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New Materials for Vacuum Chambers in High Energy Physics

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Vacuum chambers must fulfil ultra-high vacuum requirements while withstanding thermo-mechanical loads. This is particularly true in high energy particle accelerator where interactions of particles with matter may induce thermal load, material activation, background... The choice of the material of the vacuum chamber is crucial for the final application. Metals such as stainless steel, copper and aluminium are usually used. Even with outstanding mechanical and physical properties, beryllium is used for very specific applications because of its cost and toxicity. Ceramics such as alumina are usually used for fast magnet vacuum chambers. With the next generation of high energy physics accelerator generation such as CLIC and TLEP, the problematic of high cyclic thermal load induced by synchrotron radiation is raised. This paper aims at defining some figures of merit of different materials with respect to several load scenarios and presents briefly their vacuum compatibility.

Keywords

Vacuum Chamber; Figure of Merit; Transparency; Radiation Length; Carbon; Beryllium

1. Introduction

In general, vacuum chambers are designed from a pure mechanical point of view, considering mainly the buckling as the failure mode of the structure. In the high energy physics detectors, they have another constraint to interact as less as possible with the particles. The transparency of a vacuum chamber is determined as the ratio of the thickness over the radiation length, t/X0, and is therefore optimum for the thinnest chamber with the highest radiation length material. From the mechanical point of view, the failure mode of a thin walled vacuum chamber is usually the buckling. A first figure of merit for these applications can be found in [1] [2]. The authors showed the interest of carbon fiber reinforced epoxy resin composite as compared to the beryllium. Vacuum chamber development in composite material began at CERN in the 1980s [2]-[4]. The motivation for this research has

always been to offer a cheaper, more readily available alternative to beryllium vacuum chambers for the experimental areas, whilst maintaining equivalent transparency. Several attempts are recorded in the CERN archives, two examples being carbon fibre chambers that were installed in NA48 and LEP. A study has been reinitiated recently [5].

Nevertheless, this figure of merit is focused on the mechanical aspect and does not take into account thermal and/or thermal fatigue. Indeed, in modern or future high energy physics accelerators, pulsed and/or high energy deposition induced by synchrotron radiation is expected. This heat may lead to high temperature rise and also to high thermal stresses. In this paper, new figures of merits are derived from both mechanical and thermal aspects, considering heat deposition induced by particle interactions with matter.

2. Figure of Merit of Material for Vacuum Chambers

To identify the figures of merit from thermal and mechanical points of view of materials to be applied to UHV thin walled vacuum chambers, the local heat transfer equation and the 1-D constitutive law are used and read, respectively:

$$\rho C \frac{\partial T}{\partial t} = \lambda \Delta T + p \tag{1}$$

$$\sigma = E\left(\varepsilon - \alpha \left(T - T_0\right)\right) \tag{2}$$

The thermal behaviour is described by the heat transfer equation (Equation (1)) where p represents the power of heat dissipation, ρ stands for the specific mass. C and λ denote the specific heat capacity and the thermal conductivity, respectively. The heat dissipation may have several origins: electrical, mechanical, physical or chemical for examples. For high energy physic applications, it's assumed that the heat dissipation is driven by the heat deposition induced by the interaction of particles with matter.

The passage and interaction of particles through the matter is generally complex and requires dedicated Monte Carlo tools. In this paper, only general figures are looked for. High-energy electrons predominantly lose energy in matter by bremsstrahlung and high-energy photons by e + e - pair production. The characteristic amount of matter traversed for these related interactions is called the radiation length X_0 . It is both the mean distance over which a high-energy electron loses all but 1/e of its energy by bremsstrahlung and 7/9 of the mean free path for pair production by a high-energy photon

 X_0 is a material parameter that is approximated by [6]:

$$X_0 [\text{cm}] \sim \frac{1}{\rho} \frac{716.4A}{Z(Z+1) \ln\left(\frac{287}{\sqrt{Z}}\right)}$$
 (3)

with Z and A the atomic number and the atomic weight, respectively.

For electrons, the energy deposition induced by bremsstrahlung, is given by:

$$\frac{\mathrm{d}E}{\mathrm{d}x} = -\frac{E}{X_0} \tag{4}$$

For photons, the energy absorption is complex and strongly depends on the photon energy. Three main phenomena dominate successively: photoelectric effect, Compton scattering and pair production (**Figure 1**) [7].

The energy absorption length, λ_{en} , defined in [8], is obtained from the mass energy absorption coefficient given in [9] (**Figure 2**).

In the range 0.1 - 1 MeV, the Compton scattering process dominates the interaction cross section for most of the materials [10]. A power law correlation is used to approximate the energy absorption length with respect to the radiation length. A good approximation is obtained for $\lambda_{en} \sim X_0^{0.5}$, that is coherent with the linear dependence of the Compton cross section with the atomic number [8].

At higher energy, the pair production dominates. Its cross section varies as Z^2 [8]. A good linear correlation between the energy absorption length and the radiation length, $\lambda_{en} \sim X_0$, is obtained, as expected.

Therefore, in this study, it's assumed that the heat deposition induced by high energy electrons or photons is inversely proportional to the radiation length (to the square root of the radiation length for lower energy photons).

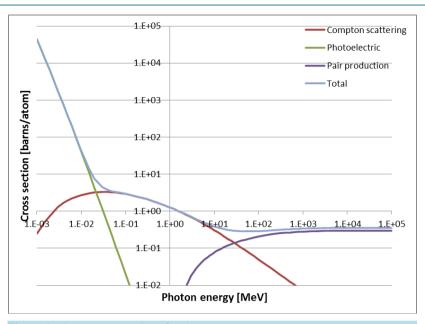


Figure 1. Photon cross section of carbon.

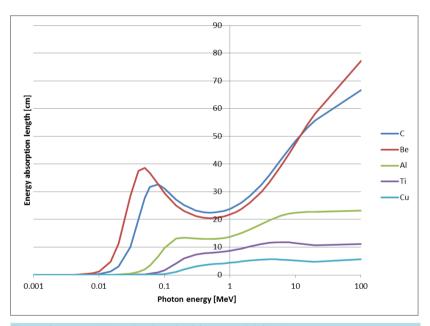


Figure 2. Energy absorption length of few material for photons.

Two thermal regimes are considered: a transitory regime for fast phenomenon where diffusion can be neglected and a steady state. They both induce temperature rise and therefore thermal stresses. For the first one, the temperature rise is proportional to:

$$\delta T \propto \frac{1}{X_0 \cdot \rho \cdot C} \tag{5}$$

To characterize the material and its ability to withstand the temperature rise, it has been normalized with respect to the melting temperature, T_f , and a first figure a merit for the material is expressed by:

$$\frac{1}{X_0 \cdot \rho \cdot C \cdot T_f} \tag{6}$$

The temperature increase (Equation (5)) and the constitutive law (Equation (2)) are used to approximate the thermal stresses that could induce thermal fatigue. This thermal stress could be compared to the endurance stress. Nevertheless they have been normalized to the yield stress, σ_y , which is a more available parameter. Therefore figure of merit related to transitory thermal stresses is given by:

$$\frac{E \cdot \alpha}{X_0 \cdot \rho \cdot C \cdot \sigma_{y}} \tag{7}$$

with E the Young modulus. α denotes the coefficient of thermal expansion. This parameter represents the ability of the material to withstand pulsed high energy deposition.

In the case of a thermal steady state, the normalized temperature rise gives a third figure of merit:

$$\frac{1}{X_0 \cdot \lambda \cdot T_f} \tag{8}$$

The transparency of a vacuum chamber is determined as the ratio thickness over the radiation length, t/X_0 , and is therefore optimum for the thinnest chamber with the highest radiation length material. From the mechanical point of view, the failure mode of a thin walled vacuum chamber is usually the buckling. The critical buckling pressure is in the form:

$$P_{cr} = \frac{E}{4} \left(\frac{t}{R}\right)^3 \tag{9}$$

where t and R denote the thickness and radius, respectively. A figure of merit for the material, considering the transparency and the buckling strength, can therefore be expressed by:

$$X_0 E^{1/3}$$
 (10)

These four figures of merit are independent even if the Equations (6) and (7) are linked because of the same transient thermal aspect.

3. Material Comparison

A few materials have been considered and compared to beryllium, usually used or proposed for these applications. Non-linear effects are not taken into account and the material parameters have been defined at room temperature (Table 1).

The ratio with respect to the beryllium of the figure of merits considering the different criteria have been evaluated (higher than 1 meaning better than beryllium) and presented in **Figure 3**.

Glassy carbon and silicon carbide (reinforced with carbon fibres or not) are two interesting candidates for "transparent" vacuum chambers. They are good intermediate solutions between beryllium and classical metallic material. Glassy carbon material exhibits advantageous thermal fatigue properties, due to its low thermal expan-

Table 1. Material parameters.

		Beryllium	Glassy Carbon	Aluminium	Titanium	316L	SiC	Copper	C/SiC 25/75	AlBeMet 162 30/70	AlBeMet 140 50/50
ρ	$[g \cdot cm^{-3}]$	1.85	1.42	2.8	4.4	8	3.2	9	3	2.1	2.28
C	[J/kg/K]	1830	820	870	560	500	1130	385	710	1506	1280
λ	[W/K/m]	200	6.7	120	16.7	26	145	400	145	210	210
α	$[10^{-6} \text{ K}^{-1}]$	12	2.6	22	8.9	16	4.8	17	2.3	13.9	16.5
X_0	[cm]	35	29	9	3.7	1.8	8	1.47	10.1	23.4	18
T_f	[K]	1560	3300	820	1820	1650	3000	1360	3000	930	930
σ_{y}	[MPa]	345	260	275	800	300	400	200	300	193	193
Е	[GPa]	230	35	73	115	200	450	115	350	193	150

sion coefficient and low Young modulus.

A similar figure is obtained in the case of lower energy photon interactions (Figure 4).

4. Alternative Materials to Beryllium

Beryllium is an outstanding material with high and homogeneous thermo-mechanical properties (**Figures 3** and **4**). Nevertheless, it has some drawbacks such toxicity, brittleness, availability and cost. In addition, it is not necessarily the best choice for thermal fatigue. Therefore, alternative materials, with comparable properties, are

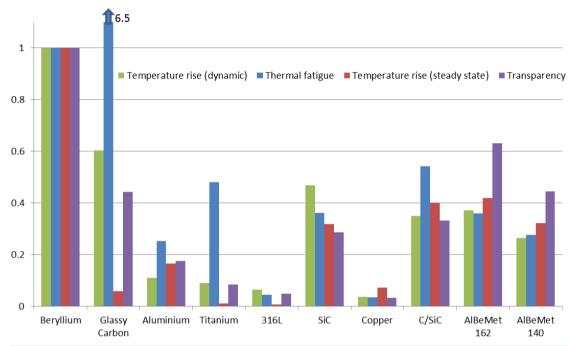


Figure 3. Comparison of materials according to different figures of merit at high energy photons and electrons.

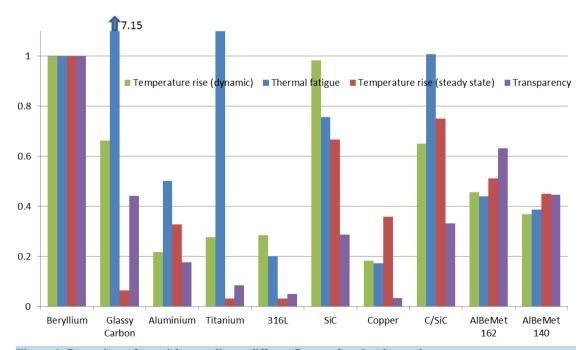


Figure 4. Comparison of materials according to different figures of merit at lower photon energy.

looked for. A study for an application in high energy physics detectors, that requires an optimisation for the transparency, has been initiated [5]. This kind of material, with low atomic number, might be used in a wider scope and not only in experimental areas. Different material technologies are considered:

- 1) Raw bulk material: the material is homogeneous and leak tight.
- 2) Material composite: The material is composed of a matrix reinforced with fibres or particulates. The leak tightness is obtained either in the bulk or with a coating. Mechanical properties that can be obtained by homogenization theories are given by the composite properties.
- 3) Structural composite: A thin membrane, likely in aluminium, is used to obtain the leak tightness. It is reinforced to withstand the mechanical loads. The macroscopic behaviour is given by the reinforcement and the aluminium foil.

After beryllium, carbon is the element with the lowest atomic number that is appropriate for this application; therefore all high carbon content materials, especially composites based on carbon fibre that offer high stiffness and high radiation length, are good candidates for transparent vacuum chambers. An overview of the materials that have been considered is presented briefly. It aims at giving main properties and limitations.

5. Raw Material

Glassy carbon (GC) is obtained by the pyrolysis at high temperature of a highly reticulated resin. The material is composed of around 98% (weight) of carbon and 2% of oxygen. The thermal outgassing and the permeation have been measured. The GC grade, manufactured at high temperature ($\sim 3000^{\circ}$ C) and provided by HTW, Germany, shows a good vacuum performance. The unbaked glassy carbon has a thermal outgassing of around 2.10^{-9} mbar·l·s⁻¹·cm⁻² after 10 hours with a decrease in t^{-1.4} (**Figure 5**), whereas the baked sample has a thermal outgassing of around $2.5 \cdot 10^{-13}$ mbar·l·s⁻¹·cm⁻² for hydrogen after a baking at 150°C for 2 hours. The main issue of this material lies in its brittleness and the difficulty to join a glassy carbon part with a standard metallic interface flange.

The aluminium well remains a good backup solution since their manufacturing processes and characteristics are known and/or manageable. Thin walled vacuum chambers can be manufactured in aluminium. New grades with lithium are now available (series 2000) or in development (series 5000). The addition of lithium tends to increase the Young modulus the mechanical strength and the transparency. Mechanical and welding tests are on-going to qualify the available grades.

The solution of silicon carbide ceramic has not yet been extensively studied. Preliminary thermal outgassing tests have been carried out on an unbaked sample and a value around 2.10^{-9} mbar·l·s⁻¹·cm⁻² after 10 hours, has been obtained (**Figure 5**). This material is available in the industry and can be soldered to metallic parts. More detailed study on vacuum performance and feasibility is required.

6. Composite Material

Sample pieces of carbon fibre reinforced resin of $100 \times 100 \times 2$ mm were coated by PVD. Coatings of Aluminium, Silicone and Titanium were applied. Coating thicknesses ranged from 100 to 500 nm. When carrying out leak tests on these samples, all items started with a background between 10^{-9} and 10^{-10} mbar·l·s⁻¹ and rose to between 10^{-7} and 10^{-8} mbar·l·s⁻¹ after several hours [11]. A non-coated sample exhibited similar behaviour. Samples have also been coated by another method (Hipims) and showed similar results. First impressions are that single layer coatings, no matter how controlled the environment, may always have small leaks. This would be coherent with previous studies of coatings on polymers [12] that exhibited pin holes due to the presence of dust. These pin holes led to high thermal outgassing. Further developments are foreseen. The first one is a multilayer coating with intermediate treatments. They would aim at a redistribution of the dust and therefore of the pin holes between two layers. The second alternative is an electrolytic coating with ionic liquid.

Metallic matrix materials have been considered, especially long carbon fibre reinforced aluminium matrix. The composite is obtained by gas pressure infiltration of liquid aluminium in a carbon fibre preform. The galvanic coupling between aluminium and carbon is an issue in a saline environment but probably not in normal conditions in an accelerator or experiment [12]. In addition, titanium coating on the fibres can prevent this effect. Some samples have been analysed. First leak tests and thermal outgassing measurements on samples provided by Thales Alenia Space and Dresden University have shown good and promising vacuum performance [11]. Nevertheless, this material is not widely commercially available. An alternative with short fibres has been pro-

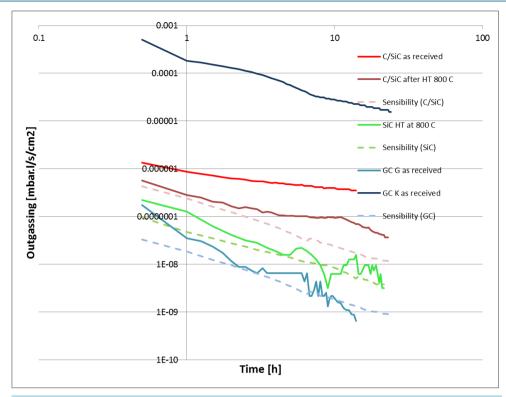


Figure 5. Thermal outgassing of unbaked glassy carbon and C/SiC composite materials.

posed. It could be obtained by powder technology and could ease the manufacturing process. Reliability under thermal cycles (bake out) has not been assessed.

A carbon fibre in ceramic matrix composite is an interesting material (**Figures 3** and **4**). A sample of short carbon fibre reinforced silicon carbide composite has been provided by CeSIC, Germany. This unbaked material, as received, has a thermal outgassing of 4.10^{-7} mbar·l s⁻¹·cm⁻² after 10 hours of pumping and a decrease with time in t^{-0.34} (**Figure 5**).

7. Structural Composite

A first solution is based on a thin metallic envelop, reinforced internally to withstand the atmospheric pressure. The internal reinforcement has to exhibit low outgassing and a proposal in C/C (Carbon Fibre Reinforced Carbon) has been studied. Outgassing tests were made with a Ti external barrier, having obtained a rate of about 1.10^{-12} mbar·l·s⁻¹·cm⁻² after a bake-out at 250°C [13], which is compatible with previous tests [14] [15]. NEG coating was successfully made and results are promising. After activating for 6 hours at 250°C the sample that has a coated surface of around 2500 cm², a pumping speed of 500 l·s⁻¹ for H2 was measured. Nevertheless, the differential thermal expansions between the aluminium envelop and the C/C reinforcement lead to a large deformation of the thin metallic membrane and therefore limits the reliability of such solution for applications requiring a bake out or thermal cycles [15].

Another solution is the external reinforcement of an internal leak tight membrane in aluminium. The reinforcement might be wrapped carbon fibres with resin or a sandwich structure. Due to the permeability of the reinforcement, the interface between the membrane and the resin has to withstand the vacuum force most importantly including the bake out when higher stresses are generated due to differential thermal expansion. For polymeric resin, no strong covalent bounds occur and creep, especially at high temperature may result. The strength of the interface between the internal membrane and the reinforcement is a limiting factor.

8. Conclusion

This paper defines four different figures of merit for materials that are relevant for vacuum chambers subjected

to high energy deposition or that interact with physics particles. They are based on both mechanical and thermal considerations. The figures of merit have to be chosen according to the final applications and requirements. They have been compared for a few materials to beryllium. Glassy carbon shows interesting properties as well as carbon fibre reinforced silicon carbide. Their thermal outgassing has been evaluated for unbaked sample. One grade of glassy carbon exhibits low outgassing and seems to be fully compatible for UHV applications. This work is still on-going and more detailed measurements are in preparation, especially for baked material.

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