

# Electronic Structure of Single-Crystal $\text{CaF}_2(111)$ with Nanoscale Phases of Ca and Si

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## Abstract

The impact of Ca and Si nano-scale structures on parameters and density of states of single-crystalline  $\text{CaF}_2(111)$  was studied. It was shown that at low concentration of ions of  $\text{Ar}^+$  ( $D \leq 5 \times 10^{15} \text{ cm}^{-2}$ ) one witnesses formation of nanoscale phases on  $\text{CaF}_2$  surface. It was revealed that these phases lead to narrowing of the forbidden band  $E_g$  between the phases by 4 - 4.5 eV. At higher concentrations ( $D \approx 6 \times 10^{16} \text{ cm}^{-2}$ ) the surface completely is covered by Ca atoms. It was shown that deposition of  $\theta = 10$  thick Si single layer on  $\text{CaF}_2$  surface manifests island picture. The concentration of Ca and Si nano-scale phases on the surface of  $\text{CaF}_2$  and the band gap of the phases were investigated as a function of  $(h\nu)$  of passing light. Nano-scale phases and nano-scale films of Ca were obtained by using the technique of bombardment with ions of  $\text{Ar}^+$  of  $\text{CaF}_2$  surface. Formation of nano-scale phases were accompanied by change in the composition and structure of  $\text{CaF}_2$  zones located between the phases. These changes led to narrowing of the forbidden band of  $\text{CaF}_2$  down to 7.5 - 8 eV. The concentration of Ca and Si nano-scale phases on the surface of  $\text{CaF}_2$  and the band gap of the phases were investigated as a function of  $(h\nu)$  of passing light.

## Keywords

Nano-Scale, Band Gap, Hetero-Structures, Surface, Bombardment

## 1. Introduction

The creation of electronic devices of new generation (microwave nanotransistors, ultra-large integrated circuits, optical resistors, solar cells, etc.) is mainly determined by the production of new materials (nanocrystals, nanofilms, nanomaterials) and nano-layered structures with desired physical properties. In this

regard, one of the main problems of modern nanoelectronics is to obtain homogeneous silicon spatial quantum-dimensional structures on the surface of semiconductors and dielectric films.

Special significance is the phenomenon of self-organized formation of nanostructures (islands)—*i.e.* spontaneous formation of a large number of nanostructures due to the formation of the type “matrix-adsorbed atom” system itself. Such structures can be obtained by deposition of atoms of various elements on the surface of special substrates. However, the size of these islands and the distance between them are random.

By creating certain conditions, it is possible to obtain regularly spaced and equally sized nanostructures of high stability. In particular, such magic clusters were obtained in [1] on a reconstructed ( $7 \times 7$ ) surface of atomically pure Si(111) by sputtering  $\sim 0.3$  aluminum monolayer at  $T = 550^\circ\text{C}$  under ultrahigh vacuum conditions.

In many cases, specially created defects or a reconstructed surface of a single crystal can be used as ordered nuclei. Our preliminary studies showed [2] that such defects can be created by the technique of low-energy ion bombardment in combination with annealing.

Over recently the composition, structure and electronic properties of  $\text{CoSi}_2/\text{Si}$ ,  $\text{Si}/\text{CaF}_2$ ,  $\text{CaF}_2/\text{Si}$ ,  $\text{SiO}_2/\text{Si}$  nanoscale hetero-structures have been investigated by several researchers [3]-[8]. The above structures might be used in the fabrication of Metal-Oxide-Semiconductor (MOS) and Semiconductor-Insulator-Semiconductor (SIS) structures, barrier layers, and contacts for various devices [8] [9]. However, little or practically no significant research was done aimed at studying physical properties of surface of dielectric samples with embedded nanoscale crystals of metals and semiconductors.

This paper is devoted to the study of the composition, structure and electronic properties of regularly located nanocrystalline phases and homogeneous 20 - 30 Å ( $\theta = 8 - 10$  monolayers) thick Ca and Si films of single-crystal  $\text{CaF}_2(111)$  samples created on the surface with the consistent use of ion bombardment deposition.

The paper reports the study of the effect of formation of nanoscale phases and films of Ca and Si on the composition and electronic structure of the surface of single-crystalline samples of  $\text{CaF}_2(111)$ . Over recently the composition, structure and electronic properties of  $\text{CoSi}_2/\text{Si}$ ,  $\text{Si}/\text{CaF}_2$ ,  $\text{CaF}_2/\text{Si}$ ,  $\text{SiO}_2/\text{Si}$  nanoscale hetero-structures have been investigated by several researchers [1]-[6]. The above structures might be used in the fabrication of Metal-Oxide-Semiconductor (MOS) and Semiconductor-Insulator-Semiconductor (SIS) structures, barrier layers, and contacts for various devices [6] [7]. However, little or practically no significant research was done aimed at studying physical properties of surface of dielectric samples with embedded nanoscale crystals of metals and semiconductors.

The paper reports the study of the effect of formation of nanoscale phases and

films of Ca and Si on the composition and electronic structure of the surface of single-crystalline samples of  $\text{CaF}_2(111)$ .

## 2. Experimental Procedure

Nano-scale phase of Ca was obtained in ultrahigh vacuum conditions by attracting the technique of bombardment of  $\text{CaF}_2(111)$  surface by ions of  $\text{Ar}^+$  in combination with thermal heating, whereas nano-scale phases of Si were obtained by Si deposition on the surface of  $\text{CaF}_2$ . Composition and electronic properties of the structures were investigated.

Composition and electronic properties of the structures were investigated by Auger Electron Spectroscopy (AES), Ultraviolet Photoelectron Spectroscopy (UVPS), Backscattering Spectrometry (BS) and by studying intensity  $I$  of passing light through samples as a function of energy. Photon energies  $h\nu$  varied in the range 0.6 - 6.0 eV ( $\lambda \approx 2000 - 190$  nm). Distribution of atoms across the depth was investigated by AES in combination with etching of the surface by  $\text{Ar}^+$  ions.

## 3. Results and Discussion

Photoelectron energy distribution curve (EDC) for  $\text{CaF}_2$  which was exposed to ions beam of  $\text{Ar}^+$  with  $E_0 = 1$  keV with various doses is shown in **Figure 1**. Photoelectron spectroscopy results were obtained at  $h\nu \approx 21, 2$  eV. It can be seen that on the spectrum line of “pure”  $\text{CaF}_2$  the initial sharp increase in the concentration of photoelectron starts at 12.2 - 12.3 eV, which is mainly caused by release of electrons from the valence band EV onto vacuum.

The area under the curve of the energy distribution is proportional to the quantum yield of photoelectrons. Extrapolation of this section of the curve against the  $E_{\text{phe}}$  axis gives values  $\sim 12, 1$  eV where  $E_{\text{phe}}$  is energy of photoelectrons. For  $\text{CaF}_2$  the values of electron affinity (the width of the conduction band) is approximately  $\chi \sim 0.1$  eV, therefore it can be assumed that the width of its band gap  $E_g$  is  $\sim 12$  eV. In the initial nearby section of the spectrum at energies of photoelectrons of 4 and 7.5 eV slightly intensive peaks are witnessed. The peak at  $E_{\text{phe}} \approx 7.5$  eV might be due to the presence in lattice sites of a certain concentration of Ca that does form bonds with fluorine atoms, whereas the occurrence of the peak at  $h\nu = 4$  eV might presumably be attributed to the presence of surface states.

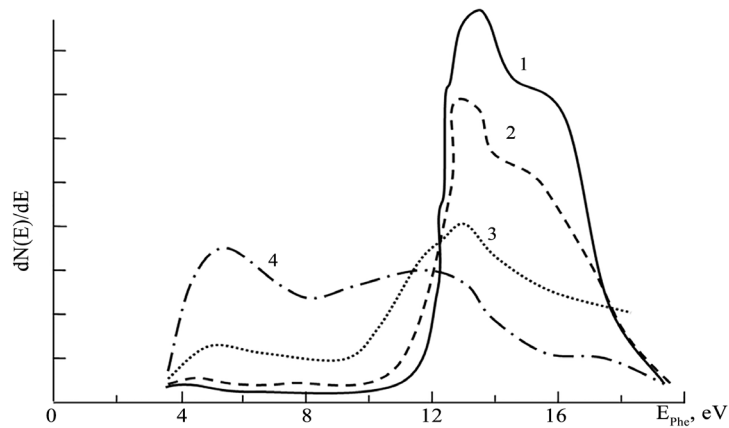
The bombardment of  $\text{CaF}_2$  by  $\text{Ar}^+$  ions as a function of the dose of ions leads to the change in the composition and electronic structure of its surface layers. Precedent to dose levels  $D = 10^{13} \text{ cm}^{-2}$  there is no noticeable change in the structure of the curve  $I(h\nu)$ .

Increasing dose to  $D = 5 \times 10^{14} \text{ cm}^{-2}$  leads to broadening of EDC, decrease of the intensity of the main peak ( $E_{\text{phe}} \approx 14$  eV) and the quantum yield of photoelectrons, as well as causes the displacement of the beginning of spectrum (the edge of the valence band EV) towards lower energy levels. Meanwhile, amplitude of the peak at  $E_{\text{phe}} \approx 7.5$  eV is slightly increased, and especially in the

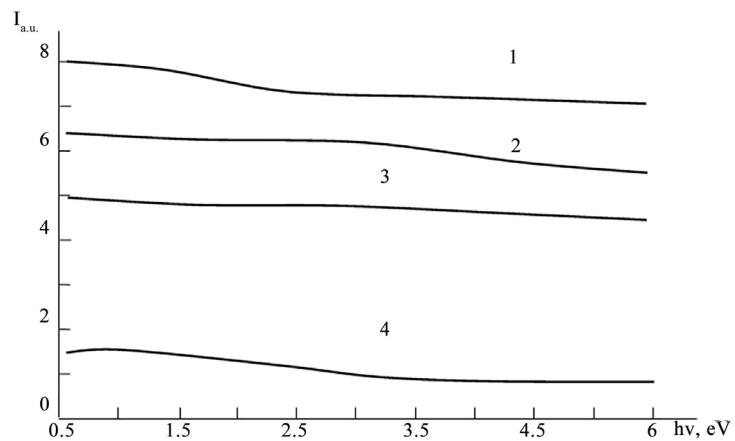
range of  $\sim 4$  eV is smoothed.

Extrapolation of the starting part of the curve against the  $E_{phc}$  axis gives values  $\sim 10.8$  eV; *i.e.*, the width of the forbidden band of  $\text{CaF}_2$  decreases by 1.1 - 1.2 eV. At  $D = 5 \times 10^{15} \text{ cm}^{-2}$  the area under the curve in the main peak section reduces roughly threefold, the quantum yield of photoelectrons reduces more than twofold, and the value of  $E_g$  by 4.5 eV. Meanwhile, intensity of peaks in the range of 7.5 - 8 eV (usually characteristic of Ca) increases more than 6 - 7 times. At  $D = 5 \times 10^{16} \text{ cm}^{-2}$  the photoelectron spectrum structure characteristic of the dielectric  $\text{CaF}_2$  completely transforms to the structure usually characteristic of the "metal-calcium" structure.

**Figure 2** shows the dependence of the passing light intensity on photon energy in the range of 0.2 - 6 eV for single crystalline sample of  $\text{CaF}_2(111)$  bombarded with  $\text{Ar}^+$  ions with energy of  $E_0 = 1$  keV with various doses. It can be seen that in case of both "pure" and ion-irradiated  $\text{CaF}_2$ , the energy  $h\nu$  increases from 0.2 eV to 6 eV and the intensity  $I$  slowly decreases.



**Figure 1.** Photoelectron spectra of  $\text{CaF}_2(111)$  exposed to the beam of  $\text{Ar}^+$  ions with  $E_0 = 1$  keV at  $D, \text{ cm}^{-2}$  1—0, 2— $5 \times 10^{14}$ , 3— $5 \times 10^{15}$ , 4— $5 \times 10^{16}$ .



**Figure 2.** Intensity  $I$  of the passing light as a function of energy of photoelectrons for  $\text{CaF}_2$  exposed to the beam of  $\text{Ar}^+$  ions with  $E_0 = 1$  keV at  $D, \text{ cm}^{-2}$  1—0, 2— $5 \times 10^{14}$ , 3— $10^{15}$ , 4— $5 \times 10^{15}$ .

It should be noted that the value of intensity  $I$  of ion irradiated  $\text{CaF}_2$  in the whole investigated diapason of  $h\nu$  is less than the intensity  $I$  of “pure”  $\text{CaF}_2$ . At  $D = 5 \times 10^{14} \text{ cm}^{-2}$ , the intensity  $I$  decreases on average 20% - 25%, while at  $D = 10^{15} \text{ cm}^{-2}$  the average value of  $I$  reduces by 35% - 40%, and at  $D = 5 \times 10^{15} \text{ cm}^{-2}$  light intensity decreases 5 - 6 times. Following the bombardment with dose of  $D = 5 \times 10^{16} \text{ cm}^{-2}$  the light practically does not pass through  $\text{CaF}_2$  films in the investigated area of  $h\nu = 0.6 - 6 \text{ eV}$ .

The authors had previously shown [10] that at low doses of  $\text{Ar}^+$ , separate cluster phases enriched with Ca atoms shape on the surface of  $\text{CaF}_2$ . As the dose of ions increases, so the dimension of these phases does indeed and at  $D = 10^{15} \text{ cm}^{-2}$  their size reach  $\sim 30 - 40 \text{ nm}$ . At  $D > 10^{16} \text{ cm}^{-2}$  overlapping of boundaries of individual phases takes place and the entire surface will be covered with atoms of Ca  $d \sim 10 - 15 \text{ \AA}$  thick.

Therefore, we assume that the change in UVPS curve's structure, reduction in the quantum yield of photoelectrons and in intensity of the passing light as the dose of ions boost, might be due to the increase in the size of cluster phases of Ca (Figure 1). Formation of these phases is allegedly accompanied by certain increase in the concentration of Ca as well as in surface layers  $\text{CaF}_2$  not exposed to bombardment.

This in turn leads to the increase in the intensity of the peak of Ca at  $h\nu = 7 - 7.5 \text{ eV}$  displacement of starting part of the EDC of  $\text{CaF}_2$  toward lower values of  $E_{\text{phe}}$ . These changes are associated with the formation of various defect states near the bottom of the conduction band and valence band. At  $D \geq 5 \times 10^{15} \text{ cm}^{-2}$  the concentration of these states increases dramatically and narrow impurity bands occur which merge with the conduction bands and valence bands.

Consequently, the band gap decreases. In particular, at  $D = 5 \times 10^{15} \text{ cm}^{-2}$  the intensity of passing light was about 70% - 80%, and the value of  $E_g \sim 7.5 \text{ eV}$ . It can be assumed that roughly 80% of the surface of  $\text{CaF}_2$  is covered with Ca atoms while the areas of  $\text{CaF}_2$  previously not exposed to bombardment form impurity bands with width of  $\sim 4 - 4.5 \text{ eV}$ . Starting from  $D = 4 \times 10^{16} \text{ cm}^{-2}$  light hardly passes through the  $\text{CaF}_2$ , *i.e.* the surface turns out to be completely covered with atoms of calcium.

Similar investigations (change in composition, change in the crystal and electronic structure) of Si nano-scale films grown on  $\text{CaF}_2(111)$  surface by molecular beam epitaxy were done. In order to deposit Si epitaxial films, the system after each cycle of deposition was annealed at  $T \approx 800 - 900 \text{ K}$ . Down to the depth of  $\theta \approx 10$  monolayers, the Si film had uneven character.

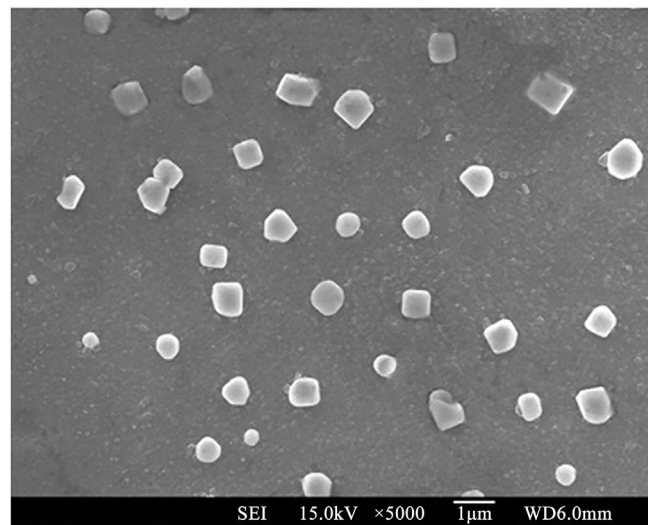
Solid homogeneous films formed down to  $\theta \geq 10 - 15$  monolayers. As a reference in Figure 3 we placed micrograph of the surface of  $\text{CaF}_2(111)$  with Si film thickness of  $\theta \approx 5$  Si monolayers. It can be seen that the Si film is uneven. Dimensions of islands are within 500 - 1000 nm. Figure 4 shows the  $I(h\nu)$  curve for  $\text{CaF}_2$  film with film thickness of  $\theta = 5$  and 15 monolayers.

It is seen that in the case of film of  $\theta = 5$  monolayers thick the light intensity

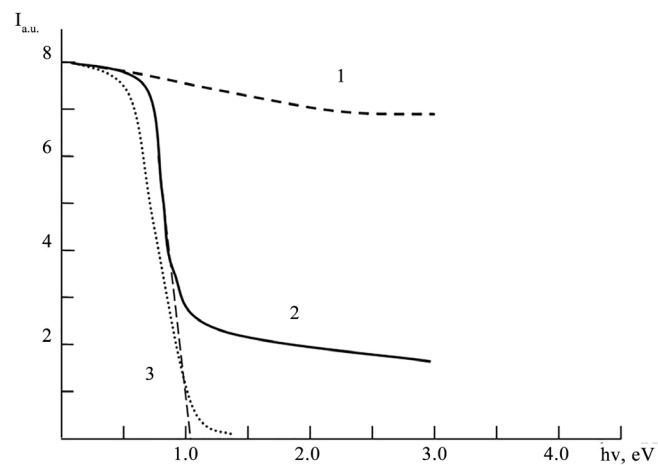
in the range of  $h\nu = 0.9 - 1.2$  eV decreases 3, 5 - 4 times (from 7.5 - 8 to 2 - 2.2). It is believed that 70% - 75% of the surface of  $\text{CaF}_2$  is covered with Si film. In the case of films of  $\theta = 15$  monolayers thick, in the range of  $h\nu = 0.8 - 1.1$  eV the intensity  $I$  decreases from 7.5 - 8 to virtually zero, *i.e.* the surface appears to be completely covered by Si atoms.

#### 4. Summary and Conclusions

One can believe that as a result of bombardment of  $\text{CaF}_2$  with  $\text{Ar}^+$  ions and depending on irradiation dose one can witness change in the electronic structure of the surface layers of ingot samples which is explained by the formation of nanocluster phases of Ca in the exposed areas of  $\text{CaF}_2$ , as well as by changes in the composition and structure of interphase (non-irradiated) areas.



**Figure 3.** SEM images of  $\text{CaF}_2(111)$  surface with a film thickness of Si  $\theta \approx 5$  monolayers.



**Figure 4.** Intensity of passing light as a function of photon energy for “pure”  $\text{CaF}_2(111)$  (curve 1) and  $\text{CaF}_2$  with a film thickness of 5 Si monolayers (curve 2) and 15 monolayers (curve 3).

These defects at  $D \geq 5 \times 10^{15} \text{ cm}^{-2}$  leads to the formation of impurity bands near the ceiling of the valence band and the bottom of conduction band of these areas of  $\text{CaF}_2$ ; therefore  $E_g$  decreases by 4 - 4.5 eV. In the course of molecular beam epitaxy, and the growth of Si films on the surface of  $\text{CaF}_2$  down to  $\theta \approx 10$  monolayer thick, they had island-like character. Judging by the dependence curve and trend of intensity of the passing light ( $I$ ) on the photon energy ( $h\nu$ ), we have been able to determine density of coating on the surface of the  $\text{CaF}_2$  consisting of Si films as well as defining the  $E_g$  of Si islands. In particular, at  $\theta \approx 5$  monolayers the degree of coverage was 70% - 75%, and the silicon  $E_g \sim 1.1 \text{ eV}$ .

### Conflicts of Interest

The authors declare no conflicts of interest regarding the publication of this paper.

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