

# Thermal and FT-IR Properties of Semiconducting SnO<sub>2</sub>-PbO-V<sub>2</sub>O<sub>5</sub> Glass System

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## Abstract

Melt quenched "SnO<sub>2</sub>(50-x)PbO:50V<sub>2</sub>O<sub>5</sub>" glass system containing x = 5, 10, 15 in molar ratio has been investigated. Effects of heating rate, glass transition, crystallization, melting temperature and infrared spectra of SnO<sub>2</sub> substituted PbO-V<sub>2</sub>O<sub>5</sub> glass system are reported. XRD results show that perfect vitrification has been achieved for all the glass samples after annealing at 150°C. DSC results have indicated that eutectic composition of the lead metavanadate has been maintained for all the glass systems up to 15 mole% of substitution. IR spectra for a SnO<sub>2</sub> substitution of 5 mole% V=O stretching frequency occur at 966 cm<sup>-1</sup> without appearance of any additional peak. But for 10 mole% and 15 mole% SnO<sub>2</sub> substituted samples, additional peaks appear at 1023 and 1005 cm<sup>-1</sup> indicating the effect of SnO<sub>2</sub> in the vanadate crystalline matrix such that there is an elongation of V=O bond. Since the crystalline matrix is affected, we can expect similar effect in the glass matrix also.

## Keywords

DSC, XRD, FT-IR, Lead Vanadate Glasses

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## 1. Introduction

Transport properties of semiconducting glasses are very interesting and provide useful information on the conduction mechanisms. Modern glasses are of crucial importance for electronics and have been widely used in industry, space explorations, computer memory units, etc. IR spectroscopy is a particularly suitable method for the structural studies of vanadate system in glass and crystalline forms because of the characteristic vanadate group's

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vibrational frequencies which can be easily identified. Structural models for PbO-V<sub>2</sub>O<sub>5</sub> glass systems have been discussed on the basis of IR spectroscopy [1] by comparing with those of known crystal structures. From IR and NMR spectroscopy, Hayakwa *et al.* [2] have proposed the existence of VO<sub>4</sub> tetrahedral and VO<sub>5</sub> trigonal bipyramids. The local structure around Pb in lead vanadate glasses mainly consists of PbO<sub>3</sub> trigonal and PbO<sub>4</sub> square pyramids. This study does not contradict the results obtained in earlier IR studies [3] that 50PbO:50V<sub>2</sub>O<sub>5</sub> glass system contains affected VO<sub>5</sub> groups but gives additional information regarding the local environment of vanadium as well as lead ions. The lead ions are located between the vanadate chains and layers. These ions affect the isolated V=O bonds by elongating them, thus reducing their frequency. In the present studies IR spectra have been recorded in order to verify if the substituent metal oxide is replacing PbO in the glass network of the equimolar lead vanadate *i.e.*, 50PbO:50V<sub>2</sub>O<sub>5</sub> glass system.

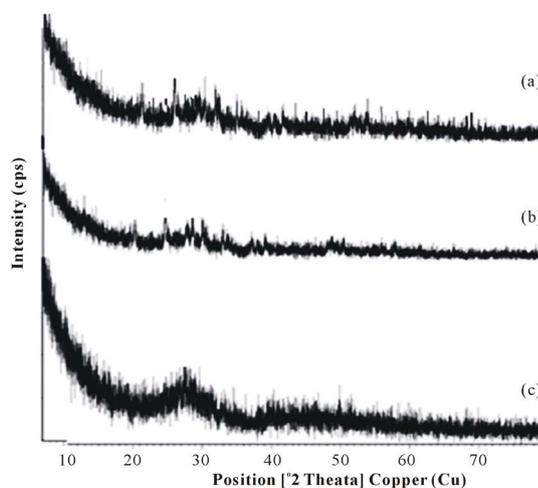
## 2. Experimental

The glasses were obtained by melting a chemically pure PbO, V<sub>2</sub>O<sub>5</sub> and SnO<sub>2</sub> in amounts varying from 5 to 15 mole% PbO and SnO<sub>2</sub> in glazed silica crucibles in the 800°C - 1200°C temperature range. Vitrification was achieved by rapid cooling of the melt using a roller technique. The glasses were subjected to crystallization at 380°C. The prepared samples were grounded into fine powder for X-ray, DSC and FT-IR studies. X-ray studies were carried out on a PAN Alytic X'Pert-PRO diffractometer using CuK<sub>α</sub> radiation at 1.5418Å and diffractometer settings in the 2θ range from 10°C - 70°C by changing the 2θ with a step size of 0.020. Differential Scanning calorimetry investigation of glass specimens was performed using DuPont, USA make model 2000 DSC instrument. DSC scans were conducted using 5 - 10 mg ground as-cast glass specimens which heats up at the rate of 10°C/min between 0°C and 600°C in a platinum crucible and alumina powder was used as the reference material. The density of the glass samples was determined by the Archimedes principle, using toluene as immersion liquid. In the present studies IR spectra were recorded for both the vitreous and non-vitreous samples in order to study the effect of a different metal oxide substitution in the place of PbO in the lead metavanadate glass systems.

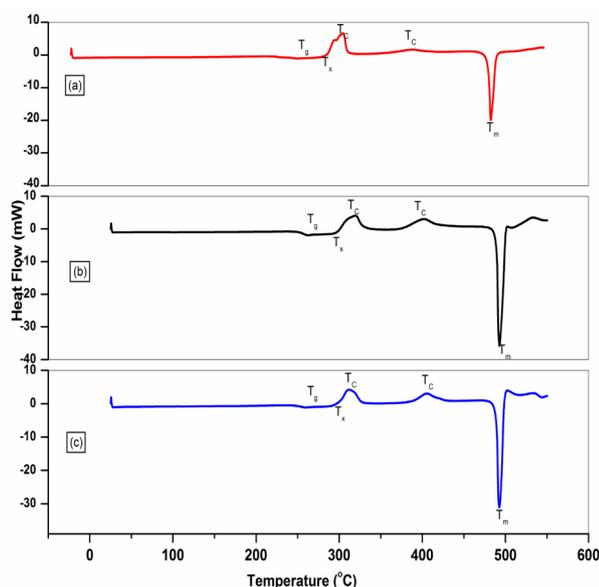
## 3. Results and Discussion

The X-ray diffractograms annealed at 150°C containing (x = 5, 10, 15 mole%) of SnO<sub>2</sub> showed no trace of crystallinity and are shown in **Figure 1**.

DSC investigations were conducted on the SnO<sub>2</sub>-PbO-V<sub>2</sub>O<sub>5</sub> glasses. The DSC patterns for these glass systems shown in **Figure 2** are slightly different when compared to the un-substituted system [4]. Values of glass transition temperature  $T_g$ , crystallization temperature  $T_c$ , melting temperature  $T_m$ , glass forming tendency  $K_g$  and densities of the xSnO<sub>2</sub>(50-x)PbO:50V<sub>2</sub>O<sub>5</sub> are given in **Table 1**.



**Figure 1.** X-ray diffractograms of xSnO<sub>2</sub>(50-x)PbO:50V<sub>2</sub>O<sub>5</sub> glass system annealed at 150°C. (a) x = 5 mole%; (b) x = 10 mole%; (c) x = 15 mole%.



**Figure 2.** Differential scanning calorimetry curves of  $x\text{SnO}_2(50-x)\text{PbO}:50\text{V}_2\text{O}_5$  glass system. (a)  $x = 5$  mole%; (b)  $x = 10$  mole%; (c)  $x = 15$  mole%.

**Table 1.** Values of glass transition temperature ( $T_g$ ), crystallization temperature ( $T_c$ ), melting temperature ( $T_m$ ) and glass forming tendency ( $K_g$ ) for the  $x\text{SnO}_2(50-x)\text{PbO}:50\text{V}_2\text{O}_5$ .

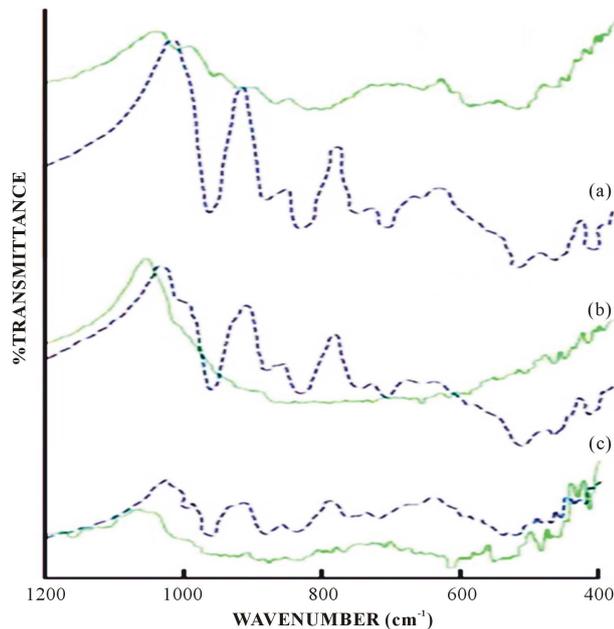
Glass Composition (mole%)			$T_g$ (°C)	$T_c$ (°C)	$T_m$ (°C)	$K_g$	D (g/cm <sup>3</sup> )
V <sub>2</sub> O <sub>5</sub> Content (mole%)	PbO Content (mole%)	SnO <sub>2</sub> Content (mole%)					
50	50	0	241	300,412	500	0.156	5.075
50	45	5	253	292,390	496	0.099	4.928
50	40	10	251	294,391	495	0.134	5.264
50	35	15	248	287,385	494	0.138	5.501

There is a slight change in  $T_g$  along with an increase in the number of crystallization peaks, the  $T_g$  values decrease with increasing SnO<sub>2</sub> contents, these results suggest that SnO<sub>2</sub> acts as a network modifier where as PbO acts as a network former [5]. Crystallization temperature  $T_c$  is the maximum temperature of the exotherm and onset of crystallization temperature,  $T_x$  is the beginning of the first exothermic reaction where the crystallization starts. As seen in **Figure 2** up to  $x = 15$  mole% there is only one endothermic peak corresponding to melting point. This indicates that the substituted samples behave like the eutectic composition upto  $x = 15$  mole%. In order to understand the devitrification tendency and thermal stability of the glass samples, glass forming tendency values  $K_g$  are calculated using the below equation.

$$K_g = \frac{T_x - T_g}{T_m - T_x} \quad (1)$$

Lower  $K_g$  suggests higher tendency of crystallization and lower thermal stability.  $K_g$  represents the temperature interval during nucleation [6]. These values indicate that the  $5\text{SnO}_245\text{PbO}:50\text{V}_2\text{O}_5$  glass samples have lowest thermal stability among four compositions, with a  $K_g$  value of 0.099. The densities seem to be increasing with an increase in SnO<sub>2</sub> substitution. The IR spectra for the glass as well as crystalline forms are for  $x\text{SnO}_2(50-x)\text{PbO}:50\text{V}_2\text{O}_5$  glass systems given in **Figure 3**. The observed IR band positions are given in **Table 2** and **Table 3**.

A consequence of disorder in the amorphous or glass systems when compared to crystalline forms is the



**Figure 3.** IR spectra of  $x\text{SnO}_2(50-x)\text{PbO}:50\text{V}_2\text{O}_5$  glass system; \_\_\_\_\_ glass; .....crystalline. (a)  $x = 5$  mol%; (b)  $x = 10$  mol%; (c)  $x = 15$  mol%.

**Table 2.** FT-IR data of  $x\text{SnO}_2(50-x)\text{PbO}:50\text{V}_2\text{O}_5$  glass system in both amorphous and crystalline forms.

Glass Composition (mole%)				
	State	$\nu(\text{V}=\text{O})$	$\nu(\text{VO}_2) \nu(\text{VO}_3)$	$\nu(\text{VOV})$
50V <sub>2</sub> O <sub>5</sub> :50PbO [2]	G	968	934	771
	C		892,869,842	765,723
50V <sub>2</sub> O <sub>5</sub> :50PbO [4]	G	964		770
	C	966	885,839	762,720
5SnO <sub>2</sub> :45PbO:50V <sub>2</sub> O <sub>5</sub>	G	912	873,831	754,717
	C	966	883,837	
10SnO <sub>2</sub> :40PbO:50V <sub>2</sub> O <sub>5</sub>	G		802	754,715
	C	966,1023	839,884	
5SnO <sub>2</sub> :35PbO:50V <sub>2</sub> O <sub>5</sub>	G	941	875,835	758,715
	C	962,1005	883,837	

**Table 3.** FT-IR data of  $x\text{SnO}_2(50-x)\text{PbO}:50\text{V}_2\text{O}_5$  glass system in both amorphous and crystalline forms.

Glass Composition (mole%)				
	State	CV	(VOV)	$\delta(\text{VO}_2, \text{VO}_3)$
50V <sub>2</sub> O <sub>5</sub> :50PbO [2]	G	646		423
	C	666	574	534,489,434
50V <sub>2</sub> O <sub>5</sub> :50PbO [4]	G	662		539,462,433
	C	668	570	532,470,438
5SnO <sub>2</sub> :45PbO:50V <sub>2</sub> O <sub>5</sub>	G		599,532	478,430
	C	667	530	
10SnO <sub>2</sub> :40PbO:50V <sub>2</sub> O <sub>5</sub>	G		599,522	478,462,435
	C	667,626	526	478
5SnO <sub>2</sub> :35PbO:50V <sub>2</sub> O <sub>5</sub>	G		545	484,412
	C	617	538	

$\nu(\text{V}=\text{O})$  symmetric stretching,  $\nu(\text{VO}_2) \nu(\text{VO}_3)$  asymmetric stretching,  $\nu(\text{VOV})$  bending frequency (symmetric and antisymmetric), C.V. combination vibration of  $(\text{VO}_3)_n$  single chain, G indicates glass.

breakdown of the wave vector selection ( $k$ -selection) rules, which allows electromagnetic radiation to couple with vibrations other than  $k = 0$ . As a result, unlike the crystalline case in which narrow well defined lines are observed, broad and diffuse bands representing a continuum of IR absorption result. Even though bands due to individual, localized structural units are observable, the identification of IR spectra of glasses alone is rather difficult unless crystalline spectra are also present. Hence the present discussion mainly corresponds to the IR bands observed in the devitrified samples.  $\text{SnO}_2$  substituted samples for  $x = 10$  and  $15$  mole% , there is an indication of additional bands at  $1023 \text{ cm}^{-1}$  beside  $966 \text{ cm}^{-1}$  and  $1005 \text{ cm}^{-1}$  beside  $966 \text{ cm}^{-1}$  the appearance of additional bands can be understood as the splitting of the original  $\text{V}=\text{O}$  stretching band into affected and unaffected components. The affected components arises due to the effect of dopant *i.e.*  $\text{SnO}_2$  in the present case on  $\text{V}=\text{O}$  bond the glass matrix. In this case from the IR spectral evidence we can summarize the following.

As  $\text{SnO}_2$  is substituted for  $\text{PbO}$ ,  $\text{SnO}_2$  is replacing  $\text{PbO}$  in the glass network in such a way that it affects  $\text{V}=\text{O}$  bond frequency, as evident by the shift in  $\text{V}=\text{O}$  band (at  $1023 \text{ cm}^{-1}$ ) whose intensity increases with the increase of  $\text{SnO}_2$  concentration. The affected  $\text{V}=\text{O}$  band (at  $966 \text{ cm}^{-1}$ ) indicates the presence of meta vanadate phase containing  $\text{PbO}$  in the usual evidence we can summarize the following. As  $\text{SnO}_2$  is substituted for  $\text{PbO}$ ,  $\text{SnO}_2$  is replacing  $\text{PbO}$  in the glass network in such a way that it affects  $\text{V}=\text{O}$  bond frequency, as evident by the shift in  $\text{V}=\text{O}$  band (at  $1023 \text{ cm}^{-1}$ ) whose intensity increases with the increase of  $\text{SnO}_2$  concentration. The affected  $\text{V}=\text{O}$  band (at  $966 \text{ cm}^{-1}$ ) indicates the presence of metavanadate phase containing  $\text{PbO}$  in the usual glass network. The occurrence of the band at  $1005 \text{ cm}^{-1}$  is due to the indirect effect of  $\text{SnO}_2$  on the  $\text{V}=\text{O}$  band in the new stable phase that is being formed. In  $x(\text{TiO}_2):(100-x)(\text{V}_2\text{O}_5)$  glass containing  $x = 20$  mole%, Dimitriev *et al.* [3] reported that the isolated  $\text{V}=\text{O}$  band is unaffected at  $1020 \text{ cm}^{-1}$ . It was observed by Dimitrov *et al.* [1] that in the IR spectra of  $x\text{PbO}:(1-x)\text{V}_2\text{O}_5$  glass systems as  $x$  is varied from 0 to 75 mole% , there appears a new band in the range  $970 - 955 \text{ cm}^{-1}$  beside  $1020 \text{ cm}^{-1}$  band. It has been suggested that [3]  $\text{Pb}^{2+}$  ions occupy a position between the  $\text{V}-\text{O}-\text{V}$  layers. They exercise a direct influence of the isolated  $\text{V}=\text{O}$  bonds of  $\text{VO}_5$  groups according to the scheme  $\text{Pb}^{2+} \dots \text{O}=\text{V}^{5+}$ . This causes an elongation of the affected  $\text{V}=\text{O}$  bond and a decrease in the vibrational frequency to  $970 - 950 \text{ cm}^{-1}$ . There may be unaffected  $\text{V}=\text{O}$  bonds whose vibrational frequency is still around  $1020 \text{ cm}^{-1}$ . With the increase of  $\text{PbO}$ , their number decreases and for eutectic composition  $50\text{PbO}:50\text{V}_2\text{O}_5$  (*i.e.*  $\text{PbV}_2\text{O}_6$  metavanadate) only one type of  $\text{VO}_5$  polyhedron results as is evidenced by the presence of a single high frequency band in the region  $955 - 970 \text{ cm}^{-1}$  [7]. This band is observed at  $966 \text{ cm}^{-1}$  in  $5\text{SnO}_2:45\text{PbO}:50\text{V}_2\text{O}_5$  system. As  $\text{SnO}_2$  concentration is increased to 10 mol% and 15 mole%, the appearance of  $1023 \text{ cm}^{-1}$  band indicates that  $\text{SnO}_2$  is replacing  $\text{PbO}$  in such a way that the  $\text{V}=\text{O}$  bond in the glass network of the eutectic composition is elongated [8].

#### 4. Conclusion

Perfect vitrification has been achieved for all the glass samples as can be seen from their X-ray diffractograms of the prepared samples after annealing at  $150^\circ\text{C}$  for two hours. DSC recordings show that eutectic composition of the lead metavanadate has been maintained for all the glass systems up to 15 mole% of substitution. The DSC data also indicate that all the glass systems are characterized by more than one crystallization peak. This can be thought as an evidence for the existence of more than one meta-stable phase in the glass systems. The dopant  $\text{SnO}_2$  is not divalent oxide like  $\text{PbO}$ . Besides  $\text{SnO}_2$  is known to be glass former unlike  $\text{PbO}$  which is replacing  $\text{PbO}$  in the glass network in such a way that the eutectic composition is maintained. In the present studies the IR spectra of  $\text{SnO}_2$  substituted samples of 5 mole%  $\text{V}=\text{O}$  stretching frequency occur at  $966 \text{ cm}^{-1}$  without the appearance of any additional peak. But for 10 mole% and 15 mole%  $\text{SnO}_2$  substituted samples, additional peaks appear at  $1023$  and  $1005 \text{ cm}^{-1}$  indicating the effect of  $\text{SnO}_2$  in the vanadate crystalline matrix such that there is an elongation of  $\text{V}=\text{O}$  bond. Since the crystalline matrix is affected, we can expect similar effect in the glass matrix also.

#### References

- [1] Dimitrov, V. and Dimitriev, Y. (1990) Structure of Glasses in  $\text{PbO}-\text{V}_2\text{O}_5$  System. *Journal of Non-Crystalline Solids*, **122**, 133-138. [http://dx.doi.org/10.1016/0022-3093\(90\)91058-Y](http://dx.doi.org/10.1016/0022-3093(90)91058-Y)
- [2] Hayakawa, S., Yoko, T. and Sakka, S. (1995) IR and NMR Structural Studies on Lead Vanadate Glasses. *Journal of Non-Crystalline Solids*, **183**, 73-84. [http://dx.doi.org/10.1016/0022-3093\(94\)00652-0](http://dx.doi.org/10.1016/0022-3093(94)00652-0)
- [3] Dimitrov, V. (1995) Structural Changes in Vitreous Vanadate Systems. *Journal of Non-Crystalline Solids*, **192**, 183-

186. [http://dx.doi.org/10.1016/0022-3093\(95\)00349-5](http://dx.doi.org/10.1016/0022-3093(95)00349-5)
- [4] Ramesh, K.V. (2000) Thermal, Electrical and Spectroscopic Studies of CuO, ZnO and TiO<sub>2</sub> Substituted for PbO in Eutectic Lead Vanadate Glass System. Ph.D. Thesis, Andhra University, Visakhapatnam.
- [5] Oz, B., Kabaci, I., Ovecoglu, M.L. and Ozen, G. (2007) Thermal Properties and Crystallization Behavior of Some TeO<sub>2</sub>-K<sub>2</sub>O Glasses. *Journal of the European Society*, **27**, 1823-1827.
- [6] Murugan, G.S. and Ohishi, Y.J. (2004) TeO<sub>2</sub>-BaO-SrO-Nb<sub>2</sub>O<sub>5</sub> Glasses: A New Glass for Waveguide Device Applications. *Journal of Non-Crystalline Solids*, **341**, 86-92. <http://dx.doi.org/10.1016/j.jnoncrysol.2004.04.006>
- [7] Tejeswara Rao, P. (2013) Preparation and Characterization of Some Lead Meta Vanadate Semi-Conducting Glass Systems with GeO<sub>2</sub>, Bi<sub>2</sub>O<sub>3</sub> and SnO<sub>2</sub> Substituted for Lead Oxide. Ph.D. Thesis, Andhra University, Visakhapatnam.
- [8] Tejeswara Rao, P., Ramesh, K.V. and Sastry, D.L. (2012) Electrical and Spectroscopic Studies of the CdO Substituted Lead Vanadate Glass System vs Crystalline Form. *New Journal of Glass and Ceramics*, **2**, 34-40. <http://dx.doi.org/10.4236/njgc.2012.21006>