

# Modeling and Simulation in Spectroscopic Study

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Received 3 May 2014; revised 5 June 2014; accepted 2 July 2014

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Dear Readers:

Modeling and simulation has been increasingly applied to spectroscopic study over the last decade. Modeling is the process of producing a model; a model is a representation of the construction and working of some system of interest. A model is similar to but simpler than the system it represents. One purpose of a model is to enable the analyst to predict the effect of changes. A simulation of a system is the operation of a model of the system. The model can be reconfigured and experimented with, usually, this is impossible, too expensive or impracticable to do in the system it represents. The operation of the model can be studied, and hence, properties concerning the behavior of the actual system or its subsystem can be inferred. In its broadest sense, simulation is a tool to evaluate the performance of a system, existing or being proposed, under different configuration of interest and over long periods of real time.

Through the following representative example of Raman and infrared spectroscopic study, let us look at how modeling and simulation plays an indispensable role in predicting and interpreting spectra of matters.

Spinel phases of lithium manganese oxides are important positive electrode and lithium-selective ion-sorption materials. Vibrational spectroscopy is a powerful tool to examine structural changes in these oxides [1], being sensitive to the short range environment of oxygen coordination around both manganese and lithium ions. In order to establish the relationship between the lattice structures and the vibrational spectra of lithium manganese oxides, there is a need of a fundamental understanding of the lattice vibrations, particularly those that are Raman and infrared active. Amundsen *et al.* [2] used atomistic simulation methods based on an interatomic potential model to determine the normal modes of the zone-center ( $k = 0$ ) lattice phonons in spinel  $\text{LiMn}_2\text{O}_4$ . They compared the results of these calculations with Raman and infrared experimental data of the  $\text{LiMn}_2\text{O}_4$  bulk phase. Their simulation allowed the Raman and infrared spectra of spinel  $\text{LiMn}_2\text{O}_4$  to be assignable.

By atomistic simulation [3]-[5], the calculations were carried out using energy minimization procedures embodied in the GULP (General Utility Lattice Program) code. The interatomic potentials are based on the Born model of the solid, which includes a long-range Coulombic interaction and a short-range term to model overlap repulsions and van der Waals forces. The electronic polarizability of the ions is described by the shell model, which is effective in simulating the dielectric and lattice dynamical properties of metal oxides.

$$V_y(r_y) = \left[ -Z_i Z_j e^2 / r \right] + \left[ A_y \exp(-r/p_y) - C/r_y^6 \right]$$

The simulation of the  $\text{LiMn}_2\text{O}_4$  oxide lattice was performed using two-body potentials for Mn-O, Li-O and O-O interatomic interactions. The shell model parameters were refined to achieve the best fit of predicted wave numbers of Raman and infrared active vibrations to experimentally observed ones. These potential and shell model parameters are given in **Table 1**. The calculated lattice parameters and atomic positions for  $\text{LiMn}_2\text{O}_4$ , after energy minimization to generate a relaxed equilibrium structure, are shown in **Table 2**. The data show excellent agreement with the experimentally observed structure (**Table 2**).

The calculated and observed wave numbers, symmetry species and optical activities of the lattice vibrations for  $\text{LiMn}_2\text{O}_4$  are given in **Table 3**. The assumption made for this lattice is that cations and O anions are completely ordered. The number of lattice phonons and their symmetry species agree with factor group analyses for the primitive rhombohedral Bravais cells [6]. For  $\text{LiMn}_2\text{O}_4$ , there are 39 phonons distributed as:

$$\Gamma = A_{1g} + E_g + T_{1g} + 3T_{2g} + A_{2u} + 2E_u + 4T_{1u} + 2T_{2u}$$

As shown in **Table 3**, the calculations predict five Raman active modes for  $\text{LiMn}_2\text{O}_4$ , one of  $A_{1g}$  symmetries, one of  $E_g$  symmetries and three of  $T_{2g}$  symmetries. The Mn ions are motionless for all of the eigenvectors. The  $A_{1g}$  and  $E_g$  vibrations involve only oxide ion displacements. The  $T_{2g}(2)$  and  $T_{2g}(3)$  phonons are characterized by large oxygen motions and very small Li displacements, but the lowest energy  $T_{2g}(1)$  phonon derives predominantly from a vibration of the Li sublattice.

The experimental Raman data for the  $\text{LiMn}_2\text{O}_4$  bulk phase consist of a series of broad bands between 300 and  $700 \text{ cm}^{-1}$ . The two  $T_{2g}(1)$  and  $E_g$  modes which are calculated at  $354$  and  $434 \text{ cm}^{-1}$  are visible as weak features at  $365$  and  $432 \text{ cm}^{-1}$ . The calculations assign the three strong bands at  $480$ ,  $590$  and  $625 \text{ cm}^{-1}$  to the  $T_{2g}(2)$ ,  $T_{2g}(3)$  and  $A_{1g}$  phonons. The three bands correspond to the modes calculated at  $455$ ,  $597$  and  $598 \text{ cm}^{-1}$ .

As seen in **Table 3**, the calculations predict that four vibrational modes will appear in the infrared spectrum of  $\text{LiMn}_2\text{O}_4$ . Two higher wave number modes at  $517$  and  $633 \text{ cm}^{-1}$  are attributed to  $T_{1u}(3)$  and  $T_{1u}(4)$ , and two lower wave number modes at  $287$  and  $383 \text{ cm}^{-1}$  are ascribed to  $T_{1u}(1)$  and  $T_{1u}(2)$ . Experimentally, the former

**Table 1.** Interatomic potentials for  $\text{LiMn}_2\text{O}_4$ .

Short-Range			
interaction <sup>a</sup>	A (eV)	F (Å)	C (eV Å <sup>6</sup> )
$\text{Mn}^{4+} \cdots \text{O}^{2-}$	1345.15	0.324	0.0
$\text{Mn}^{3+} \cdots \text{O}^{2-}$	1267.5	0.324	0.0
$\text{Li}^+ \cdots \text{O}^{2-}$	426.48	0.3000	0.0
$\text{O}^{2-} \cdots \text{O}^{2-}$	22764.3	0.1490	43.0
Shell Model			
species	Y (e)	k (eV Å <sup>-2</sup> )	
$\text{Mn}^{4+}$	4.00	150	
$\text{Mn}^{3+}$	3.00	850	
$\text{O}^{2-}$	-2.96	57	

<sup>a</sup>Potential cutoff =  $12 \text{ Å}$ .

**Table 2.** Calculated and experimental crystal properties for  $\text{LiMn}_2\text{O}_4$ .

Unit Cell Parameters		
$a_0$ (calcd) (Å)	$a_0$ (exptl) (Å)	
8.2478	8.2470	
Atomic Coordinates (Fd3m)		
species	$x,y,z$ (calcd)	$x,y,z$ (exptl)
$\text{Mn}^{3+}, \text{Mn}^{4+}$ (16d)	0.5	0.5
$\text{Li}^+$ (8a)	0.125	0.125
$\text{O}^{2-}$ (32e)	0.2630	0.2625

**Table 3.** Calculated and observed vibrational spectral properties for  $\text{LiMn}_2\text{O}_4$ .

Wave number ( $\text{cm}^{-1}$ )		Symmetry Species	Activity
calcd	obsd		
118		$T_{2u}$	no
287	277	$T_{1u}(1)$	IR
317		$E_u$	no
354	365	$T_{2g}(1)$	Raman
357		$T_{1g}$	no
383	360	$T_{1u}(2)$	IR
434	432	$E_g$	Raman
454		$T_{2u}$	no
455	480	$T_{2g}(2)$	Raman
517	513	$T_{1u}(3)$	IR
570		$A_{2u}$	no
570		$E_u$	no
597	590	$T_{2g}(3)$	Raman
598	625	$A_{1g}$	Raman
633	615	$T_{1u}(4)$	IR
649		$A_{2u}$	no

two modes are observed as intense broad bands at 513 and 615  $\text{cm}^{-1}$ , easily resolved in the mid-infrared region. The latter two modes at 287 and 383  $\text{cm}^{-1}$  are discerned as a weak band at 277  $\text{cm}^{-1}$  and a shoulder band at 360  $\text{cm}^{-1}$ , respectively.

Modeling and simulation is of significant importance in spectroscopic study. In the absence of single-crystal data, it is difficult to assign Raman spectroscopic bands of matters to modes of different symmetries. Although infrared spectra of matters have been reported in numerous studies, the interpretations have been mainly limited to empirical comparisons. There is considerable uncertainty in the assignment of bands observed, especially with complex spectra consisting of multiple bands and/or overlapping bands. Therefore, it is really necessary to predict spectroscopic bands of matters by group theory. There is a need of theoretical models to simulate and understand spectroscopic data of matters.

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