local electric field are very small/negligible. Thus, a planar bulk-dipole sum with (l,m) from  $-\infty$  to  $+\infty$  monolayers is approximately equal (an error of <1%) to the planar bulk-dipole sum with (l,m) varying from -50 to +50 monolayers.

Since the other individual bulk-dipole planes  $(n \neq 0)$  make a positive, but much smaller contribution to the local electric field, then for very thin dielectrics:  $L = 1 + \beta \approx 1 + \beta_0 = 1 - 0.72 \approx 1/3$ . However, for much thicker dielectrics the cumulative positive contributions from all the other bulk-dipole layers, above and below the n = 0 reference plane, start to make a significant contribution to the local electric field such that  $\beta = \beta_0 + \text{Sum}(\beta_{n\neq 0}) \approx 0$  and  $L = 1 + \beta \approx 1$ . Thus, for any location in the dielectric where the thickness of the dielectric is sufficient to allow 50 monolayers of bulk-dipoles above and 50 monolayers of bulk-dipoles below the n=0 reference plane, then a perfect cube of dipoles forms and  $\beta = 0$ . Thus,  $L = 1 + \beta \approx 1$  at any location in a thick dielectric where (l,m,n) can all be summed unrestricted from -50 to +50 monolayers.

Even though two different thicknesses of a dielectric may have exactly the same macroscopic field  $E_{diel}$  in the dielectric, the local electric field is not the same. The thicker dielectric will have a larger L and thus a larger local field  $E_{loc}$ . Because of the higher local electric field  $E_{loc}$  in the thicker dielectric, the thicker dielectric will have lower breakdown strength. The definitions of thin and thick dielectrics have been established through planar bulk-dipole summations for L determination. L increases from  $L \approx 1/3$  for thin dielectrics (<20 monolayers) to nearly  $L \approx 1$  for very thick dielectric (>200 monolayers). The increase in Lorentz factor L with increase in dielectric thickness  $t_{diel}$  leads to higher local electric fields that produce more polar-bond distortion, thus a weakening of such bonds. Bond weakening leads to lower breakdown strength in thicker dielectrics and is independent of actual bond-breakage mechanism (field, current, or hydrogen release).

While the  $E_{bd}$  tends to reduce with dielectric thickness, all thicknesses tend to breakdown at approximately the same local electric field. As for SiO<sub>2</sub>, the local electric field at breakdown is  $(E_{loc})_{bd} \approx 40$  MV/cm and is independent of thickness. This local electric field (40 MV/cm) corresponds to a coordination bond energy of 2.7 eV. This is consistent with the reported energy required to transform the silicon-ion from four-fold coordination to three-fold coordination in the  $O - Si \equiv O_3$  tetrahedral structure.

#### **10. Conclusion**

A Lorentz factor of L = 1/3 has long been assumed valid without consideration for dielectric thickness. While L = 1/3 is valid for thin dielectrics (having either cubic crystalline or amorphous structure), it is not valid for thick dielectrics. *L* increases from  $L \approx 1/3$  for thin dielectrics (<20 monolayers) to nearly  $L \approx 1$  for very thick dielectric (>200 monolayers). The increase in *L* with  $t_{diel}$  leads to higher local electric fields that produce more polar-bond distortion and thus a weakening of such bonds. Bond weakening leads to lower breakdown strength in thicker dielectrics and is independent of actual bond-breakage mechanism (field, current, or hydrogen release). While the  $E_{bd}$  tends to reduce with thickness, all thicknesses tend to breakdown at the same local electric field. As for SiO<sub>2</sub>, the local electric field at breakdown is  $(E_{loc})_{bd} \approx 40$  MV/cm and is independent of thickness. This local

electric field of 40 MV/cm corresponds to an energy of 2.7 eV that is required to break bond-coordination in the  $O - Si \equiv O_3$  tetrahedral arrangement.

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