

A New Rechargeable Battery Design Based on Magnesium and Persulfate*

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Abstract

A battery concept based on the chemical system of magnesium (anode) and persulfate (cathode) is presented. A complete procedure is given to prepare the battery for testing, although no experimental data is presented herein. The similarities of this system to a well-tested Li||LiFePO₄ system lend strong credibility to the concept, and the estimated performance characteristics presented. The advantages of this design include the following many areas. First, inexpensive, and available, battery reagents exist. Second, by analogy to the lithium ion battery for which comparisons are made, the full fabrication process for battery separator design is known and efficient; and both the kJ/kg and Amps/kg values are estimated to be substantially larger than the lithium ion battery (e.g., Li||LiFePO₄) experimental design. Finally, flammability of the Mg||MgS₂O₈ system can be expected to provide less of a potential flammability concern, compared to comparable lithium ion batteries. This is because lithium metal, as with any alkali metal, is aggressively flammable even under reduced moisture environments. The proposed magnesium persulfate battery calculated metrics yield an improvement of 194% greater output power (W/cm²·kg), and 154% greater stored energy (MJ/kg) than state-of-the-art lithium iron phosphate batteries.

Keywords

Magnesium Persulfate, Rechargeable Battery, Low Flammability, Energy Storage

1. Introduction

Rechargeable battery technology offers promise as a means to store energy for a wide variety of applications. Possible uses include: peak shaving and renewable energy storage stations; electric car and truck energy storage; electronic devices such as computers and cell phone energy storage; and small battery applications (e.g., 9 V, 1.5 V) such as for watches, toys, games, etc. In this list from the large to small storage applications, it is seen the

*Magnesium persulfate rechargeable battery.

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energy storage can range from more than 10 MJ to less than 1.0 kJ. To date, there does not exist a rechargeable battery concept that is suited to a broad range of energy storage capabilities such as these. The reason for existing rechargeable design limitations are due to the high cost of materials, such as lithium and other expensive metals [1] and sophisticated fabrication methods [2]-[5] in order to circumvent the low specific energy storage values (kJ/kg), and unacceptable performance of battery output (e.g., electrical performance) as in low values of Amps/kg.

In prior lithium ion battery (LIB) work by Nitta and Yushin [1], they examined a variety of pure metal composites, or their carbon composites (e.g., C, Si, Ge, Sn, Pb, P, As, Sb, Bi, Al, Ga, Zn, and Ag). They also examined a Li-Mg alloy here in their lithium ion battery study. In the latter study, their Li-Mg alloy exhibited a low lithiation potential of 0.05 V (e.g., a desirable result), but a somewhat high delithiation potential of 0.24 V (an undesirable result). Aside from this reference to magnesium as an Li-Mg anode LIB alloy material, there were no other information found to the use of magnesium alone (e.g., pure metal) in reference to a rechargeable battery employing magnesium-persulfate system.

A wide variety of unique and sophisticated LIB fabrication methods have been utilized. These include: work by Lee *et al.* [2] using cation-disordered oxides; a study by Mohanty *et al.* [3] on Li-Mn-rich oxides illustrating a unique phase change; a study by Kennedy *et al.* [4] employing nanowire LIB anodes demonstrating extended cycling more than 1000 times; and work by Li *et al.* [5] that examined the Li-Ni-Co-Mn-O atomic layer deposition.

2. Proposed Experimental Fabrication Method

Although no prototype battery was constructed, the fabrication method is identical to a method employed for a $Li||LiFePO_4$ rechargeable battery design recently published [6]. The only difference is the chemical system is changed to Mg||MgS₂O₈. Reference to that publication is made for experimental details. Here, only the experimental differences are highlighted.

The method of fabrication of the magnesium metal anode is easy compared to lithium, as the metal can be (acid) polished in ambient air, or with more care under an inert gas environment to eliminate the possibility of a surface oxide layer.

The MgS₂O₈ cathode material requires more preparation attention, as the starting reagent for this is solid persulfuric acid, $H_2S_2O_8(s)$. The persulfuric acid can be dissolved in a suitable organic solvent (e.g., one with a large electrochemical window), such as methylene chloride, into which is added magnesium hydroxide, Mg (OH)₂. Either the monohydrate, or anhydrous, magnesium hydroxide is preferred. This is because the desired precipitated product, MgS₂O₈, has a redox potential greater than the water redox potential breakdown voltage of 1.23 V. In essence, a two layer separatory funnel (with desired solid MgS₂O₈(s) plus CH₂Cl₂ on the bottom layer, and aqueous top layer) is separated, and the organic solvent is dried (under vacuum) from the pure MgS₂O₈(s). Other solvents may be preferable to methylene chloride. Further purification of MgS₂O₈(s) by recrystallization is an option, depending upon initial purity obtained.

The third battery component necessary to fabricate, and one that is critical to battery performance, is the battery separator. As referenced above, the work by Zhu *et al.* [6] describes the use of gel polymer doped glass fiber mats in a lithium ion battery experiment. With only minor modifications to the chemical system here this same synthesis can be applied here. The gel is comprised of poly(vinylidene fluoride) (PVDF) with a Mg(PF₆)₂ (1M) + ethylene carbonate-dimethyl carbonate-ethyl methyl carbonate (1/1/1 weight ratio solvent system). Fortunately, a supplier of magnesium diflurophosphate is American Elements (www.americanelements.com), hence this approach a straightforward adaptation of the method of Zhu *et al.* [6] that utilized LiPF₆, at 1M in the organic solvent mixture, instead. Also important is the major impurity in Mg(PF₆)₂ is sulfate, the reduction product of persulfate—an additional benefit to use of this electrolyte. Other suppliers, or synthesis of Mg(PF₆)₂, may be possible, but have not been explored. The full synthetic procedure of the PVDF-glass fiber mat (GFM) procedure is straight forward, but multi-stepped and fully detailed. Of significance here is the fact that the PVDF-GFM separator has a wide electrochemical window (ECW) of 4.8 V [6], which is large enough for the work of Zhu *et al.* [6], but also for the larger ECW here of the Mg||MgS₂O₈, described below.

A mention can be made regarding the specifics of the fabrication method of the proposed battery type presented here. Certainly additional work is needed in choosing particle size domains and thicknesses of the individual (anode, membrane, and cathode) construction regions of the battery components. However, these details, although important, are not speculated upon here, but are solvable by a disciplined approach to its optimization.

3. Rechargeable Battery Theory and Feasibility

3.1. REDOX Chemistry

The oxidation-reduction reactions of the $Mg || Mg S_2 O_8$ system are given below. Standard potential data was taken from Bard *et al.* [7].

$$Mg \rightarrow Mg^{2+} + 2e^{-}$$
 - oxidation - $E^{\circ}(Ox.) = +2.356 V$ (1)

$$S_2 O_8^{2^-} + 2e^- \rightarrow 2SO_4^{2^-}$$
 - reduction - E° (Red.) = +1.963 V (2)

Total:
$$Mg + S_2 O_8^{2-} \rightarrow Mg^{2+} + 2SO_4^{2-} = E^o (Tot.) = +4.319 V$$
 (3)

The potentials for reactions (1)-(3) are based on aqueous standard state data [7], and so actual potentials in the organic matrix, as proposed here, may vary somewhat but are not expected to be reduced by more than 5%. The redox potential of 4.3 V is seen to be below the ECW of the organic matrix system chosen, as described above, demonstrating its compatibility.

3.2. Voltage-Ionic Conductivity-Areal Power Performance Comparison

There are many considerations to take into account when making as estimate of the electrical performance of the magnesium persulfate battery. First, the migration of Li⁺ versus Mg²⁺ in the battery must be made. As studied by Malik *et al.* [8] for the LiFePO₄ system, the size of the microcrystalline LiFePO₄ domains and their defects affect Li⁺ diffusion. Most notably here, however, is the expected similarity between the LiFePO₄ and MgS₂O₈ domains that could be tailored to be comparable in size. Hence, these microcrystalline domains alone need not lead to differences in Li⁺ and Mg²⁺ diffusion. Second, the implicit differences between the Li⁺ and Mg²⁺ masses, collision diameters, and charges, can be taken into account according to Equation (4).

$$D \approx \frac{C}{\sqrt{Ms^2}} \tag{4}$$

where *D* is the diffusion constant (cm²/s), *c* is the ion charge, *M* is the ion mass (amu), and *s* is the collision cross-section (nm²). The full expression of diffusion is unnecessary here, as the additional terms are constant between Li⁺ and Mg²⁺, but can be found elsewhere [9]. Table 1 below lists the parameters for these two ions.

The collision areas were based on ion radii of R. Hancock, Acc. Chem. Res. 1990, 23, 253-257 [10]. Computing the relative diffusion constant ratio between the Mg^{2+} and Li^+ ions, namely $D(Mg^{2+})/D(Li^+)$, results in a value of 1.12. Hence, the Mg^{2+} diffusion constant is 12% larger than the Li^+ ion, largely because of it having twice the charge, despite having a larger square of its mass term. This 12% enhancement will be utilized in further calculations below, as it increases the ionic conductivity by this factor.

The rechargeable battery power output is given the formula:

$$P = \frac{KV^2d}{10A} \tag{5}$$

where *P* is the power (W), *K* is the ionic conductivity (mS/cm), *V* is the flat discharge voltage, *d* is the PVDF-GFM electrolyte gel polymer glass fiber mat thickness in microns (175 μ m), and *A* is the test area of the disk (e.g., cross-sectional area, cm²). The difference between the theoretically computed battery voltage and flat discharge voltage is a measure of the polarization losses. This information can be summarized in **Table 2**.

The result of the calculations in Table 2 are that the flat discharge power of the proposed Mg||MgS₂O₈ re-

Table 1. Comparision of parameters of magnesium and numum ions yielding differences in their diffusion constants.			
Parameter	Mg ⁺² Ion	Li ⁺ Ion	
C-ion charge	+2	+1	
M-ion mass (amu)	24.3	6.9	
S-collision area (nm ²)	0.0670	0.0707	

Table 1. Comparision of parameters of magnesium and lithium ions yielding differences in their diffusion constants

chargeable battery of 173 W/cm² is 42% larger compared to the battery in this study, that is representative of existing LiFePO₄ battery performance. This is an appreciable gain.

3.3. Power and Energy Performance Comparison

The conversion of power performance of the last section, into specific power performance, is readily achieved by noting the following. The power performance of W/cm^2 is actually W/cm^2 ·mol, where the mole refers to one reaction mole for either the Li||LiFePO₄, or Mg||MgS₂O₈ reactions. The mass of the lithium iron phosphate system (for both half-reactions) is 164.6 g/mol, whereas the magnesium persulfate system has a mass of 240.8 g/mol. Because the proposed magnesium system is a two electron system, and the lithium system is a one electron system, the result is the magnesium battery mass is only 73% that of the lithium system per mole of electrons tranferred.

Using this 73% reduction in battery mass, this translates into the specific power increase (W/cm²·kg) of the magnesium persulfate system divided by the lithium iron phosphate system to be 194% (nearly twice as efficient).

Of additional interest, is the specific energy storage capability. For example, using the flat discharge voltages of **Table 2**, the masses in kg/mol for each system, and Faraday's constant can yield values of stored energy in MJ/kg. This calculation has shown that the magnesium persulfate system has a value 154% larger than that of the lithium iron phosphate battery. A summary of the property comparison between the two battery systems is given in **Table 3** below.

In every category in **Table 3** it is seen that the proposed magnesium persulfate battery concept is superior to the currently used lithium iron phosphate battery. It must be noted that the reason why percentages are listed in

Table 2. Comparison of Li Li	PO4 and estimated Mg MgS2O8 battery voltages, ionic conductivity, and areal power percent
gain.	

Property	Li LiFePO4	$Mg \ Mg S_2 O_8$
Theoretical open circuit voltage (V)	3.82a	4.32a
Flat discharge voltage, V (V)	3.3b	3.71c
Ionic Conductivity, K (mS/cm)	1.13d	1.27e
Calculated battery flat discharge power increase (W/cm ²) as %	100%	142%f

a—Calculated based on the standard state ion potentials of Baird *et al.* [7]; b—Observed experimentally (see Ref. [6]); c—Computed value based on the data of Zhu *et al.* [6] illustrating the flat discharge voltage divided by the theoretical open circuit voltage is 86%; d—Measured in Ref. [6]; e—The ion transference number for Mg^{2+} was assumed to be 12% larger, based on the diffusion calculations above, compared to that for Li^+ . This increase is reflected in the larger ionic conductivity of the $Mg||MgS_2O_8$ system; f—Estimated based on data in Ref. [6] and Equation (5) here.

Table 3. A summary of the comparisons between the $Li ||LiFePO_4$ and projected $Mg ||MgS_2O_8$ battery operation parameters is presented.

Li LiFePO4	$Mg MgS_2O_8$
3.82	4.32
3.3	3.71
100%	112%
100%	142%
164.6	120.4
100%	194%b
100%	154%с
	3.82 3.3 100% 100% 164.6 100%

a—Given as 85% of theoretical open circuit voltage—this study; b—Calculated using weight per mole electron (g/mole \cdot e⁻) from this table, and W/cm² values from Table 2; c—Calculated using flat discharge voltage (V), ionic conductivity (mS/cm), thickness of membrane (see text), and weights per mole electron transferred (g/mole \cdot e⁻) from this table.

Table 3 is because it is more useful to note the percentage gain of a particular property. Hence, by not giving absolute values there is no tie to a particular study, making the comparison more universally valid and helpful.

4. Conclusion

The benefits of the $Mg||MgS_2O_8$ rechargeable battery here includes inexpensive reagents, known synthetic and fabrication methods, a lessened flammability risk, and substantial gains in specific power (194%) and specific stored energy (154%) performances. These issues make the proposed magnesium persulfate rechargeable battery superior to state-of-the-art lithium iron phosphate battery technology in particular and rechargeable batteries as a whole.

Acknowledgements

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References

- [1] Nitta, N. and Yushin, G. (2013) High-Capacity Anode Materials for Lithium-Ion Batteries: Choice of Elements and Structures for Active Particles, Particle & Particle Systems Characterization. Materials Views, Wiley-VCH, 1-20.
- [2] Lee, J., Urban, A., Li, X., Su, D., Hautier, G. and Ceder, G. (2014) Unlocking the Potential of Cation-Disordered Oxides for Rechargeable Lithium Batteries, *Science*, 343, 519-522. <u>http://dx.doi.org/10.1126/science.1246432</u>
- [3] Mohanty, D., Sefat, A.S., Payzant, E.A., Li, J., Wood III, D.L. and Daniel, C. (2015) Unconventional Irreversible Structural Changes in a High-Voltage Li-Mn-Rich Oxide for Lithium-Ion Battery Cathodes. *Journal of Power Sources*, 283, 423-428. http://dx.doi.org/10.1016/j.jpowsour.2015.02.087
- [4] Kennedy, T., Mullane, E., Geaney, H., Osiak, M., O'Dwyer, C. and Ryan, K.M. (2014) High-Performance Germanium Nanowire-Based Lithium-Ion Battery Anodes Extending over 1000 Cycles. *Nano Letters*, 14, 716-723. http://dx.doi.org/10.1021/nl403979s
- [5] Li, X., Liu, J., Banis, M.N., Lushington, A., Li, R., Cai, M. and Sun, X. (2014) Atomic Layer Deposition of Solid-State Electrolyte Coated Materials with Superior High Voltage Cycling for Lithium Ion Battery Application. *Energy & En*vironmental Science, 7, 768-778.
- [6] Zhu, Y., Wang, F., Liu, L., Xiao, S., Yang, Y. and Wu, Y. (2013) Cheap Glass Fiber Mats as a Matrix of Gel Polymer Electrolytes for Lithium Ion Batteries. *Scientific Reports*, 3, 1-6. <u>http://dx.doi.org/10.1155/2013/958506</u>
- [7] Bard, A.J., Parsons, R. and Jordan, J. (1985) Standard Potentials in Aqueous Solution. IUPAC-Marcel Dekker Inc., New York.
- [8] Malik, R., Burch, D., Bazant, M. and Ceder, G. (2010) Particle Size Dependence of the Ionic Diffusivity. *Nano Letters*, 10, 4123-4127. <u>http://dx.doi.org/10.1021/nl1023595</u>
- [9] Geankopolis, C.J. (1983) Transport Processes and Unit Operations. 2nd Edition, Allyn and Bacon Inc., Boston.
- [10] Hancock, R.D. (1990) Molecular Mechanics Calculations and Metal Ion Recognition. Accounts of Chemical Research, 23, 253-257. <u>http://dx.doi.org/10.1021/ar00176a003</u>