Synthesis and Characterization of Copper Substituted Nickel Nano-Ferrites by Citrate-Gel Technique

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ABSTRACT

Mixed copper substituted nickel nano-ferrites having the chemical formula $Ni_{1-x}Cu_xFe_2O_4$ (where x = 0, 0.2, 0.4, 0.6, 0.8, 0.9 and 1.0) were synthesized by citrate gel technique. The crystal structure characterization and morphology were investigated by X-ray diffraction (XRD) and Scanning electron microscopy (SEM). An elemental composition of the samples were studied by energy dispersive X-ray Spectroscopy (EDS). Lattice parameter, X-ray density, Volume of the Unit Cell and The values of the hopping length for octahedral (d_B) and tetrahedral (d_A) sites were calculated. The observed results can be explained on the basis of composition.

Keywords: Ferrites; Nano-Materials; X-Ray Diffraction; SEM; EDS

1. Introduction

Nano crystalline spinel ferrites have been investigated in the recent years by many researchers. These ferrites are materials of interest because of their unique electric, dielectric and magnetic properties. These ferrites are very important group of magnetic materials due to their extensive use in a wide range of applications from low to high permeability devices including electronics, ferrofluid, magnetic drug delivery microwave devices, and high density information storage devices [1-5].

They have the general formula of MFe₂O₄ (where M (divalent metallic ion): Fe, Co, Ni, Cu, Mg, etc.,) and a unit cell contains 32 O-atoms in a cubic close packing with 8Td (tetrahedral) (A site) and 16 Oh (octahedral) (B site) occupied sites [6]. In most ferrite materials, the substituents play an important role in determining the variation of the physical properties, the magnetic and electric transport properties are affected by the substitutents [7]. The substitution can be divided into the following types: direct replacement of Fe⁺³ on tetrahedral (A) or octahedral (B) sub lattice by the substituent ion, with conesquential redistribution of Fe⁺³ ions between A- and Bsub lattices which lead to the change of ferromagnetic spin structure. The extent of iron redistribution depends on the specific nature of the substituting ion. Many investigators have carried out research work on Ni ferrite

containing various cations such as Zn and Cu [7-13]. The substitution of Cu brings about a structural phase transition accompanied by the reduction in the crystal symmetry due to cooperative Jahn-Teller effect [14,15], which ultimately results in some interesting electrical and magnetic properties.

Several investigators have focused their attention on Ni-Cu ferrite because copper containing ferrites have interesting magnetic and electrical properties [16,17]. Ni-Cu ferrites are preferred because of improved magnetic properties, high frequency response and high resistance. In the field of electronics, researchers continue to investigate the design of superstructures comprised of magnetically ordered arrays to enhance the sensitivity in magnetic sensors. The stability of electronic and high density data storage devices made by magnetic nanocomposites is limited by their super paramagnetic transition. Thus, while current efforts are aimed at increasing the thermal energy barrier against the magnetic reversal in magnetic nanocomposites, there is strong interest to understand the magnetization processes in the super paramagnetic region [18].

In the past, mixed spinel ferrites are prepared by conventional methods, which has disadvantages such as high period heating which may result in divalent volatilization and change the stoichiometry [19]. In recent years, numbers of physical and chemical techniques have been de-

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veloped to prepare nanosized materials. The chemical techniques for the synthesis of nanostructured materials offer some advantages in comparison with the physical techniques in relation to simplicity, energy saving and product homogeneity. The widely used chemical methods are electro deposition [20] sol-gel route [21], coprecipitation [22], micro emulsion technique [23], glyoxylate precursor method [24], hydrothermal method [25], reverse micelle technique [26], sucrose method [27, 28]. The citrate-gel auto combustion technique has been used for the preparation of mixed nanocrystalline spinel ferrites with specific properties, such as controlled stoichiometry and narrow particle size distribution. The low cost, simplicity and short time of production and the purity and homogeneity of final product are included among its advantages. In addition the speed and low temperature of the synthesis process prevent the sintering of the obtained powder; thus, it presents a large superficial area and a good sinterability in relation to powders obtained by other synthesis techniques [29-35].

2. Experimental Details

The mixed Ni-Cu ferrite powders having the chemical formula $Ni_{1-x}Cu_xFe_2O_4$ (where x = 0, 0.2, 0.4, 0.5, 0.6, 0.8 and 1.0) were synthesized by citrate gel auto combustion technique using the below mentioned raw materials.

2.1. Raw Materials

- Nickel Nitrate-99%Pure (AR Grade) (Ni(No₃)₂6H₂O)
- Copper Nitrate-99%Pure (AR Grade) (Cu(No₃)₂3H₂O)
- Ferric Nitrate-99%Pure (AR Grade) (Fe(No₃)₂9H₂O)
- Citric Acid-Citrate-99%Pure (GR Grade) ($C_6H_8O_7H_2O$)
- Ammonia-99% Pure(AR Grade) (NH₃)

2.2. The Flow Chart for Synthesis of Ni-Cu Ferrites

Preparation of the mixed Ni-Cu ferrite powders were synthesized by citrate gel technique using the below mentioned flowchart:



3. Results and Discussion

3.1. Structural Characterization

The structural characterization was carried out using the X-ray diffractomerter Bruker (Karlsruhe, Germany) D8 advanced system with a diffracted beam monochromatic Cu K_a radiation ($\lambda = 1.5405$ Å) radiation source between the Bragg Angles 20° to 80° in steps of 0.04°/Sec. The 2 θ vs. intensity data obtained from this experiment and plotted a graph as shown in **Figure 1**.

The X-ray diffraction pattern, crystalline phases were identified by comparison with PDF-4 reference data from the international centre for diffraction data (ICDD). All Bragg reflections have been indexed, which confirm the formation of cubic spinel structure in single phase without any impurity peak. The strongest reflection comes from the (311) plane, which denotes the spinel phase. The peaks indexed to (111), (220), (311), (400), (511) and (440) planes of a cubic unit cell, all planes are the allowed planes which indicates the formation of cubic spinel structure in single phase [36].

The crystallite size was calculated for the all the compositions using the high intensity 311 peak and using Scherrer Formula [37] while taking into account the instrumental broading [38].

$$D_{hkl} = \frac{0.91\lambda}{\beta\cos\theta}$$

where D_{hkl} is the crystalline size perpendicular to (*hkl*) pane,

 λ : the wave length of X-ray [39] used,

 β : width of diffraction peak *i.e.*, Full Width Half Maxima (FWHM)

 θ : the peak position.

The calculated values of crystallite size for the different compositions are given in **Table 1**.

It can be seen from the table that the values of the crystal size veries from 36.7 nm to 58.91 nm. Though all



Figure 1. X-ray diffraction pattern of Ni-Cu nano ferrites (with x = 0, 0.2, 0.4, 0.5, 0.6, 0.8 and 1.0).

the samples were prepared under identical condition, the crystallite size was not the same for all Cu concentrations. This was probably due to the preparation condition followed here which gave rise to different rate of ferrite formation for different concentrations of Cu, favoring the variation of crystallite size. Conventional methods need to high temperatures and prolonged heating time [40,41], but in this method, ferrite phase can produce very fast at low temperature.

The values of lattice parameter (a) for all the compositions of Ni-Cu nano ferrites have been calculated from the values of d-Spacings and are given in **Table 1**.

The lattice parameter (a) of individual composition was calculated by using the formula

$$a = d\sqrt{h^2 + k^2 + l^2}$$
 Å [42]

where a = Lattice Constant

(hkl) are the Miller Indices,

d is inter planner spacing.

A plot of lattice parameter (a) vs. Cu ion content (composition) is shown in **Figure 2**.

It can be seen from the **Figure 2** that the lattice parameter decreases with the increasing Cu substitution in Ni ferrites, the variation may be attributed to the ionic size difference between Cu^{+2} (0.73 Å) and Ni⁺² (0.69 Å) where Cu^{+2} ion replaces Ni⁺² ion on B site high ionic radius Cu^{+2} (0.73 Å) is substituted in the low ionic radius Ni⁺² (0.69 Å) place, that the mixed Ni-Cu ferrite system obeys Vegard's law [43]. The lattice parameter is found to decrease, which may be attributed to shifting on some Fe³⁺ ions form A site to B site for higher composition [36]. The average value of lattice parameter of all compositions is 8.313 Å which is close to 8.345 Å *i.e.* the reported value for bulk sample of mixed Ni-Cu ferrite [44].

The X-ray density (d_x) was computed from the values of lattice parameter (a) using the formula is

X-ray density
$$(d_x) = \frac{8M}{Na^3}$$
 [g/cm³] [45] shown in

Table 1.

Where 8 represents the number of moleules in a unit cell of spinel lattice,

M is the molecular weight of composition,

N is the Avogadro's number a is lattices parameter

A plot of the X-ray density (d_x) Vs Cu ion content (composition) is shown in **Figure 3**.

It can be seen from the **Figure 3** that the X-ray density (d_x) increases with the increasing Cu substitution in Ni Ferrites, it shows the densification of the material. The X-ray density is found to depend on the lattice parameter and molecular weight of the samples, hence the substituting Cu⁺² is increase lattice parameter decreases and X-ray density increases.



Figure 2. Graph between the lattice parameter (a) vs. Cu ion content (composition).



Figure 3. Graph between the X-ray density (d_x) vs. Cu ion content (composition).

Table 1. Values of crystallite size, lattice parameter (a), d-spacing (d), X-ray density (d_x) and hopping length for A-site (d_A) and B-site (d_B) of Ni-Cu nano ferrite (with x = 0, 0.2, 0.4, 0.5, 0.6, 0.8 and 1.0) with 2θ (degrees) and β (degrees) (FWHW).

Sl. No.	Sample (X Value)	$\begin{array}{l} \text{Sample} \\ \text{SValue} \end{array} \begin{array}{l} 2\theta \text{ (degrees)} \begin{array}{l} \beta \text{ (degrees)} \\ \text{(FWHW)} \end{array} \end{array}$		Crystallite Lattice size (nm) parameter in Å (a)		X-ray density (d_x) [g/cm ³]	Volume of unit cell (V) (Å) ³	A-site (<i>d</i> _A)	B-site (<i>d</i> _B)
1	$\mathbf{H}_1 \left(x = 0 \right)$	35.695	0.1419	58.91	8.345	5.396×10^{24}	581.32	2.950	3.614
2	$H_2 (x = 0.2)$	35.680	0.1802	46.3	8.336	5.403×10^{24}	579.28	2.947	3.610
3	$H_3 (x = 0.4)$	35.681	0.1694	49.2	8.330	5.429×10^{24}	578.06	2.944	3.607
4	$H_4 (x = 0.5)$	35.683	0.2236	37.3	8.324	5.447×10^{24}	576.88	2.942	3.604
5	$H_5 (x = 0.6)$	35.690	0.2257	36.9	8.315	5.467×10^{24}	574.89	2.939	3.600
6	$H_6 (x = 0.8)$	35.678	0.1925	43.33	8.295	5.522×10^{24}	570.89	2.932	3.592
7	$H_7(x=1)$	36.059	0.2274	36.7	8.244	5.648×10^{24}	560.37	2.914	3.57

The volume of the unit cell $V = a^3 (Å)^3$.

Calculated values are shown in **Table 1** the volume of the unit cell is decreases with increasing the Cu Substitution in Ni Ferrite, because the lattice Parameter is decreases with increasing the Cu Substitution in Ni Ferrite.

Figure 3 also indicate that the X-ray density (d_x) increases with the increase in Cu ion content, this can be

ascribed to the atomic weight and density of Cu^{+2} (63.546, 8.96 gm/cm³) which are higher than those of nickel (53.933, 8.91 gm/cm³). The X-ray density of nickel ferrite was in good agreement with earlier reported values 5.38 gm/cm³ [46]. The distance between magnetic ions on B and A Sites is calculated according to the following relations [47].

$$d_{\rm B} = 0.25a\sqrt{2}$$
 and $d_{\rm A} = 0.25a\sqrt{3}$

where (a) is the lattice parameter

The values of the hopping length for octahedral (d_B) and tetrahedral (d_A) sites are listed in **Table 1**. It is clear that the distance between the magnetic ions increases as the Cu⁺² content increases.

3.2. Morphological and Chemical Analyses

The morphological and chemical analyses were performed using SEM (Scanning Electron Microscope with an Energy Dispersive Spectrometer (EDS). The Secondary electron images were taken at different magnifications to study the morphological and their Composition of elements of each sample analyzed by (% in weight), obtaied using EDS, chemical compositions were determined by processing the corresponding K-line spectra obtained with an incident beam of 10 KeV electrons. A standardless semi-quantitative analysis was performed for this purpose.

It can seen from SEM micrographs various compo-

sitions that the morphology of the particles is very similar. They indicate that the particle size of the samples lies in the nanometer regine having a sperical shape and a narrow size distribution. The particle sharpness is more or less spherical with some cluster/agglomeration between the particles.

The EDS pattern obtained for all the samples which gives the elemental% and atomic % composition in samples shown in **Table 2**. The elements detected in all the samples were O, Fe, Ni, Cu; the presence of Carbon element is mainly due to the coating applied on the samples prior to the SEM-EPMA (Electron Probe micro-analysis) analysis.

4. Conclusions

- The citrate-Gel auto combution technique to be a conveniant and varsalite for obtaining homogeneous and reactive nanostructured mixed Ni-Cu ferrites. No impurity peak or material loss during processing.
- X-ray diffraction pattern confirm that the formation of single phase cubic spinel structure. The crystallite size of the nano ferrites in the range 36 58 nm.
- The lattice parameter is decreases with the increase of Cu substitution in Ni ferrites indicate that the mixed Ni-Cu ferrite system obeys Vegard's law.
- X-ray density increases with of increase of Cu substitution in Ni ferrites shows the densification of the material.



Figure 4. SEM micrographs of compositions of Ni-Cu nano ferrite (with x = 0, 0.2, 0.4, 0.5, 0.6, 0.8 and 1.0).

Table 2. Composition of elements of each sample analyzed by (% in weight), obtaied using EDS.

Composition	$\mathrm{H}_{1}\left(x=0\right)$		$H_2 (x = 0.2)$		$H_3 (x = 0.4)$		$H_4 (x = 0.5)$		$H_5 (x = 0.6)$		$H_6 (x = 0.8)$		$\mathrm{H}_{7}\left(x=1\right)$	
Element	Element %	Atomic %	Element %	Atomic %	Element %	Atomic %	Element %	Atomic %	Element %	Atomic %	Element %	Atomic %	Element %	Atomic %
С	5.58	14.97	-	-	4.73	9.19	-	-	0.12	0.38	5.49	15.07	-	-
0	21.95	44.20	22.06	40.87	22.43	32.74	16.16	41.71	20.09	48.12	21.34	43.94	21.86	50.94
Fe	38.20	22.03	35.21	18.69	23.00	19.28	25.99	19.22	24.82	17.03	22.21	13.31	39.75	26.54
Ni	34.26	18.80	20.24	20.16	24.04	19.37	27.35	19.24	26.87	17.54	24.46	13.94	-	-
Cu	-	-	22.50	20.27	25.80	19.38	30.00	19.82	28.10	16.94	26.50	13.74	38.39	22.52

- The distance between magnetic ions on B and A Sites is calculated. It is clear that the distance between the magnetic ions increases as the Cu²⁺ content increases.
- SEM analyses explain that the morphology of the particles is very similar. The particle sharpness is more or less spherical with some cluster/agglomeration between the particles.
- EDS data gives the elemental % and atomic % in the composition of mixed Ni-Cu ferrites, and it shows the presence of Ni, Cu, Fe and O without precipitating cations.

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