

Effect of Hot Dip Galvanized Steel Surface Chemistry and Morphology on Titanium Hexafluoride Pretreatment

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Abstract

Titanium hexafluoride pretreatments are known to improve paint adhesion and function as a barrier between the coating and the hot dip galvanized (HDG) steel surface. Interactions at the zinc/pretreatment interface are of utmost importance for the formation of pretreatment layers and the corrosion resistance of color coated hot dip galvanized steels. Removal rate of inert aluminum oxide from HDG steel samples by chemical dissolution was studied. XPS measurements showed that the surface Al_2O_3 layer thickness decreased rapidly already at mild alkaline cleaning, while complete removal of Al required severe etching. Low reactivity of an Al_2O_3 -rich surface was confirmed by impaired formation of a titanium hexafluoride pretreatment layer. Grain boundaries and deformation twinings were shown to be of importance for the reactivity of the HDG surface and for the precipitation of the pretreatment chemical. Helium ion microscopy images and electron probe microanalysis (EPMA) of a pretreated sample showed accumulation of the pretreatment chemical at the grain boundaries. Al removal rate was fast at the deformation twinings at the grain plateaus. Slow Al removal was observed at dendritic valleys and grain boundaries. The results increase understanding of the reactivity of hot dip galvanized steel surface.

Keywords

Hot Dip Galvanized Steel, Aluminum Oxide, Pretreatment, Surface Morphology, Helium Ion Microscopy

1. Introduction

Hot dip galvanized (HDG) zinc coatings are known for good formability and

high corrosion resistance. HDG steel is coated in a continuous color coating line in order to improve the corrosion protection of the final product, provide desired functionality and for aesthetical reasons. Color coated HDG steel has wide range of applications from household appliances to construction industry. The various applications set different demands for the performance of the coated steel products. The degradation reactions (due to physical, chemical or biological degradation, oxidation or photo-oxidation) of polymer coatings are fairly well known [1], and polymer matrixes can be tailored according to the requirements of the end use environment. The corrosion properties of color coated steel are regulated by incorporating corrosion inhibitors into the coatings. The performance of different inhibitors has also been widely studied and the non toxic inhibitors have now mostly replaced the chromates [2] [3] [4] [5].

Coating adhesion on pretreated HDG surfaces depends mainly on physical bonds (dispersive and electrostatic forces) and chemical bonds (covalent, ionic, hydrogen and polar) [6]. In order to improve metal/coating adhesion, zinc coated steel is pretreated before color coating. Although the pretreatment layers are significantly thinner than the organic coatings, they are of crucial importance for the coating adhesion as well as for the corrosion resistance [7] [8] [9] [10] [11]. The corrosion resistance is strongly affected by the electrolytic environment that prevails at the metal/polymer interface [12] [13]. The term “barrier effect” is often used to describe the overall performance of the Cr-free pretreatments, indicating that the modern Cr-free pretreatments lack the active corrosion resistance typical for chromates. Pretreatment solutions based on titanium or zirconium hexafluorides are widely used in the color coating industry for hot dip galvanized steel. The uniformity of a titanium hexafluoride pretreatment layer on the HDG surface has been shown to correlate with the humidity resistance of coated panels [14].

It remains a challenge to achieve the functional performance of chromate pretreatments with Cr-free pretreatments, which increases the importance of better understanding of the molecular level interactions between the metal coating applied by hot dip galvanizing and the Cr-free pretreatments [15]. The effect of surface chemistry of zinc coatings to the performance of thin passivation and pretreatment layers has been to some extent discussed in earlier studies [7] [15] [16]. The surface topography and its effect on coating adhesion have also been addressed, but the studies have mostly concentrated on the effect of temper rolling [6].

In this study, the chemistry and morphology of HDG surfaces, as well as their effects on the alkaline cleaning and titanium hexafluoride pretreatment were evaluated.

2. Materials and Methods

Non temper rolled hot dip galvanized steel (~0.2% Al in the zinc bath) with a total coating mass of 275 g Zn/m² was obtained from SSAB Europe Oy. The HDG steel was obtained without protective oil. The laboratory pretreatments

were carried out as follows. The alkaline cleaning solution was prepared in deionized water from powder (Gardoclean 338, Chemetall). The prepared solutions and the corresponding pH values of the solutions are shown in **Table 1**. The cleaning was performed by dipping the panels in a bath at 60°C for 4 s, followed by two times of rinsing with deionized water (60°C for 4 s) in separate baths. The cleaning procedure was repeated two times. The pretreatment chemical was applied by dipping the panels into the pretreatment chemical bath (15% chemical concentration) for 4 s, followed by removal of excess liquid by pressing with a Teflon rolling pin on a magnetic chuck. Finally, the panels were oven dried at 60°C for 5 minutes.

A mildly acidic pretreatment chemical (Bonderite M-NT 1455T, Henkel) containing titanium hexafluoride, manganese phosphate and organic polymer was used [12] [17]. The chemical was characterized by XPS (X-ray photoelectron spectroscopy). For the measurement, a microscopy glass slide was submerged in a pretreatment chemical solution sample for 30 s. The XPS analysis was performed after air drying of the slide. An XPS spectrum is shown in **Figure 1** and the atomic-% concentrations of different elements are shown in **Table 2**.

A CG842 instrument (Schott, Germany) was utilized for measurement of pH. Element analysis and depth profiling was performed using X-ray photoelectron

Table 1. Concentrations and pH values of the prepared alkaline cleaning solutions.

Gardoclean 338, g/L	pH (60°C)
0.00	6.1
0.03	9.4
0.06	9.7
0.10	9.9
0.20	10.2
0.30	10.3
0.50	10.5
1.00	10.8
2.00	11.0
4.00	11.2
8.00	11.4

Table 2. Characterization of the pretreatment chemical by XPS. Three parallel measurements were performed.

Element	Atomic-%	Stdev
F1s	30.2	0.5
O1s	26.4	0.8
C1s	15.7	1.2
Mn2p3	13.8	1.6
Ti2p	6.3	0.4
P2p	5.2	0.4
N1s	2.4	1.1

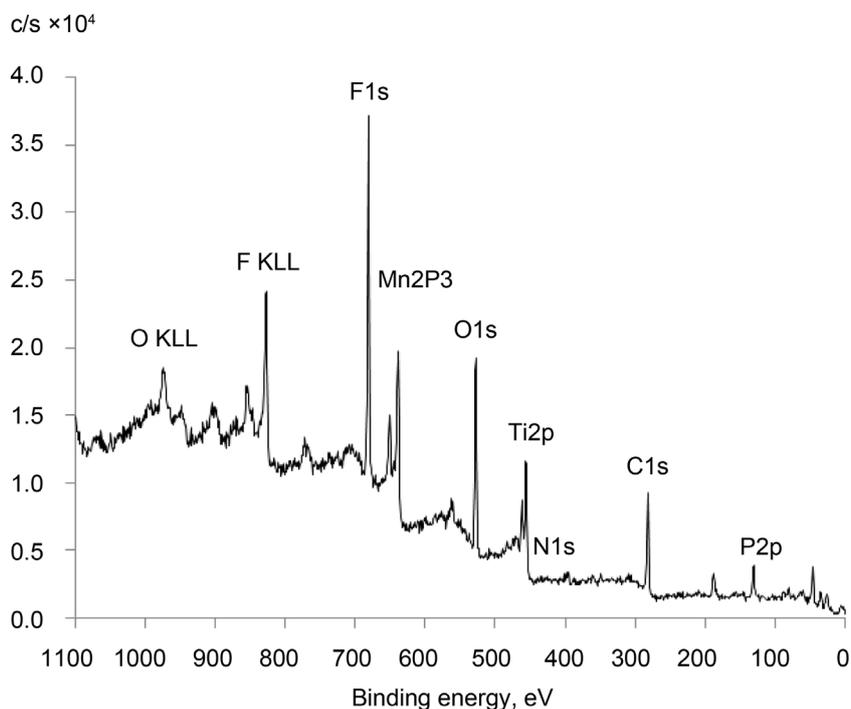


Figure 1. XPS spectrum recorded on a dried pretreatment chemical droplet on a glass slide.

spectroscopy (PHI Quantum 2000). The spectra were obtained using monochromated Al K α beam with a 200 μm spot size. The sputtering was done with Ar $^+$ -ions, and the approximative sputtering rate was calculated using a SiO $_2$ calibration standard. Before spectral acquisition, a mild ~ 1 nm sputtering was employed in order to remove the surface carbon contamination. Helium ion microscopy was performed with a Carl Zeiss Orion NanoFab microscope using 30 keV helium beam energy and 0.4 - 0.5 pA beam current. Secondary electrons were collected with an Everhart-Thornley detector at 500 V grid voltages. Working distance was ~ 9 mm and dwell time was 50 μs per pixel. No line or frame averaging was used. Element mapping was performed with a time of flight secondary ion mass spectrometer (ToF-SIMS, PHI Trift II). The measurements were carried out at 25 kV and 15 min measurement time. Mild sputtering with Ga $^+$ -ions at 15 kV and 2 nA for 250 s/mm 2 preceded the measurements. Element mapping was also carried out with an electron probe microanalyzer (EPMA) with wavelength-dispersive X-ray spectrometers (WDS) (JEOL JXA-8600). The maps were recorded at 15 kV and 25 nA.

3. Results and Discussion

3.1. Alkaline Cleaning

In a continuous coil coating process, surface impurities are removed by leading the HDG sheet through alkaline cleaning steps before the pretreatment. During the alkaline cleaning, the surface is also activated by dissolving some of the inert Al $_2$ O $_3$ [18] [19]. Removal of aluminum oxide as a function of alkaline cleaning

solution concentration is shown in **Figure 2**. On the surface of non-etched HDG surface, over 30% of the atoms detected in the outermost few nanometers were aluminum. The rest was mostly oxygen with small contributions from zinc and carbon [20]. The surface aluminum concentration decreased when the sample was treated in an alkaline solution. The Al/Zn atomic ratio decreased rapidly already at low concentrations of the cleaning solution. Similar trends have been obtained by increasing the treatment time and temperature [19]. In this study, the concentration of aluminum decreased much slower than the Al/Zn ratio. The zinc signal, derived from metallic Zn below the Al_2O_3 layer, increased rapidly as the Al_2O_3 layer thickness decreased, which explains the difference. For instance, at 2 g/L the decrease in Al atomic-% was about 50% while the decrease in Al/Zn atomic ratio was about 90%. The Al/Zn atomic ratio has been used in some studies to depict the aluminum content of a surface, since the Al atomic-% values are generally rather small [6] [19]. Also, decrease in signal intensity due to local impurities such as carbon plays a lesser role if the Al/Zn atomic ratio is considered instead of solely the Al atomic-% data. Good adhesion and corrosion resistance properties have been reported with about 70% removal of Al [21]. A complete removal of Al requires severe etching and may not be process economically feasible [7] [19].

3.2. XPS Depth Profiling of Pretreated Samples

Samples with low removal of aluminum (0.2 g/L) and high removal of aluminum (4.0 g/L) were pretreated. XPS depth profiles of the different elements within the pretreatment layers are shown in **Figure 3** and **Figure 4**. Over 60 atomic-% of the outermost surface was composed of carbon in both samples. The carbon signal decreased rapidly after mild sputtering, which confirmed that the carbon on the surface was present as a surface impurity. The true carbon signal,

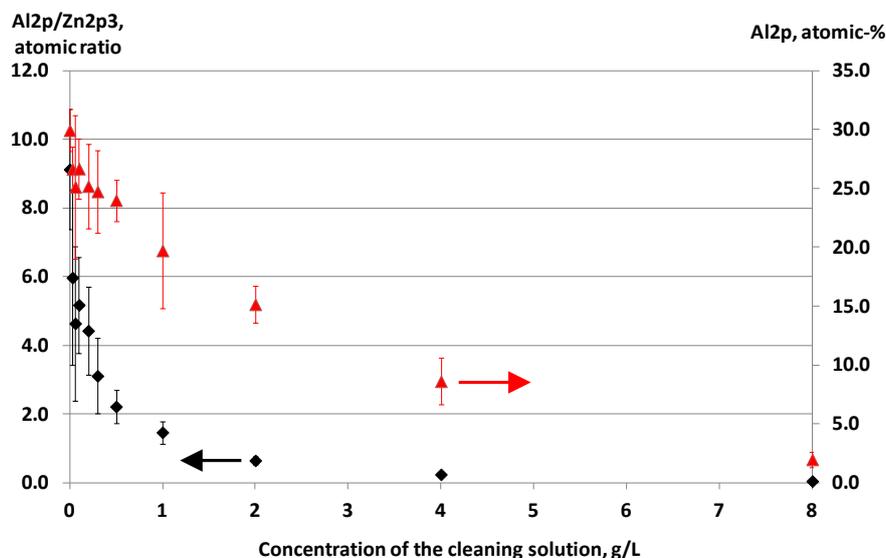


Figure 2. Dissolution of aluminum (oxide) from HDG samples as a function of the cleaning solution concentration. The results are average values of nine measurements.

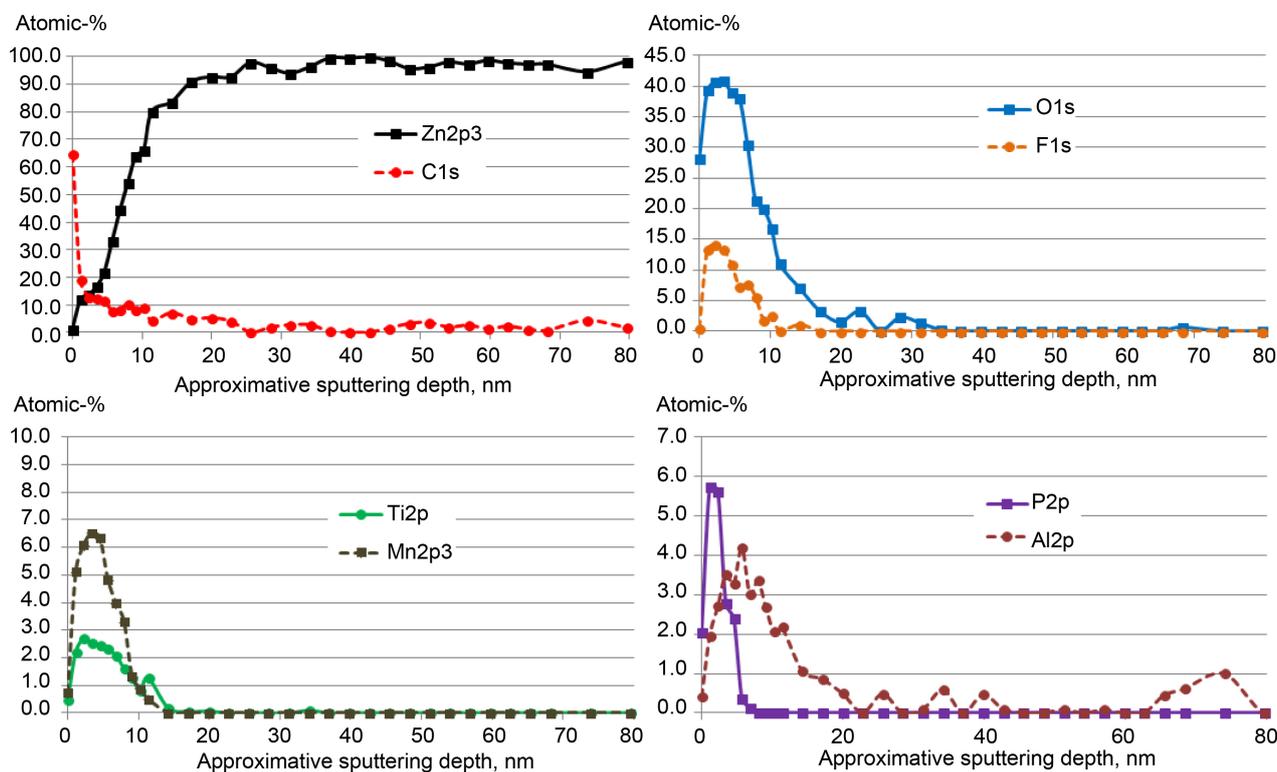


Figure 3. XPS depth profiles of the different elements in the pretreated layer on a non temper rolled hot dip galvanized steel sample (concentration of the cleaning chemical was 0.2 g/L). The results are averages of three parallel measurements.

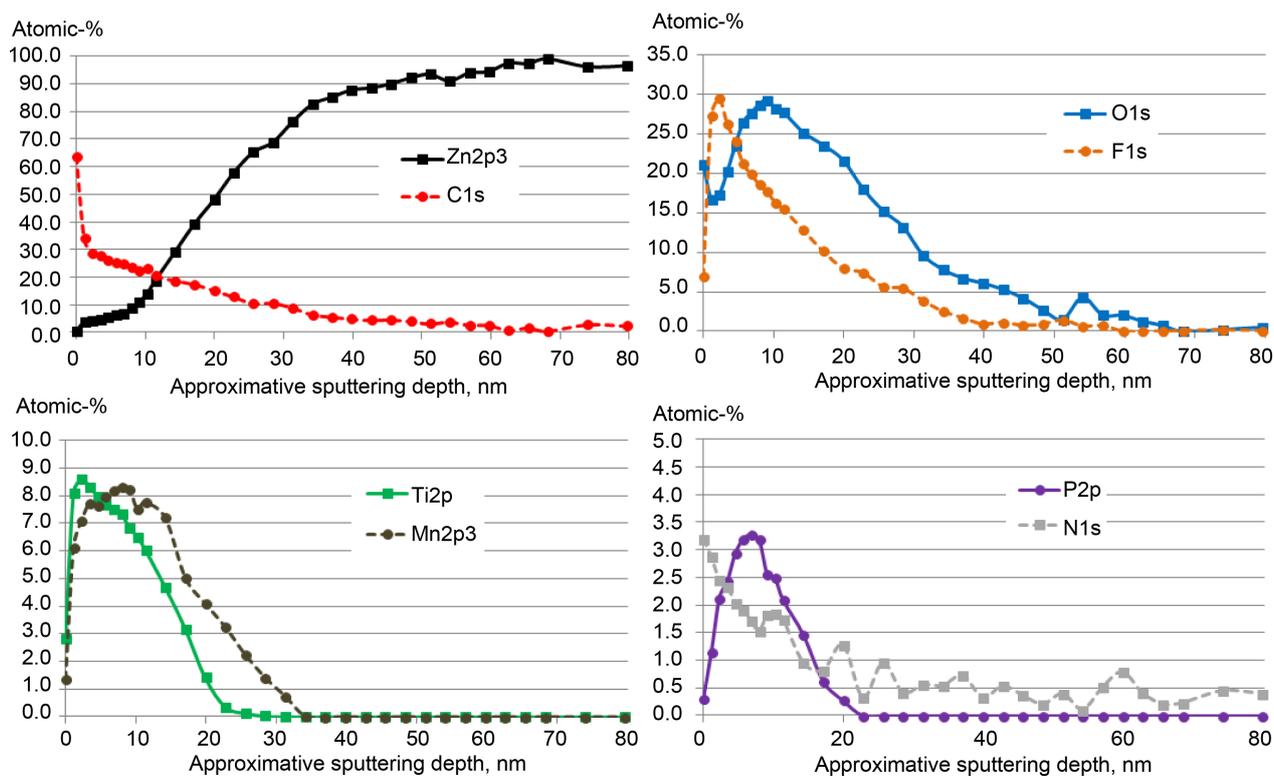


Figure 4. XPS depth profiles of the different elements in the pretreated layer on a non temper rolled hot dip galvanized steel sample (concentration of the cleaning chemical was 4.0 g/L). The results are averages of three parallel measurements. Al was not detected.

representing the polymer in the pretreatment solution (**Table 2**), decreased steadily across the pretreatment layer. Bucheit and Hughes [22] reported the polymer to be more concentrated towards the surface of the pretreatment layer. However, in a study by Le Manchet *et al.* [23] the carbon originating in the polymer was present in a significant amount across the layer, which is in accordance with the results of this study (**Figure 3** & **Figure 4**). The pretreatment layer was also rich in oxygen. The presence of oxygen has been attributed to the presence of phosphates and metal hydroxides [23]. Fluorine was prominent on the surface of the pretreatment layer and its intensity decreased very rapidly, as seen especially in **Figure 4**. Residues of fluorine were, however, also present throughout the layer. In the dried pretreatment solution the Ti:F ratio was about 1:5, close to the stoichiometry of those elements in the pretreatment solution (**Table 2**). In the pretreated samples, a rough comparison of the Ti and F depth profile areas revealed that the Ti/F atomic-% ratio was close to 1:2 or 1:3, which is much lower than the stoichiometric 1:6 ratio. This confirms that the TiF_6^{2-} complex is broken as the pretreatment chemical reacts with the zinc surface [24]. In the sample cleaned at 0.2 g/L, phosphorous prevailed more on the surface of the pretreatment layer than manganese and titanium. The small amounts of nitrogen that could be observed in the pretreatment solution (**Table 2**) and in the well cleaned sample (**Figure 4**) could originate in an amine containing organic water-soluble polymer that is added in some pretreatment chemical compositions [17].

The approximative thickness of the pretreatment layer was significantly higher in the well cleaned sample (**Figure 4**) than in the sample with mild cleaning (**Figure 3**). Also, aluminum was detected only in the latter sample. Since the pretreatment step was identical for the both samples, it can be deduced that the sample with mild cleaning was less reactive than the well cleaned sample. Stellnberger *et al.* [25] showed by electrochemical quartz crystal microbalance measurements that the reactivity of a galvanized steel sample increases with alkaline cleaning. Similar results have been obtained in other pretreatment and coating studies [6] [7] [26]. The investigations are underway to establish a critical level of surface aluminum oxide removal for good coating adhesion.

3.3. Helium Ion Microscopy of a Pretreated Sample

Helium ion microscopy images of the pretreatment layer deposited on a sample after alkaline cleaning with a 4 g/L concentration are shown in **Figure 5**. Grain boundaries have a dark appearance, which indicates strong presence of the pretreatment chemical [27]. Inside the grains there are remarkable contrast variations, which have earlier been shown to depend on the microscale thickness variations within the pretreatment layer [28]. Also deformation twinings were seen within the grains. A close-up of the pretreated surface further highlights the microscopic thickness variations in the pretreatment layer, higher concentrations seen as darker spots. Deformation twinings are seen on the surface. The surface of the twinings is composed of wavy lines, and the edges of the twinings are rather sharp. Formation of a deformation twinning is described as a

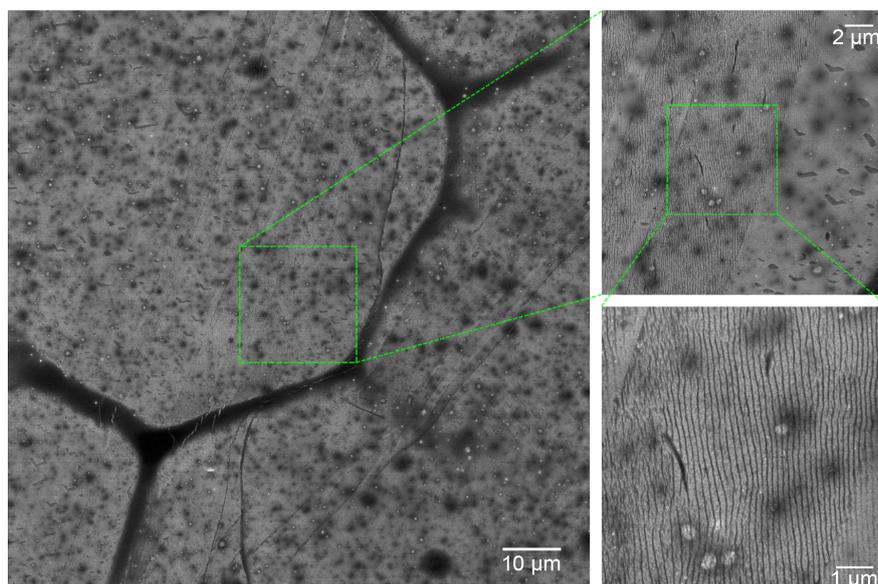


Figure 5. Helium ion microscopy images of a well cleaned (4 g/L), pretreated HDG sample.

homogeneous movement leading to macroscopic deformation accompanied by shuffling of atoms [29] [30]. The dislocations due to deformation take place at the atomic level, extending up to several atomic layers. The collective atomic movements are localized along discontinuity lines, which probably are the lines visible in the HIM images. Twinning is typically in contact with at least one grain boundary and usually extend across the grain [29]. Also, some darker ~ 1 μm wide cavities in the zinc were observed. Additionally, bright unidentified sub-micrometer sized spots with significant size variation were visible beneath the pretreatment layer.

3.4. Deformation Twinning

A variety of deformation modes can occur in zinc coatings [29] [31] [32]. At low strain levels ($<5\%$), the deformation occurs predominantly by twinning [30] [33] [34] [35]. About 1% tensile elongation takes place at the end of the zinc coating process [16] [31], causing extensive formation of deformation twinings. The effect of deformation twinings on the reactivity of the zinc surface was studied with selected samples after alkaline cleaning. At mild alkaline cleaning (0.2 g/L) the deformation twinning that was easily observed in a light microscopy image could not be distinguished in ToF-SIMS element maps (Figure 6). The surface was covered by Al (oxide) and only trace amounts of Zn were detected. At stronger alkaline cleaning, 2 g/L, a twinning was seen also in ToF-SIMS element maps (Figure 7). In general, the Al counts were decreased and the Zn counts were increased when stronger alkaline cleaning was applied. Compared with the surrounding areas, at 2 g/L the deformation twinning was depleted in Al, while similar changes were not seen in the Zn element map. Al was also prominent in the dendritic valleys crossing the deformation twinning and in the grain boundary [36] [37]. This indicates a clear chemical difference in the surface composition of

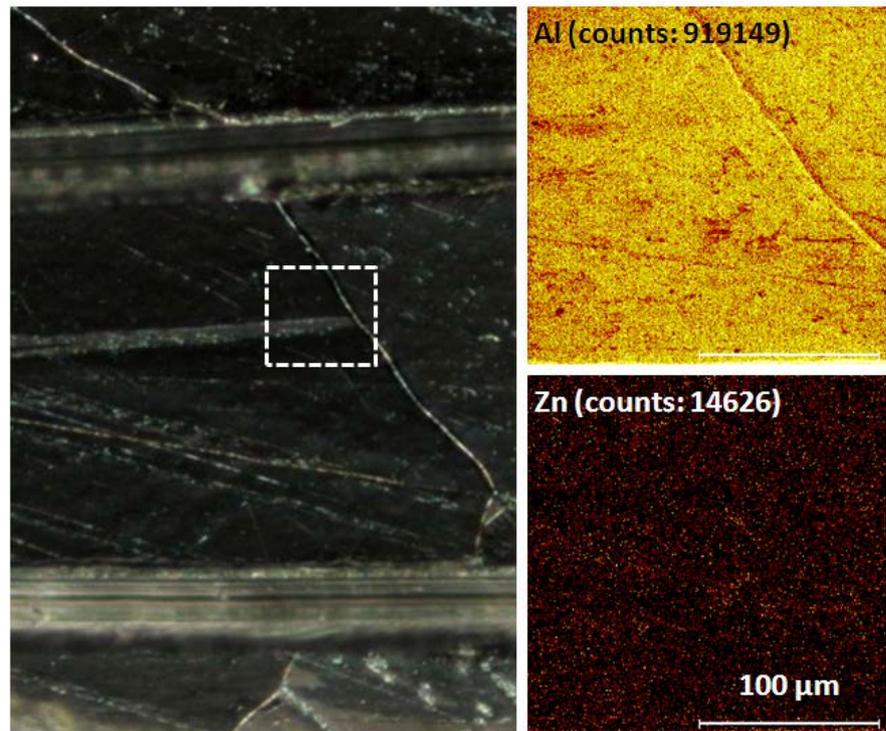


Figure 6. ToF-SIMS element maps of Al and Zn of a deformation twinning close to a grain boundary. Brighter color in the ToF-SIMS maps indicates higher intensity of the element. The sample was cleaned with a 0.2 g/L cleaning solution prior to measurement. Light microscopy was employed for choosing of the imaging spot (image on the left).

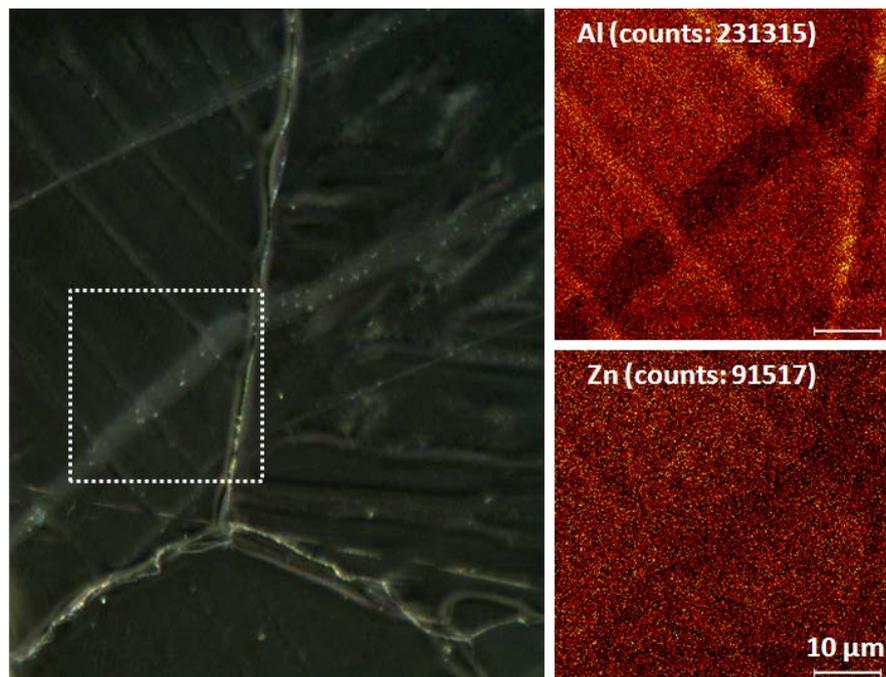


Figure 7. ToF-SIMS element maps of Al and Zn of a deformation twinning close to a grain boundary, intersected by dendrites. Brighter color in the ToF-SIMS maps indicates higher intensity of the element. The sample was cleaned with a 2 g/L cleaning solution prior to measurement. Light microscopy was employed for choosing of the imaging spot (image on the left).

the different structural details. The deformation by twinning has been proposed to cause a slight increase in surface roughness [34]. The higher reactivity of the twinning, seen in this study as stronger Al dissolution, could be due to disruption of the Al_2O_3 layer. At higher concentrations of the alkaline cleaning chemical, due to extensive removal of Al, the twinings could not be distinguished by ToF-SIMS anymore.

3.5. Accumulation of Pretreatment Chemical in Grain Boundaries

The origin of the dark appearance of the grain boundaries that was detected with HIM (Figure 5) was further studied with EPMA. Element maps were recorded at a grain boundary of the pretreated sample (cleaning at 4 g/L, followed by pretreatment) (Figure 8). Higher Ti, Mn and O intensities were measured at the grain boundary than at the grain plateaus, confirming higher pretreatment coating weight at the grain boundaries. Greater surface roughness (depressions in a zinc coating) has been proposed to increase the coating weight deposited per unit of the apparent surface area due to increased reactive area during the treatment [38]. However, on the other hand the dendritic valleys and grain boundaries are often populated with high amounts of aluminum and other impurities rejected from the molten zinc during the solidification of the zinc coating [36]. Thus, the higher coating weight of the pretreatment chemical at the grain boundaries may also be a drying effect.

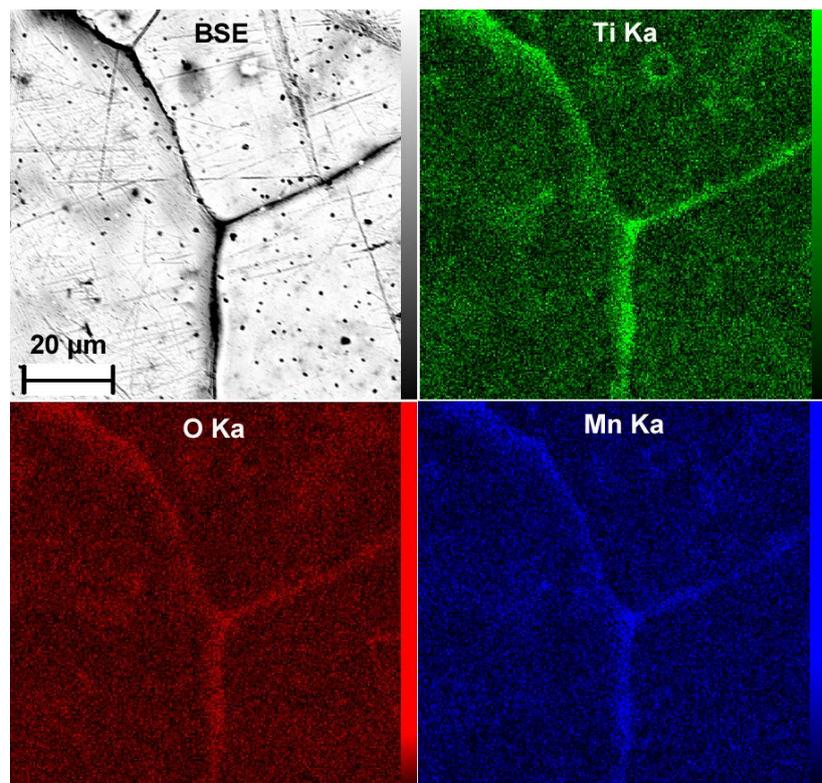


Figure 8. EPMA element maps measured on a grain boundary of a cleaned and pretreated HDG sample (sample treated with a 4 g/L cleaning solution before pretreatment). Brighter color in the maps indicates higher intensity.

4. Conclusion

Chemical composition and morphology of the outermost surface of hot dip galvanized steel strongly affect the reactivity of the surface and the formation of a titanium hexafluoride pretreatment layer. Residual Al_2O_3 is the most important factor affecting the reactivity of non-oiled HDG steel surfaces. The Al/Zn atomic ratio decreased rapidly when HDG samples were exposed to mild alkaline cleaning, because the decrease in Al_2O_3 layer thickness strongly increased the Zn signal intensity. XPS measurements showed buildup of a thicker pretreatment barrier when the Al_2O_3 was removed. The dissolution of Al_2O_3 occurred more easily at deformation twinings, which could be explained by an increased reactive surface area available for etching. On the other hand, the Al removal was slower at dendritic valleys and grain boundaries. Accumulation of the pretreatment chemical in the grain boundaries was confirmed by helium ion microscopy imaging and EPMA measurements. This could be due to higher reactivity of the grain boundaries, higher reactive area of the surface depressions, or a drying effect. The results of this study lay a foundation for our next study that will focus on evaluation of the effect of alkaline cleaning and different surface morphologies on paint adhesion and corrosion resistance of coated HDG samples.

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