

Influence of Mn Doping on the Sensing **Properties of SnO₂ Nanobelt to Ethanol**

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Abstract

Mn doped SnO_2 nanobelts (Mn:SnO₂ NBs) and pure SnO_2 nanobelts (SnO₂ NBs) were synthesized by thermal evaporation technique at 1355°C with Ar carrier gas (25 sccm, 150 Torr). The SEM, EDS, XRD, TEM, HRTEM, SAED, XPS, UV-Vis techniques were used to characterize the attained samples. The band gap of Mn doped SnO₂ NBs by UV-Vis was measured to be 3.43 eV at room temperature, lower than that of the pure counterpart with ~3.66 eV. Mn:SnO₂ NB and pure SnO₂ NB sensors were developed. It is found that Mn:SnO₂ NB device exhibits a higher sensitivity with 62.12% to 100 ppm of ethanol at 210°C, which is the highest sensitivity among the three tested VOC gases (ethanol, ethanediol, and acetone). The theoretical detection limit for ethanol of the sensor is 1.1 ppm. The higher response is related to the selective catalysis of the doped Mn ions.

Keywords

SnO₂ Nanobelts, Mn³⁺ Doping, Gas Sensor, Single Nanobelt Device

1. Introduction

Metal-oxide semiconductor gas sensors have been used to detect gases for their efficiency and spread applicability [1] [2]. In particular, one-dimensional semiconductor nanostructures, such as nanobelt, nanowire and nanotube, due to their unique physical and chemical properties caused by their nanometer size effect, have aroused great interest due to their unique characteristics [3] [4] [5] [6] [7]. Therefore, metal-oxide semiconductor nanostructured materials, such as SnO₂, In₂O₃, WO₃ and ZnO, have been widely manufactured as sensors for de-

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tecting air composition and organic or poisonous gases due to low cost, easy accessibility, high performance, and reliable stability [8] [9] [10] [11]. Among them, SnO₂ with a band gap (3.6 eV), has high sensitivity and fast response [12]. Hence, SnO₂ nanomaterials have been investigated to detect many gases, such as ethanol, H₂S, and so on [13] [14] [15]. It is well known that the dopant of rare metals may improve the sensitive performance of nanomaterials [16]. For instance, Li *et al.* have reported that the sensitivity of Er-doped SnO₂ nanobelt device is 9 to the formaldehyde gas [17]. Ma *et al.* have found that the response of Sb-doped SnO₂ nanoribbon device reaches 56 (10) to 100 ppm (ppb) of H₂S at 150°C and 19 (1.6) to 100 ppm (ppb) of H₂S at 25°C [18].

Doping enhances the properties of semiconductors by providing a powerful strategy to control their optical, electronic, transport, and spintronic properties [19]. The optoelectronic properties such as photoluminescence and optical band gap of SnO_2 can also be improved by metals doping. Many studies were reported on the SnO_2 doped by Sb, In, and Mn etc. [20] [21] [22]. Much emphasis is put on manganese (Mn) due to its large equilibrium solubility and nearly the same ionic radii as with Sn^{4+} ion for substitution.

In this paper, we systemically investigated the sensing and optical properties of a single Mn:SnO₂ NB sensor to volatile organic (VOC) liquids and reported interesting results.

2. Experimental Section

2.1. Synthesis and Characterization of SnO₂ NBs and Mn:SnO₂ NBs

Single SnO₂ and Mn:SnO₂ NBs were obtained by thermal evaporation method. For the synthesis of Mn:SnO₂ NBs, the mixture of pure SnO₂ powders (>99.99 wt.%) and MnC₂O₄·2H₂O powders premixed in the weight ratio of 20:1 was put into a ceramic boat. The ceramic boat was placed into the central position of the horizontal alundum tube, which was put into a high temperature furnace. A silicon substrate coated with about 10 nm Au film was placed into the tube, the distance of silicon substrate and ceramic boat was about 15 cm. After cleaning the tube several times with nitrogen gas, the tube was evacuated by a mechanical pump to a pressure of 1 to 5 Pa. The precursors of SnO₂ and MnC₂O₄·2H₂O powders were maintained at 1355°C for 2 h and deposited on the Si substrate with Ar carrier gas (25 sccm, the pressure inside the tube is 150 Torr). After the furnace was cooled to room temperature naturally, white wool-like products were obtained. In order to compare the sensing properties of Mn:SnO₂ NBs and pure SnO₂ NBs, we also prepared pure SnO₂ NBs by similar method.

The nanobelts were characterized by scanning electron microscopy (SEM) and energy-dispersive X-ray spectroscopy (EDS), X-ray diffraction (XRD), transmission electron microscopy (TEM) and high-resolution electron microscopy (HRTEM), selected area electron diffraction (SAED), X-ray photoelectron spectrometer (XPS), and ultraviolet and visible spectrophotometer (UV-Vis).

2.2. The Manufacture and Characterization of a Single Nanobelt Gas Sensor

SnO₂ NBs and Mn:SnO₂ NBs were picked out and then dispersed into ethanol by the tweezers. A few of the resulting suspensions were dropped onto a silicon substrate with thickness of a 500 nm SiO₂ layer. The suspensions dried naturally and led to the nanobelts stick to the substrate closely. The mask plate was placed on the top of this substrate to prepare the electrodes. Patterned Ti (10 nm) and Au (100 nm) electrodes were successively deposited on the nanobelts in high vacuum by dual-ion beam sputtering (LDJ-2a-F100-100 series) with Ar carrier gas (10 mA/cm², 2.2×10^{-2} Pa).

The measurement of the gas sensor was processed with the equipment designed by our laboratory [23]. The process was conducted in a hermetic stainless steel box (20 L). Then, the device was put on a heating station, on which its temperature can be accurately controlled. The sensing properties of the device were measured by Keithley 4200 semiconductor test system. The applied bias voltage was 1 V and the testing interval was 200 seconds. The target liquid can be injected into a heater to evaporate the VOC liquid violently and the fan was used to produce a homogeneous atmosphere in the chamber. The optical microscopic image of a single nanobelt device is displayed in **Figure 1**.

3. Results and Discussion

3.1. Structural Characterization

3.1.1. SEM and HRTEM

The morphology of the as-synthesized materials is displayed in Figure 2. $Mn:SnO_2 NBs$ and $SnO_2 NBs$ have similar morphology. Figure 2(a), Figure 2(b) are SEM micrographs of $Mn:SnO_2 NBs$ by low and high magnifications. The product consists of a large quantity of ribbon-like structures with different thickness and width, as shown in Figure 2(a). Most of the nanobelts have uniform thickness and width. Figure 2(b) depicts that their thickness is less than



Figure 1. The microscope photograph of a prepared SnO₂ NB device.





Figure 2. (a) Low and (b) high magnification SEM micrograph of Mn:SnO₂ NBs. (c) Low and (d) high magnification SEM micrograph of SnO₂ NBs. (e) TEM image of Mn:SnO₂ NB. (f) HRTEM image of Mn:SnO₂ NB, its inset is SAED pattern.

100 nm, the width is from 250 nm to 1 μ m, and the length is about 50 μ m. It is also seen that the Mn:SnO₂ NBs have a good shape with smooth surface, which are suitable for preparing gas sensors. **Figure 2(c)** is SEM image of SnO₂ NBs by low magnification, it can be obviously seen that the SnO₂ NBs are band-like structure, the length of them is about 30 - 100 μ m. **Figure 2(d)** is a high magnification SEM micrograph of SnO₂ NBs, the SnO₂ NBs have smooth surface, the width is less than 1 μ m, and thickness is about 50 - 100 nm.

For further characterizing their microstructure, we carried out TEM, HRTEM and SAED examination and are presented in Figure 2(e) and Figure 2(f). TEM image illustrates that the width of a selected NB is about 300 nm. Figure 2(d) presents HRTEM image of a single Mn:SnO₂ NB and its corresponding SAED pattern. It is seen that the interplanar spacing is 0.3362 nm, matching with the (1 0 0) plane for tetragonal structure SnO₂, no defects have been detected. The regular arrangement of the diffraction spots is shown in the upper-right inset of Figure 2(d), revealing that the growth direction of Mn:SnO₂ NBs is along [1 0 0].

3.1.2. XRD and EDS

XRD pattern of Mn:SnO₂ NBs is presented in **Figure 3(a)**. The diffraction peaks can be indexed as the tetragonal structure SnO₂ with lattice parameters a = b = 0.4738 nm, c = 0.3189 nm (JCPDS file No. 71-0467).

In order to decide whether Mn ions were doped into SnO_2 NBs or not, energy-dispersive X-ray diffraction spectroscopy (EDS) pattern of a single Mn:SnO₂ NB is conducted, as shown in **Figure 3(b)**. It is seen that the doped Mn ions content of SnO_2 NBs is only 1.64 at.%. The Sn and O atomic percentage is 1:1.8, which is smaller than tin dioxide stoichiometric ratio (1:2), indicating that there are oxygen vacancies in the sample.

3.1.3. XPS Analysis

The XPS observation of as-prepared $Mn:SnO_2$ sample is shown in Figure 4(a). From the global XPS profiles, the lines related to Sn, O, Mn, and C elements are observed. Carbon is ubiquitously presented on all surfaces for XPS spectra. It is well known that the peak at 285.0 eV of carbon C 1 s is used as a reference for charge correction. The Sn(3d) band presents double peaks located at binding energies of 487.0 of Sn(3d_{5/2}) and 495.4 eV of Sn(3d_{3/2}), as shown in Figure 4(b). The separation distance between the two peaks is 8.4 eV, which confirms the formation of Sn⁴⁺ oxidation state in SnO₂ nanobelts [24]. The Mn (2p) state can be separated into two peaks. One is centered at 640.0 eV, the other at 644.9 eV, as displayed in Figure 4(c).

The deconvolution of the O (1s) peak shows two Gaussian peaks centered at 530.9 and 531.7eV, respectively (as illustrated in **Figure 4(d)**). The peak located at the lower binding energy corresponds to the O (1 s) core peak of O^{2-} bound to Sn⁴⁺ and the other one at higher binding energy originates from O defects [25].

$$\frac{n_{Sn}}{n_O} = \frac{I_{Sn}}{I_O} \times \frac{S_O}{S_{Sn}}$$

 S_{Sn} and S_O are sensitive factors ($S_{Sn} = 4.095$, $S_0 = 0.711$); I_{Sn} and I_O are the peak areas; n_{Sn} and n_O are atomic concentrations on behalf of Sn and O elements, respectively. The ratio of n_{Sn} and n_O is 0.56 by fitting. However, the ratio of n_{Sn} and n_O in pure SnO₂ is 0.5. It is further corroborated that the Mn:SnO₂ NBs have oxygen vacancies.



Figure 3. (a) XRD of Mn:SnO₂ NBs. (b) EDS of Mn:SnO₂ NB.





Figure 4. XPS spectra of the Mn:SnO₂. (a) global survey spectra and high-solution (b) Sn 3d, (c) O 1 s and (d) Mn 2p spectrum.

3.2. UV-Vis Spectra

The optical absorption coefficient a of a semiconductor close to the band edge can be expressed by the following Wood-Tauc Equation [26]:

$$\alpha = k \left(h \nu - E_g \right)^n / h v$$

where *a* is the absorption coefficient, *k* is a constant about material properties, hv is the energy of a photon, E_g is band gap, and *n* is a parameter that depends on the nature of the transition. In this case, *n* is equal to 1/2 for a direct bandgap material. The band gap can be estimated from a plot of $(ahv)^2$ versus photon energy.

Figure 5 shows the absorption spectra of Mn:SnO₂ NBs and SnO₂ NBs. The band gaps (Eg) of Mn:SnO₂ NBs and SnO₂ NBs are 3.43 and 3.66 eV, respectively. The shift may be due to the quantum size effect. The plots of $(ahv)^2$ as a function of the energy (hv) of the incident radiation for Mn:SnO₂ NBs and SnO₂ NBs are presented in the inset of **Figure 5**. All plots show non-linear nature. Compared with that of SnO₂ NBs, the redshift is observed in the absorption edge of Mn:SnO₂ NBs. The change on the bandgap suggests that the size of the nanobelt has an influence on the optical properties of the materials, which can be tuned by doping.

3.3. Sensing Properties

The sensor's response is defined as the relative change of resistance in the surrounding gas atmosphere divided by the resistance in synthetic air [27], which is defined as

$$S\% = \frac{Ra \cdot Rg}{Ra} \times 100\% = \left(1 - \frac{Rg}{Ra}\right) \times 100\%$$

In the formula of S(%), R_a is the sensor resistance in the air and R_g is the resistance in the tested gas.

The curves exhibit a linear shape, so both doped and pure NBs possess a good Ohmic contact, as shown in Figure 6. The slope of pure SnO₂ NB is less than that of Mn:SnO₂ NB. The resistances of Mn:SnO₂ NB and its pure counterpart are about $1.65 \times 10^8 \Omega$ and $1.61 \times 10^9 \Omega$ respectively at 210°C, indicating that the resistance of SnO₂ is greatly reduced after doping.

Upon exposed to 100 ppm of ethanol, ethanediol, and acetone gases, the responses of the sensors based on Mn:SnO₂ NB and its undoped counterpart have been tested as a function of the temperatures from 50°C to 300°C, as shown in Figure 7(a), Figure 7(b). The best operating temperature of Mn:SnO₂ NB to the three gases is 210°C and the responses to ethanol, ethanediol, and acetone are



Figure 5. The UV absorption spectra of Mn:SnO₂ NBs and SnO₂ NBs.



Figure 6. I-V curves of pure SnO₂ NB and Mn:SnO₂ NB devices.





Figure 7. (a) and (b) The gas sensitivity of $Mn:SnO_2$ NB and pure SnO_2 NB to 100 ppm of various gases at different temperature; (c) Histogram sensitive responses of $Mn:SnO_2$ NB and pure SnO_2 NB at 210°C; (d) The sensitivity of $Mn:SnO_2$ NB to ethanol from 5 to 1000 ppm at 210°C; (e) The curve of response versus ethanol concentration in the range of 5 - 100 ppm; (f) Responses to 100 - 500 ppm of ethanol at 210°C; (g) The fitting curve of response versus ethanol concentration in the range of response versus ethanol concentration in the range of 100 - 500 ppm.

62.12%, 47.92% and 33.33% respectively. However, the best working temperature of the SnO_2 NB to the three gases are 220°C and its responses are only 42.86%, 24.24% and 16.67%. Obviously, the sensitivity of Mn:SnO₂ NB is much better than that of the latter, especially to ethanol. The best working temperature of Mn:SnO₂ NB is lower and the sensitivity to ethanol reaches 62.12%, far higher to the others. As evidenced from **Figure 7(c)**, the sensitive responses of Mn: SnO₂ NB for ethanol, ethanediol and acetone are 1.51, 1.45, and 1.25 times as large as those of pure SnO_2 NB. The results manifest that doping of Mn reduces the optimum working temperature, enhances the sensitivity effectively, and greatly improves the selectivity of SnO_2 NB.

The response of the Mn:SnO₂ NB has also been tested and are illustrated in **Figure 7(d)**, **Figure 7(e)** under ethanol from 10 to 1000 ppm at 210°C. It is noted that the response increased drastically in the range of 10 - 100 ppm, then moderately in 100 - 600 ppm, and slowly in 600 - 1000 ppm. The response tends to become saturated when ethanol concentration goes up, which can be attributed to the surface coverage of the adsorbed molecules [24]. The response of Mn:SnO₂ sensors are provided upon repeatedly ethanol, ethanediol, and acetone gases exposure/removal cycles, as displayed in **Figure 7(f)**. Five cycles are successively recorded, corresponding to 100, 200, 300, 400, and 500 ppm of ethanol gase. For all testing cycles, the resistance returns completely to its original value once the gases are pumped out. It can be seen that Mn:SnO₂ NB has a high sensitivity to ethanol. The corresponding response/recovery time is about 18 s/20 s.

Figure 7(g) shows the fitting curve of response versus ethanol concentration in the range of 100 - 500 ppm. The response is approximately linearly, and the slope of it is 0.0334 ppm⁻¹ with a fitting quality of R = 0.9788. The sensor noise is calculated by the variation of the relative response at the baseline with help of root-mean-square deviation (RMSD) [22] [23] [28]. Then, 200 data points (*N*) of **Figure 7(f)** are collected, and the standard deviation (*S*) is obtained as 0.17369. According to $RMS_{noise} = \sqrt{S^2/N}$, RMS_{noise} is 0.01228 for the ethanol sensor. Based on the signal-to-noise ratio using $DL(ppm) = 3 \times RMS_{noise}/Slope$, the theoretical detection limit for ethanol of the sensor is 1.1 ppm.

3.4. Gas Sensing Mechanism

As a solid material, the gas-sensing mechanism of SnO_2 sensor belongs to surface phenomenon [29]. It is well known that the gas sensing involves three key reaction processes: adsorption, oxidation, and desorption [30]. The oxygens are adsorbed in several species such as O_2^- , O^- and O^{2-} through Equations (1)-(4) [31].

 O_2 (Gas) $\Leftrightarrow O_2$ (adsorption) (1)

 O_2 (adsorption) + $e^- \Leftrightarrow O_2^-$ (adsorption) (2)

- O_2^- (adsorption) + $e^- \Leftrightarrow 2O^-$ (adsorption) (3)
- O_2^- (adsorption) + $e^- \Leftrightarrow O_2^-$ (adsorption) (4)

when ethanol liquid is injected into the equipment and then evaporates, its vapor contacts the sensor and reacts with the surface chemisorbed oxygen. The process is as follows:

$$C_2H_5OH + 3O_2^-$$
 (ads) = $2CO_2 + 3H_2O + 3e^-$ (5)

$$C_2H_5OH + 6O^- (ads) = 2CO_2 + 3H_2O + 6e^-$$
 (6)

$$C_2H_5OH + 6O^{2-} (ads) = 2CO_2 + 3H_2O + 12e^{-}$$
 (7)



Obviously, many electrons are released and hence the resistance decreases during the reaction. In our work, the sensitivity to ethanol is availably improved by Mn ions. The reasonable sensitive improvement mechanism is proposed as follows. As shown as the results, Mn ions can promote the crystallinity of SnO₂ NBs. Besides, Mn doping can produce more oxygen vacancies and reduces the barrier height of the material [32].

$$Mn^{3+} + H_2O \rightarrow MnO^+ + H^+$$
(8)

$$O_0^{\times} \leftrightarrow V_0^{\bullet} + 2e' + 1/2O_2$$
 (9)

$$2MnO^{+} + O_{O}^{\times} \rightarrow Mn_{2}O_{3} + V_{O}^{*} + 2e'$$
(10)

As expressed by Equations (8)-(10), Mn ions may enhance the surface dehydrogenation resulting in the oxidation of ethanol needs lower energy, so that the liberation of electrons will be promoted [33] [34]. Hence, the electric conductivity of SnO_2 NB increases.

4. Conclusion

Mn:SnO₂ NBs and pure SnO₂ NBs were synthesized by thermal evaporation. The band gap of Mn doped SnO₂ nanobelts by UV-Vis was measured and is 3.43 eV respectively at room temperature, lower than that of the pure SnO₂ NBs (~3.66 eV). SnO₂ NB and Mn:SnO₂ NB sensors were developed. It is found that the Mn:SnO₂ NB device exhibits a higher sensitivity of 62.12% to 100 ppm of ethanol at 210°C, which is the highest sensitivity among the three tested VOC gases. The higher response is related to the selective catalysis of doped Mn ions.

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