

Molybdenum Phosphide Flakes Catalyze Hydrogen Generation in Acidic and Basic Solutions

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Abstract

Molybdenum phosphide (MoP) flakes were synthesized by the reduction of hexaammonium heptamolybdate tetrahydrate and ammonium dihydrogen phosphate. The flakes are porous and constructed by MoP nanoparticles with *ca.* 100 nm diameters. The lateral size of flakes ranges from less than 1 μ m to larger than 5 μ m, and the thickness of MoP fakes is *ca.* 200 nm. The mixture of MoP flakes and carbon black exhibits effective catalytic activity in the hydrogen evolution reaction. The optimal overpotential required for 20 mA·cm⁻² current density is 155 mV in acidic solution and 184 mV in basic solution. The mixture can work stably in long-term hydrogen generation in both acidic and basic solution. The faradaic yield of mixture in hydrogen evolution reaction is nearly 100% in both acidic and basic solution. The Mo and P species in MoP flakes are found to have small positive and negative charge, respectively. The catalytic activity of MoP flakes is likely to be correlated with this charged nature.

Keywords

Molybdenum Phosphide, Hydrogen Evolution Reaction, Catalyst, Electrolysis

1. Introduction

As a clean and renewable resource, hydrogen is believed to be one of the most promising alternative energy carriers [1] [2]. Hydrogen can be generated by electrolysis or photoelectrolysis of water *via* the hydrogen evolution reaction (HER). Effective electrocatalysts for HER are essential for efficient hydrogen generation from electro-

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lysis and photoelectrolysis. Though platinum-group metals have shown excellent catalytic activity in HER, their widespread commercial application is inhibited by high cost and low abundance. The exploitation of effective and low-cost HER catalysts is therefore highly desirable.

Recently, the development of cost-effective HER catalysts has gained extensive attention. Great efforts have has been devoted to explore HER catalysts among transition metal chalcogenides (e.g., molybdenum sulfide [3]-[7], tungsten sulfide [8] [9], cobalt dichalcogenide [10] [11], iron dichalcogenide [10] [11], and nickel dichalcogenide [10] [11]), carbides (e.g., molybdenum carbide [12]-[14], and tungsten carbide [15] [16]), as well as nitrides and carbonitrides (e.g., molybdenum nitride [13], cobalt-molybdenum nitride [17], tungsten carbonitride [18]). On the other hand, nickel phosphide has been predicted to be an excellent catalyst in HER in 2005 [19], whereas the catalytic activity of nickel phosphide (Ni₂P nanoparticles) was experimentally confirmed in 2013 [20]. The successful demonstration of catalytic activity of Ni₂P inspires research concerning the application of metal phosphide in hydrogen generation (e.g., Ni₁₂P₅ [21], and CoP [22] [23]).

Herein the catalytic activity of MoP flakes in HER is demonstrated. We found that MoP flakes show efficient catalytic activity in HER in both acidic and basic solution, and the corresponding overpotential required for 20 mA·cm⁻² current density is 155 mV in acidic solution and 184 mV in basic solution, respectively. The performance is favorably comparable to most values of reported nonprecious metal catalysts. Potentiostatic electrolysis and accelerated degradation experiments demonstrate the long-term stability of MoP flakes in hydrogen generation in acidic and basic solution. The influence of synthesis temperature on catalytic activity of MoP flakes was revealed. Tafel slope suggests that the HER occuring on the surface of MoP flakes proceeds along Volmer-Heyrovsky mechanism. Mo and P in MoP flakes were found to have slight charge, and the catalytic activity of MoP might be correlated with these features.

2. Experimental

2.1. Synthesis of MoP Flakes

Hexaammonium heptamolybdate tetrahydrate $((NH_4)_6Mo_7O_{24}\cdot 4H_2O, 0.9 \text{ g})$ and ammonium dihydrogen phosphate $(NH_4H_2PO_4, 0.586 \text{ g})$ were dissolved in 100 mL deionized water, and the mixture was stirred at 90°C to evaporate all deionized water. The resulting dried powder was grinded in a mortar mixer, and then loaded in a quartz tube mounted in a tube furnace. The quartz tube was pumped to 20 Pa and filled with 5% H_2/N_2 . This procedure was repeated five times prior to heating to remove oxygen in the tube. After that, the temperature was increased to 800°C (heating rate: 3°C·min⁻¹), and maintained at 800°C for 120 min. During heating, the quartz tube was flowed with 5% H_2/N_2 (flow rate: 100 sccm). After the reaction, the furnace was cooled naturally to room temperature.

2.2. Characterization

The structure of MoP flakes was investigated by X-ray diffraction (XRD) using a Bruker D8 Advance diffractometer with graphite-monochromated Cu K α radiation ($\lambda = 1.54178$ Å). The morphology of MoP flake was revealed by transmission electron microscopy (TEM) and scanning electron microscopy (SEM). TEM and SEM experiments were carried out on a JEM2100 (JEOL) and a JSM7001F (JEOL), respectively. The EDX spectra were recorded using an Oxford Instruments' INCA system equipped on the JSM7001F. For the TEM investigation, MoP flakes were dispersed in ethanol and then loaded onto a carbon-coated copper grid (300-mesh) by drop-coating. The X-ray photoelectron spectroscopy (XPS) experiments were carried out on an ESCALAB-250Xi System (ThermoFisher) equipped with a monochromatic Al K α (1486.6 eV) source and a concentric hemispherical energy analyzer.

2.3. Electrochemical Measurement

All electrochemical measurements were carried out on a CHI 614D electrochemical workstation (CH Instrument) in a three-electrode electrochemical cell. A graphite rod (6 mm diameter) was used as counter electrode, and the counter electrode was separated from working chamber by porous glass frit. A mercury/mercurous sulfate electrode (MSE) or mercury/mercury oxide electrode (MMO) was used as reference electrode. The catalyst (4 mg), certain amount of carbon black (0, 2, 4, or 6 mg), and Nafion solution (5 wt%, 80 μ L) were dispersed in 1 ml of water/ethanol (4/1, v/v) by ultrasonication (ultrasonic probe, 2 mm diameter, 130 W, 1 h) to form homogeneous

ink. Then different amounts of catalyst ink were loaded onto a glassy carbon electrode (3 mm diameter).

All solutions were purged with high purity H_2 (99.999%) for 30 min prior to electrochemical measurements and during electrochemical measurements. The experiments carried out in H_2SO_4 solution (0.5 M) used the MSE as reference electrode, and those in KOH (1 M) solution used the MMO. The reversible hydrogen evolution potential (RHE) was determined by the open circuit potential of a clean Pt electrode in the solution of interest bubbled with H_2 (99.999%), being -0.694 V *vs* MSE for 0.5 M H_2SO_4 solution and -0.876 V *vs* MMO for 1 M KOH solution.

Polarization curves were measured at a sweep rate of 5 mV·s⁻¹ in rigorously stirred solution (1600 rpm). The uncompensated cell resistance (R) was determined by the current-interrupt method. Cyclic voltammetry (CV) sweep was carried out 50 mV·s⁻¹ sweep rate. Electrochemical impedance spectroscopy (EIS) measurements were carried out at different potentials in the frequency range 10^{-2} to 10^{6} Hz with 10 mV sinusoidal perturbations and 12 steps per decade in 0.5 M H₂SO₄ solution.

The current and charge passing the circuit were measured with the electrochemical workstation and the voltage change of the MPXV7002DP was monitored with a digital multimeter (4 1/2 digits). Prior to experiment, the relationship between volume of gathered gas and the variation of output voltage of the MPXV7002DP (*i.e.*, pressure variation in the gas gathering tube) was calibrated by injecting known amounts of air into the gas gathering tube and recording the variation of output voltage of the MPXV7002DP.

The volume of H_2 during the potentiostatic electrolysis experiment was monitored by volume displacement method in a configuration shown in **Figure 1**. A Freescale MPXV7002DP differential pressure transducer was employed to monitor pressure variation in the gas gathering tube, and then the volume of generated H_2 was computed from pressure variation in the gas gathering tube. If the initial height of water in the gas gathering tube before the gas gathering experiment is h_0 , and after the potentiostatic electrolysis experiment the generated gas is gathered into the tube and the final height of water in the gas gathering tube become h_1 , then the volume of generated gas (V) should be $s(h_0 - h_1)$, where s is the inner cross-sectional area of the gas gathering tube.

At the initial status, the pressure inside the gas gathering tube (P_0) is $P - \rho gh_0$, where P is the atmospheric pressure, is the density of water, and g is the acceleration due to gravity. The output voltage of the differential pressure transducer would be $U_0 = k(P - P_0) = k\rho gh_0$, where k is the sensitivity of the differential pressure transducer (1 mV/Pa for a Freescale MPXV7002DP). When the height of water in the gas gathering tube decreases to h₁, the pressure inside the gas gathering tube becomes P₁, and P₁ = P - ρgh_1 . Then, the output of the differential pressure transducer is $U_1 = k(P - P_1) = k\rho gh_1$.

Accordingly, the volume of generated gas can be computed by $V = s(h_0 - h_1) = s(U_0/k\rho g - U_1/k\rho g) = s(U_0 - U_1)/k\rho g = C(U_0 - U_1)$, where C is a coefficient that can be calibrated by injecting a known volume of gas into the gas gathering tube and recording the variation of output voltage of the differential pressure transducer.

3. Results and Discussion

MoP was synthesized by the reduction of hexaammonium heptamolybdate tetrahydrate ($(NH_4)_6Mo_7O_{24}\cdot 4H_2O$) and ammonium dihydrogen phosphate ($NH_4H_2PO_4$) at 800°C. The XRD pattern of product Figure 2(a) exhibits



volume of generated gas.



Figure 2. (a) XRD pattern of MoP. (b) Low and (c) high magnification SEM images of MoP flake. (d) TEM image of MoP flake. (e) HRTEM image of MoP flake.

distinct peaks, and these peaks can be assigned to those of hexagonal phase MoP (JCPDS No. 65-6487, a = 3.223 Å, c = 3.191 Å). These distinct peaks suggest the good crystallinity of MoP. The composition information of product was accessed by EDX analysis carried out in SEM. The average atomic ratio of Mo to P recorded from five different sites is 0.976 ± 0.053 (Figure 3 and Table 1), in accordance with that of stoichiometrical MoP (1:1).

Typical morphology of MoP was investigated by SEM and TEM. Figure 2(b) shows the low magnification SEM image of MoP, revealing a flake like morphology. The lateral size of MoP flakes ranges from less than 1 μ m to larger than 5 μ m, and the thickness of MoP fakes is *ca*. 200 nm (Figure 2(c)). It is further revealed that the MoP flakes are porous and constructed by MoP nanoparticles with diameter *ca*. 100 nm (Figure 4(d)). The high-resolution TEM (HRTEM) image suggests that these small MoP nanoparticles are single crystalline (Figure 4(e)). The good crystallinity of the MoP nanoparticles can be suggested by the well defined lattice fringes. The distances between fringes are measured to be 2.8 and 3.2 Å, respectively, corresponding well to those between the (100) and (00-1) planes of hexagonal phase MoP. The pattern resulted from fast Fourier transform (FFT) of the lattice fringes is shown in the inset of Figure 2(e), matching well that of the (010) zone axis diffraction pattern of hexagonal phase MoP.

The MoP flakes exhibit effective catalytic activity in HER. Linear sweep voltammetry (LSV) experiments were carried out to obtain polarization curves of MoP flakes. Figure 4 shows the optimal performance of MoP flake loaded on glassy carbon electrode (GCE) with and without addition carbon black (Vulcan XC-72R). For comparison, the polarization curves of bare GCE and GCE loaded with carbon black (C on GCE) are also shown. It is found that in acidic solution (H₂SO₄, 0.5 M) overpotential required for current density of 10 mA·cm⁻² (η_{10})



Figure 4. Polarization curves of MoP/C on GCE (loading amount: $1.425 \text{ mg} \cdot \text{cm}^{-2}$ of MoP and $0.7125 \text{ mg} \cdot \text{cm}^{-2}$ of C), MoP on GCE (loading amount: $0.855 \text{ mg} \cdot \text{cm}^{-2}$), C on GCE (loading amount: $0.7125 \text{ mg} \cdot \text{cm}^{-2}$), and bare GCE. Potentials are corrected with iR drop.

Table 1. Atomic ratio of Mo to P determined by EDX experiments in SEM.										
	1	2	3	4	5	Average				
Mo:P	0.923	1.05	0.938	0.958	1.01	0.976 ± 0.053				

for MoP on GCE is *ca.* 423 mV, while η_{10} is markedly reduced to 141 mV and η_{20} (overpotential required for current density of 20 mA·cm⁻²) is as small as 155 mV for the hybrid materials of MoP flakes and carbon black (MoP/C on GCE). The current density of carbon black on GCE (C on GCE) is smaller than that of MoP on GCE, and bare GCE shows neglectable current density in the potential range of -0.5 - 0 V *vs* RHE, suggesting that the current in MoP/C on GCE sample can be associated with MoP flakes. The small current density in MoP on GCE sample might be correlated with slow electron transport between less-conducting MoP flakes. The introduction of carbon black enables rapid electron transport from electrode to MoP flakes at catalyst/electrolyte interface, and results in markedly enhanced hydrogen generation performance.

The η_{10} and η_{20} are usually adopted as key parameters for the comparison of the catalytic activity of different HER catalysts, because the typical current density of photoelectrochemical water splitting cell ranges in 10 - 20 mA·cm⁻² under the solar photon flux of 1 Sun AM 1.5 illumination [24]. Typical η_{10} and η_{20} of nonprecious HER catalysts are listed in **Table 2**. It is shown that the optimal η_{10} and η_{20} of MoP/C on GCE are favorably comparable to most values of reported nonprecious metal HER catalyst.

The HER performance depends heavily on the loading amount of MoP flake and carbon black. A series of

Catalyst	Substrate	Mass density (mg/cm ²)	η_{10}^{b} (mV)	η_{20}^{c} (mV)	Electrolyte
Amorphous MoS ₃ -CV [7]	GCE		211	229	1 M H ₂ SO ₄
WS ₂ nanosheets [8]	GCE	0.285	151	177	1 M H ₂ SO ₄
WS ₂ nanosheets [9]	GCE	0.0001 - 0.0002 or <i>ca.</i> one continuous layer	233	275	0.5 M H ₂ SO ₄
FeSe ₂ [11]	GCE				$0.5 \text{ M} \text{ H}_2 \text{SO}_4$
CoS_2 [11]	GCE		232		$0.5 \text{ M} \text{ H}_2 \text{SO}_4$
Fe _{0.43} Co _{0.57} S ₂ [11]	GCE		264		$0.5 \text{ M} \text{ H}_2 \text{SO}_4$
CoSe ₂ [11]	GCE	0.037	231		$0.5 \text{ M} \text{ H}_2 \text{SO}_4$
Co _{0.56} Ni _{0.44} Se ₂ [11]	GCE		250		$0.5 \text{ M H}_2 \text{SO}_4$
$Co_{0.32}Ni_{0.68}S_2$ [11]	GCE				$0.5 \text{ M H}_2 \text{SO}_4$
NiS ₂ [11]	GCE				$0.5 \text{ M} \text{ H}_2 \text{SO}_4$
NiSe ₂ [11]	GCE		250		$0.5 \text{ M} \text{ H}_2 \text{SO}_4$
Bulk Mo ₂ C [12]	Carbon paste electrode	1.4	208	224	$0.50 \text{ M} \text{ H}_2 \text{SO}_4$
Bulk MoB [12]	Carbon paste electrode	2.5	212	227	0.50 M H ₂ SO ₄
Mo ₁ Soy [13]	Carbon paper	1.4	177		0.1 M HClO ₄
Mo ₁ Soy-RGO [13]	Carbon paper	0.47	109		0.1 M HClO ₄
Mo ₂ C/C [13]	Carbon paper	2	311		0.1 M HClO ₄
Mo ₂ C/CNT [14]	Carbon paper	2	149		0.1 M HClO ₄
Fe-WCN [16]	RRDE	0.4	220		$H_2SO_4 (pH 1) + Na_2SO_4 (0.5 M)$
$Co_{0.6}Mo_{1.4}N_2$ [17]	GCE	0.243	202	267	0.1M HClO ₄
FeS ₂ [18]	GCE		192.6		0.5 M H ₂ SO ₄
Ni ₂ P [20]	Ti foil	1	117	130	0.50M H ₂ SO ₄
Ni ₁₂ P ₅ [21]	Ti foil	3		143	0.50 M H ₂ SO ₄
CoP/CNT [22]	GCE	0.285	122		0.50 M H ₂ SO ₄
CoP [23]	Ti foil	2		85	0.50 M H ₂ SO ₄
CoP [30]	carbon cloth	0.92	67	100	0.50 M H ₂ SO ₄
Ni-Mo nanopowder [31]	Ti foil	1		70	2 M NaOH
Ni-Mo nanopowder [31]	Ti foil	3		80	0.5 M H ₂ SO ₄
Ni-Mo nanopowder [31]	Ti foil	1	79	107	1 M NaOH
MoS ₂ (33%)/MWCNT-NC [32]	Silver electrode	0.255	206	226	1 M H ₂ SO ₄
Core-shell MoO ₃ -MoS ₂ nanowires [33]	FTO		254	272	0.5 M H ₂ SO ₄
Defect-rich MoS ₂ nanosheets [34]	GCE	0.285	190	214	0.5 M H ₂ SO ₄
MoS ₂ @Au [35]	Au electrode	0.00103	226		$0.5 \text{ M H}_2 \text{SO}_4$
MoS ₂ /RGO hybrid [36]	GCE	0.285	154	176	$0.5M H_2 SO_4$
MoS ₂ /MGF [37]	GCE	0.21	146	159	0.5 M H ₂ SO ₄
MoS ₂ /CNTs [38]	Glass carbon disk	0.136	184	230	$0.5 \text{ M H}_2 \text{SO}_4$
Cu ₂ MoS ₄ [39]	GCE	0.0425	321		pH 0 H ₂ SO ₄
WS ₂ /RGO [40]	GCE	0.4	265	292	0.5M H ₂ SO ₄
Cobalt-sulfide catalyst [41]	FTO		165	227	1.0 M pH 7 PBS
NiWS _x [41]	FTO		373	430	pH 7 PBS
CoWS _x [41]	FTO		271	311	pH 7 PBS
CoMoS _* [41]	FTO		241	282	pH 7 PBS

Table 2. Summary of HER performance of representative catalysts.

 $a^{a}\eta_{10}$: overpotential required for 10 mA·cm⁻² current density; $b^{a}\eta_{20}$: overpotential required for 20 mA·cm⁻² current density.

LSV experiments were carried out to determine the optimal performance of MoP flake. These results are shown in **Figure 5**. The η_{10} of MoP on GCE sample firstly increases with the loading amount of MoP, with the optimal loading amount of 0.855 mg·cm⁻², and further increase the loading amount of MoP results in the decreasing of η_{10} . The relationships between η_{20} and the loading amount of MoP in MoP/C on GCE samples with different weight ratios of MoP to C are summarized (**Figure 5(b)**), and the corresponding polarization curves are shown in panels c-e of **Figure 5**. It is shown that the smallest η_{20} can be found in MoP/C on GCE with the weight



Figure 5. (a) Polarization curves of GCE loaded with different amount of MoP flake. (b) The relationship between η_{20} and loading amount of MoP in MoP/C on GCE. Different weight ratios of MoP to C have been evaluated to find the optimal η_{20} . The polarization curves of MoP/C on GCE for different weight ratio of MoP to C: (c) 2/1, (d) 1/1, and (e) 2/3. Potentials were corrected with iR drop.

ratio of MoP to C being 2:1 and the loading amount of MoP being 1.425 mg·cm⁻².

The fabrication temperature influences the catalytic activity of MoP flake in HER. MoP flakes were fabricated at different temperatures, and the products are denoted as MoPxxx, where xxx is the fabrication temperature. The plots of η_{20} versus the loading amounts of MoP750 and MoP850 are shown in Figure 6(a), and the polarization curves of MoP750, and MoP850 can be found in Figure 6(b) and Figure 6(c), respectively. The optimal η_{20} is 210 mV for M750 (the loading amount of MoP: 1.425 mg·cm⁻²) and 190 mV for M850 (the loading amount of MoP: 1.425 mg·cm⁻²), respectively, both larger than that of MoP800 (Figure 4). The XRD experiments reveal that the structure of M750 and M850 is the same as that of MoP800 (Figure 7(a)), and no impurity phase can be detected from the corresponding XRD patterns. The peaks are more distinct for sample obtained at higher temperature, suggesting that higher temperature would be beneficial for the crystallization of MoP flake. The specific surface areas of MoP750, MoP800, and MoP850 were measured by Brunauer, Emmett and Teller (BET) method Figure 7(b), being 4.798, 4.835, and 3.767 m²·g⁻¹, respectively. The results of XRD and BET experiments suggest that the crystallinity and surface area of MoP flake work cooperatively, resulting in an optimal catalytic activity of sample obtained at 800°C.

The faradaic efficiency of MoP flakes in HER was evaluated by the comparison of measured volume and theoretical volume of generated hydrogen. The generated hydrogen was measured by water displacement method, and the theoretical volume was computed by assuming that all elecrons passing through the circuit are consumed by the reduction reaction of $H^+ (2H^+ + 2e^- \rightarrow H_2)$. Figure 8 shows the plots of measured volume and theoretical volume of hydrogen generated during a potentiostatic electrolysis. The measured volume of hydrogen matches well the theoretical one within experimental error, suggesting that the faradaic efficiency of MoP flakes is nearly 100% in hydrogen generation.



Figure 6. (a) The relationship between η_{20} and loading amount of MoP in MoP/C on GCE for MoP fabricated at different temperatures. Polarization curves of MoP fabricated at (b) 750°C and (c) 850°C. The weight ratio of MoP to C is kept as 2:1. Potentials were corrected with iR drop.



Figure 7. (a) XRD patterns of products obtained at different temperatures. (b) Nitrogen adsorption/desorption isotherm of MoP obtained at different temperatures.



Figure 8. Plots of theoretical and measured volume of generated hydrogen in a potentiostatic electrolysis experiment. The sample is MoP/C on GCE (loading amount: $1.425 \text{ mg} \cdot \text{cm}^{-2}$ of MoP and $0.7125 \text{ mg} \cdot \text{cm}^{-2}$ of C), and applied potential is -0.25 V vs RHE (without iR correction).

Long-term stability during hydrogen generation is also important for the practical application of HER catalyst. MoP flake was found to work stably in acidic solution during hydrogen generation, as indicated by potentiostatic electrolysis and accelerated degradation experiments (**Figure 9**). A potentiostatic electrolysis experiment shows that only slightly decrease of current density occurs in the first 5000 s, and thereafter the current density is nearly unchanged. An accelerated degradation experiment was carried out by repeated cyclic voltammetry (CV) sweep. Negligible difference can be found between the polarization curves of the initial and 2000th scan. The potentiostatic electrolysis and accelerated degradation experiments demonstrate the long-term hydrogen generation capability of MoP flakes.

To obtain insight into the HER process of MoP flakes, EIS experiments were carried out in acidic solution. The spectra corresponding to different applied potentials are shown in a Nyquist plot in **Figure 10(a)**. All spectra contain two semicircles, and they can be well fitted with a two-time-constant equivalent circuit (**Figure 11**). Charge transfer resistance (R_{ct}) at solid/liquid interface was derived from semicircles at low frequencies range. R_{ct} is associated with HER kinetics of a catalyst, and a smaller R_{ct} suggests faster kinetics. **Figure 10(a)** shows that R_{ct} decreases with increasing applied potential, in accordance with larger current density at larger applied potential. The applied potentials are plotted *versus* inverse R_{ct} on a logarithmic scale (**Figure 10(b**)), and the Tafel slope was determined by the slope in the plot, being 71.77 mV·dec⁻¹.

The HER process can be revealed by Tafel slope. In general, a classic two-electron-reaction model suggests that the HER process can proceed in two steps: a discharge step (Volmer reaction: $H_3O^+ + e^- \rightarrow H_{ads} + H_2O$) followed by a desorption step (Heyrovsky reaction: $H_{ads} + H_3O^+ + e^- \rightarrow H_2 + H_2O$), or a discharge step followed by a recombination step (Tafel reaction: $H_{ads} + H_{ads} \rightarrow H_2$), where H_{ads} represents a H atom absorbed at the active site of the catalyst. The rate-limiting step can be identified by Tafel analysis of the catalyst, and a Tafel slope of 116, 38, or 29 mV·dec⁻¹ assigns the rate-determining step in the HER process to Volmer, Heyrovsky, or Tafel reaction. The Tafel slope of 71.77 mV·dec⁻¹ lies between 38 and 116 mV·dec⁻¹, suggesting that a Volmer-Heyrovsky mechanism might be responsible for the HER process [13], and that the rates of the discharge step and the desorption step might be comparable during the HER process [25].

The chemical states of Mo and P in MoP flake were investigated by XPS experiments to find possible origin of the HER catalytic activity of MoP flake. The corresponding results are shown in **Figure 12**. The XPS spectrum of the Mo 3d window can be fitted to two doublets with peaking energy at 228.3 eV and 232.5 eV, respectively **Figure 12(a)**. The peak at 228.3 eV can be associated with Mo in MoP [26] [27], and this binding energy is very close to that of elemental Mo (227.9 eV) [28], implying that these Mo species have very a small positive charge. On the other hand, the peak at 232.5 eV suggests the presence of Mo⁶⁺ species in product [26] [27]. Two doublets (129.5 and 133.8 eV) can be found in the P 2p window **Figure 12(b**). The doublet at 129.5 eV can be attributed to P bonding to Mo [26] [27], and these P species have a slightly negative charge, because their binding energy (129.5 eV) is only 0.5 eV smaller than that of elemental P (130.0 eV) [29]. The doublet at 133.8 eV



Figure 9. Relationship of current density and experimental time in potentiostatic experiment (applied potential: -0.15 V *vs* RHE). Inset shows the polarization curves corresponding to the initial and 2000th scan in CV sweep. The sample is MoP/C on GCE (loading amount: 1.425 mg·cm⁻² of MoP and 0.7125 mg·cm⁻² of C), and potentials were not corrected with iR drop.



Figure 10. (a) Nyquist plot of EIS spectra at different overpotential recorded from MoP/C on GCE sample. (b) Plot of overpotential and inverse R_{ct} on a logarithmic scale. The sample is MoP/C on GCE (loading amount: 1.425 mg·cm⁻² of MoP and 0.7125 mg·cm⁻² of C), and potentials were not iR corrected.

can be related to P in PO_4^{3-} species [26] [27]. The Mo⁶⁺ and PO_4^{3-} possibly come from un-reacted precursor or oxidation product of MoP due to air exposure.

In Mo₂C and Mo₂N, Mo was found to have very small positive charge because of charge transfer from Mo to C or N [13] [14]. The d band electron structure becomes similar to that of Pt as a result of the charge transfer, and the HER catalytic activity of Mo₂C and Mo₂N have been associated with the modified d band structure of Mo [14]. The XPS spectrum of the Mo 3d window from MoP flakes reveals that Mo species in MoP flakes also have a very small positive charge. In addition, active sites in hydrogenase [19], its analogues (Ni(PS3^{*})(CO)]⁻ and [Ni(PNP)₂]²⁺) [19] and Ni₂P [19] contain hydride acceptors (Ni), which have a small positive charge. The charged nature of Mo in MoP resembles that of Mo in Mo₂C and Mo₂N, as well as the hydride acceptors in hydrogenase, its analogues, and Ni₂P, implying that the weakly charged Mo species in MoP might contribute to the catalytic activity of MoP. On the other hand, the proton acceptors are another kind of active sites in hydrogenase,



Figure 11. Equivalent circuit used for fitting of EIS data. R_s is the overall series resistance, CPE₁ and R_1 are the constant phase element and resistance describing electron transport at MoP/C and GCE interface or between MoP/C, respectively, CPE_{dl} is the constant phase element of the MoP/electrolyte interface, and R_{ct} is the charge transfer resistance at MoP/electrolyte interface.



Figure 12. XPS spectra of (a) Mo 3d window and (b) P 2p window collected from MoP flake.

its analogues, and Ni₂P. The proton acceptors are nonmetal sites having a small negative charge to trap protons (e.g., O of Glu23 in hydrogenase, -0.44 e; S in Ni(PS3^{*})(CO)]⁻, -0.4 e; N in [Ni(PNP)₂]²⁺, -0.34 e; P in Ni₂P, -0.07 e) [19]. The P species in MoP flakes are also slightly negatively charged. This charged nature is analogous to that of the proton acceptors in hydrogenase, its analogues, and Ni₂P. It is reasonable to postulate that the slightly charged P species in MoP work similarly to the proton acceptors in hydrogenase, its analogues, and Ni₂P. It is analogues, and Ni₂P, also contributing positively to the catalytic activity of MoP flakes. Therefore, the catalytic activity of MoP flakes is likely to be correlated with the charged natures of Mo and P.

Finally, the HER performance of MoP flakes in basic solution (KOH, 1M) was evaluated. The results are summarized in **Figure 13**. The corresponding η_{10} and η_{20} is 166 and 184 mV, respectively (**Figure 13(a)**). Current density recorded in a potentiostatic electrolysis experiment shows slight decrease in 10000 s, and the increase of η_{20} after 2000 scan in repeated CV scans is as small as 14 mV (inset of **Figure 13(b)**). In addition, the faradaic yield is also nearly 100% within the experimental error (**Figure 13(c)**). These experiments demonstrate that MoP flake can also work efficiently and stably in basic solution for hydrogen generation.

4. Conclusion

In summary, MoP flake was found to show efficient and stable catalytic activity in HER in acidic and basic solution. The optimal η_{20} of MoP flakes mixed with carbon black is as small as 155 mV in acidic solution and 184 mV in basic solution. The fabrication temperature of MoP flakes was found to influence the catalytic activity of MoP flakes. The samples fabricated at 800°C showed superior performance to those at 750°C or 850°C. The difference of catalytic activities might be associated with the crystallinity and specific surface area of products obtained at different temperatures. Charge transport resistances at the interface of catalyst/electrolyte suggest a Tafel slope of 71.77 mV·dec⁻¹, implying that a Volmer-Heyrovsky mechanism might be responsible for the HER process on the surface of MoP flakes. Potentiostatic electrolysis and accelerated degradation experiments show that MoP flakes can work stably in long-term hydrogen generation in both acidic and basic solution. In



Figure 13. Evaluation of MoP/C on GCE in basic solution (KOH, 1M). (a) Polarization curve (loading amount: $1.14 \text{ mg} \cdot \text{cm}^{-2}$ of MoP and 0.57 mg $\cdot \text{cm}^{-2}$ of C). (b) The relationship of current density and experimental time in potentiostatic experiment (applied potential: -0.18 vs. RHE). Inset shows the polarization curves corresponding to the initial and 1000th scan in CV sweep. (c) The plots of theoretical and measured volume of generated hydrogen in potentiostatic electrolysis experiment (applied potential: -0.28 vs. RHE). Only potentials in (a) were corrected with iR drop.

addition, the faradaic yield of MoP flakes in acidic and basic solution is nearly 100%. XPS experiments reveal that Mo and P in MoP flakes have a slight charge, and the catalytic activity of MoP flakes might be associated with the charged natures of Mo and P.

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