Composition and Refractory Properties of Mixtures of Moroccan Silica-Alumina Geomaterials and Alumina

Chaouki Sadik^{1*}, Iz-Edinne El Amrani², Abderahman Albizane¹

¹Department of Chemistry, Faculty of Science and Technology, University Hassan II Mohammedia, Casablanca, Morocco; ²Department of Earth Sciences, Geo-Materials and Geo-Environment Team (Geo M&E), Scientific Institute, University Mohammed V Agdal, Rabat, Morocco.

Email: *schawki37@gmail.com

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ABSTRACT

Five mixtures (M1 to M5) of silica-alumina geomaterials and two varieties of alumina (A_P and A_R) were used for the elaboration of mullite refractory materials between 1500°C and 1600°C. An X-ray diffraction (XRD) analysis showed that the refractory samples are composed of mullite, corundum and silica. The length of the mullite crystals was measured by a method of image analysis of scanning electron microscopy (SEM). Chemical and mechanical properties of these materials were investigated and correlated with their microstructure. Resistance towards Acid Attack test showed that the refractory samples present good resistance, as well as, the alumina powder A_R obtained from waste of silica-alumina bricks proves to be efficient for an eventual use.

Keywords: Clay; Silica-Alumina; Mullite Refractory; Mechanical Properties

1. Introduction

Refractory materials most commonly used for lining furnace walls are synthesized mainly from silica-alumina geomaterials [1-3]. The control of the quality of refractory demands a wide knowledge in various scientific fields (geology, mineralogy, geochemistry, thermodynamics, mechanics …). In fact, the production chain has a number of steps (choice of raw material, mixture preparation, drying, pressing, firing …) requiring perfect mastery and which the slightest error or neglect of any parameter results in failures in the final refractory product quality.

Many research studies have been conducted, until now, on mullite (3Al₂O₃-2SiO₂) as a mineral phase with many properties: low thermal expansion and conductivity, excellent creep resistance, high temperature strength, and good chemical stability [4]. These studies have focused on their synthesis methods, their phase equilibrium, their microstructures and their thermo-mechanical properties [4,5].

Different types of mullite can be synthesized according to the nature of the raw materials and the used process [6-11]. The application fields of mullite are multiple

*Corresponding author.

and depend on the properties of each type. In fact, the performance of a refractory (good resistance of heat and thermal shock) is directly related to its texture and its richness in mullite.

Kaolinitic clays are used to make fire-clays and silicaclays refractory materials. Alumina-silica (cyanide, sillimanite and andalousite) and alumina hydrates (boehmite, diaspore, gibbsite) are the main raw materials for the alumina refractory products. Silica in a form of quartzite, sand or aggregates is used for the silica based refractory materials elaboration [1]. Kaolin has the advantage to be easily accessible and not expensive in natural open work seams [8]. The main mineral constituent of the kaolin is the kaolinite. Its theoretical formula is $Al_2O_3 \cdot 2SiO_2 \cdot 2H_2O$. The kaolinite is a phyllosilicate composed of a silica tetrahedral layer $[Si_2O_5]^{2-}$ and an octahedral hydroxide layer [Al₂(OH)₄]²⁺. Kaolinite firing induces numerous complex structural and microstructural transformations leading to the formation of mullite and a silica phase, mainly in a vitreous form, according to the following reaction [1]:

 $3(Al_2O_3 \cdot 2SiO_2 \cdot 2H_2O) = 3Al_2O_3 \cdot 2SiO_2 + 4SiO_2 + 6H_2O$

The aim of this study is to synthesize the mullite by high temperature treatment of local Moroccan silicaalumina geomaterials (kaolin clay, red clay and silica



sand), as well as two varieties of alumina, pure alumina (A_p) and alumina (A_R) recovered from waste of silicaalumina refractory bricks.

2. Materials and Experimental Procedure

The samples studied were made from a mixture selected among various compositions of kaolin clay, red clay and silica sand [8,12]. The clays were used in crude forms.

Kaolin clay (ArgK) is resulting from hydrothermal alteration of alkali granite of Oulmès, located in the center of the Moroccan Meseta. This alteration caused the kaolinization of alkali-feldspar of the granitic rock and gives an aluminous friable material (Table 1), rich in kaolin clay with an appreciable quantity of quartz, flakes of muscovite, and chloritized biotite (Table 2). Red Clay (ArgS) is a very fine-grained silt of lower Cretaceous age, rich in clay minerals and poor in carbonates (Tables 1 and 2). Very large outcrops of these clay deposits exist in the region of Safi. Clays of this region are intensely exploited by the industrial ceramists for the manufacture of bricks and tiles. Silica Sand (SabM) used in this study comes from an artisanal quarry, located at 3 km SE of the Mechraa Hammadi Dam. The sands appear as a thin layer of a white-pink and friable rock, belonging to the serie of limestone of the upper Jurassic (130 Ma).

The very fine alumina powders " A_P " commercialized by the industrial unit "Super Cerame" have characteristics and specific functions in terms of rheology, mechanical and chemical strength and refractoriness. Our choice was mainly guided by its high purity of alumina (99.9%). An X-ray diffraction (XRD) analysis showed that this powder is mainly composed of Corundum (**Figure 1**). The contamination level of silica-alumina refractory materials used as coatings in kilns in the ceramics industry is low. It is for this reason that firebrick having undergone three years of use as a coating in a kiln was finely milled to obtain a white powder " A_R ", and that contains 71.14% of alumina. Mineralogical analysis of this powder shows that it consists mainly of mullite (**Figure 1**).

X-Ray Diffraction (XRD) analyses of the two materials (ArgK and ArgS) were used to determine the nature of their clay minerals and semi-quantitative composition (**Table 2**). The Table also inserts the carbonate contents, measured using Bernard's calcimeter. The mineralogical composition of ArgK consists mainly on kaolinite and small quantities of quartz and illite. The following mineralogical phases were identified for the red clay (ArgS): illite as the principal mineral, with kaolinite and quartz. Other secondary minerals phases found in this clay are hematite and dolomite (**Figure 1**). Rational mineralogical composition analysis is showed in **Table 2**.

To attain the objectives of this research, we proceed in two steps.

2.1. Preparation of Mixtures

We prepare five mixtures (M1 to M5) that allowed obtaining ceramics resisting to a temperature exceeding 1500° C, which we can already considered as refractories (**Table 3**).

2.2. Preparation of Refractory Bricks and Crucibles

A quantity of 100 grams is prepared for each mixture (M1 to M5) according to the proportions indicated in **Table 3**. Each mixture is passed in planetary mill (mean grain size $<100 \ \mu$ m). After 10% moistening, the powders are shaped by uniaxial pressure at 30 MPa into molds of $100 \times 50 \ \text{mm}^2$. The obtained samples are submitted to a drying 110° C/24 h and calcination at 600° C during 1h. Then they are fired at high temperatures up to 1600° C (5°C/mn) in an oven (Thermolyne 46,200) [13]. During their firing, the briquettes are subjected to various tests to determine their physic-mechanical properties: thermal sh-

Table 1. Chemical compositions by XRF of the geomaterials used in this study (wt%) (Chemical analysis were performed in laboratories UATRS, CNRST "Morocco" in 2012).

	SiO ₂	TiO	Al ₂ O ₃	Fe_2O_3	MgO	CaO	Na ₂ O	K ₂ O	P_2O_5	PAF	Total
ArgK	57.20	0.08	29.40	4.50	0.65	0.26	1.55	3.08	0.00	3.28	100.0
ArgS	51.60	1.06	19,20	9.00	5.36	0.90	0.58	3.89	0.23	8.16	99.98
SabM	96.06	0.15	2.22	0.51	0.17	0.09	0.06	0.20	0.02	0.49	99.97

Table 2. Semi-quantitative analysis of clay mineral and carbonate content (Mineralogical analysis were performed in laboratories UATRS, CNRST "Morocco" in 2012).

Materials	Kaolinite	Illite	Quartz	Dolomite	Hematite	CaCO ₃ %
ArgK	44.00%	25.67%	30.32%	-	-	2.62
ArgS	21.35%	39.65%	25	10	4	2.14

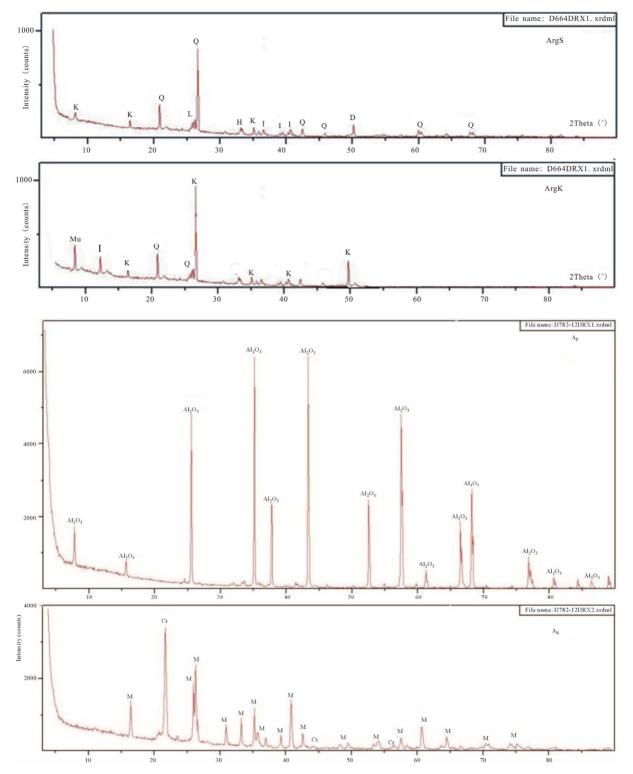


Figure 1. XRD patterns of materials. (K: Kaolinite, I: Illite, H: Hematite, Q: Quartz, D: Dolomite, Mu: Muscovite, M: Mullite, Cr: Cristobalite).

rinkage, porosity and flexural strength. The tests were performed according to the procedures recommended by ASTM [14,15].

Starting from the knowledge of the laboratory of Geo-

materials and Geo-Environment in the field of ceramics, we have elaborated refractory crucibles specifically adapted to our needs, which can solve the problems found in laboratory experiments. The studied mixtures are

 Table 3. Composition of mixtures from different types of geomaterials.

	ArgK	ArgS	SabM	$\mathbf{A}_{\mathbf{P}}$	A_R
M1	30	10	10	25	25
M2	40	15	5	-	40
M3	50	-	20	30	-
M4	-	-	40	60	-
M5	-	-	40	-	60

moistened until obtaining homogeneous paste clay. The shaping is done by pressing between two aluminum molds cylindrical format (10.1 cm diameter) with perfectly smooth walls, which allows applying a controlled force at the shaping (**Figure 2**).

3. Results and Discussions

3.1. Physical Properties

The physical characteristics of the five samples are showed in Figure 3 and Table 4. Several studies carried out on refractory clay materials [1-3,8,9] showed that the sintered brick characteristics are influenced by the mineralogical composition of clays and the firing temperature. The firing temperature of brick made using the clavs also influences properties, with higher temperatures enhancing densification by the formation of a liquid phase. The liquid phase is formed by reactions that occur in the presence of free silica, alumina and alkaline, in accordance with the ternary SiO₂-Al₂O₃-R₂O system [2]. The used clays are known for their high plasticity, which allows the introduction of a considerable quantity of silica sand to reduce the heating shrinkage and to ensure dimensional stability during thermal treatments. The incorporation of red clay in the mixture enables adjustment of the physical properties of the fired material.

Refractoriness of a material is strongly dependent on microstructural parameters such as grain size and shape, the relative volume of solid and vitreous phases, the viscosity of the vitreous phase, and sample porosity [2]. Refractoriness can be evaluated by examining the evolution of the morphology of test samples with increasing temperature under their own weight (without any other load). Refractoriness in this paper is defined as the temperature corresponding to the moment when a material begins to lose its shape (melting point). The samples tested here maintained their shape without undergoing any deformation up to 1600°C. They retain their original characteristics: no cracks, no deformation, no color change, especially a very low shrinkage appears at the end of the thermal cycle.

Table 4 gives the values of the flexural strength of the samples. The briquette M2 and M3 significantly more silica-alumina, have a greater flexural strength compared with others. The substitution of 60% of pure alumina



Figure 2. Image of the used mold and elaborated refractory crucibles.

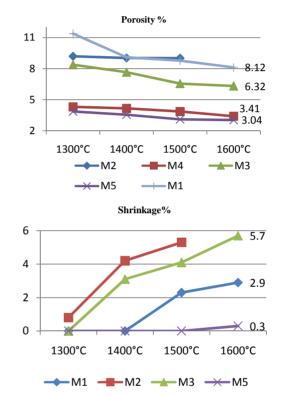


Figure 3. Evolution of porosity and shrinkage from 1300°C to 1600°C.

Table 4. Flexural strength of samples fired at 1500°C.

Samples	Flexural strength (MPa)
M1	66.54
M2	71.41
M3	70.36
M4	41.82
M5	55.87

powder (P_A) in the briquette M4 by the recovery alumina (A_R) in the mixture M5 allows increasing the flexural strength of 41.82 to 55.87 MPa.

3.2. Mineralogical Characteristics

The main crystalline phases identified by XRD after samples were fired at 1600°C were mullite, Corundum and cristobalite (**Figure 4**). Spectra show the presence of corundum in the mixtures M1 and M3 (presence of pure alumina powder in the starting mixtures). This phase disappears in the mixture M2. The characteristic diffraction peaks of cristobalite are clear and persist up to 1600°C.

The SEM micrograph of **Figure 5** clearly shows the mullite phase identified by its needle like shape. The briquette M2 shows a homogeneous microstructure and

especially richness in very fine needles of mullite. These images show the number and shape of the crystals of mullite synthesized according to the nature of the sample. An EDX analysis shows that the mullite doesn't contain foreign inclusions (**Figure 6**).

The length of mullite crystals was measured by a method of image analysis of scanning electron micros-

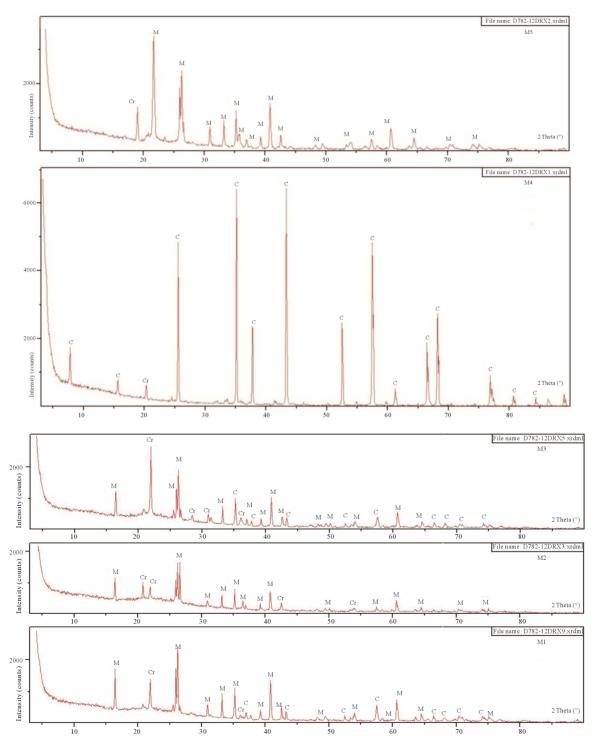
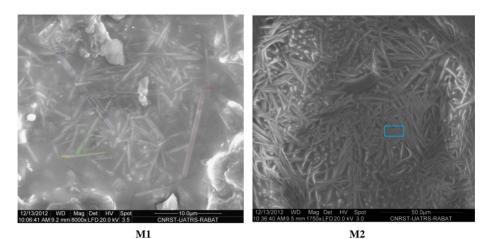


Figure 4. XRD samples of M1, M2, M3, M4 and M5 fired at 1600°C (M: Mullite, C: Corundum, Cr: Cristobalite).



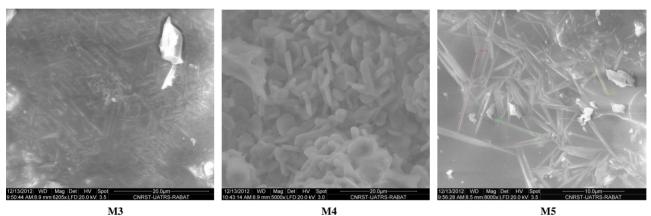
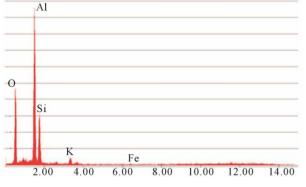
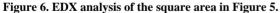


Figure 5. SEM microphotography of samples fired at 1600°C.





copy (**Figure 7**). However, there are variations in the size of the mullite to the same sample, which is due to local fluctuations of composition [16]. **Figure 7** shows that the length distributions are relatively homogeneous and tightened around an average value for the two samples M1 and M2. There was an increase in the length of mullite for sample M2 [20 - 25 microns] compared to others. Simultaneously, the average width remains constant for the three samples (0.5 to 1 μ m). This confirms the existence of a growth phenomenon preferred in the direction [001] [16].

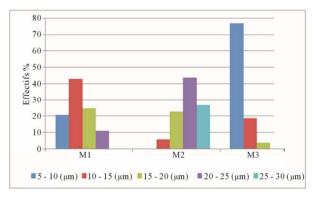


Figure 7. Average length of mullite crystal.

3.3. Thermal Behavior

As shown in **Figure 8**, the obtained curves are typical of materials undergoing microstructural transformations inducing nonlinear and irreversible effects. Their forms presented above are typical of those obtained with materials containing clay minerals [17].

Sintering of aluminosilicate phases which causes shrinkage in the matrix is more important for M3 as M1 and M2. This can be explained by the content of alumina in the initial mixtures. Indeed, the material M3 has the

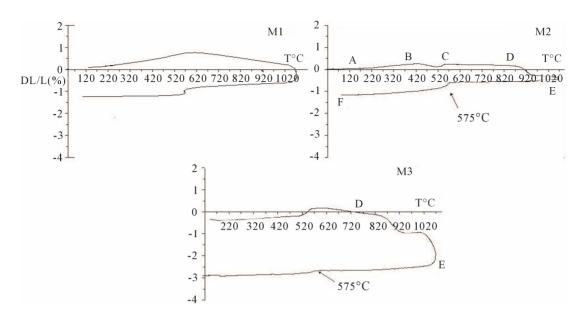


Figure 8. Thermal expansion measured from room temperature to 1100°C.

lowest quantity of alumina (30%) and thus undergoes the largest shrinkage (3% at 1100°C). The total shrinkage amplitude for M1 and M2 is low: about 1.2% at 1100°C. During cooling, the shrinkage is linear up to room temperature. The reversible transformation of quartz $\alpha \rightarrow \beta$ for the three materials is observed at 575°C. For samples M4 and M5, these two materials have given respectively 0.0% and 0.3% of thermal shrinkage at 1600°C. They both do not undergo large dimensional changes due to their high alumina content, unlike mixtures M1, M2 and M3 which have varying amounts in geomaterials susceptible of high dimensional variations.

3.4. Resistance towards Acid Attack

The resistance towards sulfuric acid was checked according to NF EN 993-16 standards dealing with dense shaped refractory products [18]. First, the refractory material was ground to a mesh size comprised between 63 and 80 μ m, and then dried. A mass m_i of this material was then treated in a boiling solution of concentrated sulfuric acid (18 N). The slurry was rinsed, filtered, dried and the obtained powder was weighted (final mass m_f). The relative mass reduction after acid treatment is reported in **Table 5**.

Table 5 presents a comparison of the acid resistance of elaborated products. One of the three elaborated refractories M2, which is in fact, the richest one of mullite (40% of A_R in the starting materials), proves to be the most efficient one. It exhibited better resistance against acid compared to M1 and M3.

4. Conclusions

Silica-alumina refractory materials made from such as

Table 5. The relative mass reduction for samples.

Samples	The relative mass reduction (%)		
M1	2.3		
M2	1.2		
M3	2.9		
M4	0.5		
M5	0.2		

kaolin clays, red clay and silica sand are widely used in the manufacture of ceramic products (as kiln furniture and bricks). Because the use of these materials is limited at high temperature, knowledge of their thermo-mechanical and deformation properties in fired refractory materials is necessary. Thus, mechanical tests related to flexural strength were carried at 1500°C, as well as a study of microstructure changes occurring in these compositions. From these studies, the following conclusions were drawn:

- The refractory materials contain mullite, corundum and cristobalite
- Very good dimensional stability is obtained upon firing
- Good mechanical characteristics and chemical resistance to acid are obtained
- Dilatometric tests show a slight and regular variation of the thermal shrinkage/expansion between 100°C and 1100°C, which helps to provide a good thermal shock resistance to the material
- Alumina powder A_R obtained from waste of silica-alumina refractory bricks proves to be efficient for an eventual use.

On the other hand, the results of this study have a double economic interest: it contributes to the valorization of

Moroccan geomaterials and leads to the local synthesis of refractories, where the national need is mainly imported from abroad.

5. Acknowledgements

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