

Photoluminescence Compounds ZnGa₂Se₄ and ZnGa₂Se₄:Eu²⁺

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ABSTRACT

Photoluminescence (PL) spectra measurements have been carried out in the ternary chalcopyrite semiconductor compounds ZnGa₂Se₄ and ZnGa₂Se₄:Eu²⁺ using single-wavelength excitation of a Hg lamp with $\lambda = 365$ and 375 nm. Measurements were performed at the temperature range of (120 - 220 K) and (110 - 230 K) for ZnGa₂Se₄ and $ZnGa_2Se_4:Eu^{2+}$, respectively. No PL was observed for both crystals at the temperatures higher than 220 K ($ZnGa_2Se_4$) and 230 K (ZnGa₂Se₄:Eu²⁺). At temperatures lower than ~220 K and ~230 K one and three lines were observed for ZnGa₂Se₄ and ZnGa₂Se₄:Eu at 591 nm and 566, 591, 646 nm, respectively. 566 nm line was assigned as due of the $4f^{6}5d \rightarrow 4f^{7}(^{8}S_{7/2})$ transition of Eu²⁺ ions, whereas the rest two lines were attributed to the donor-acceptor recombination pairs. Probability of non-radiation transfers ($A = 10^8 - 10^9 \text{ s}^{-1}$), energy of optical phonons ($h\omega = 25 - 30 \text{ meV}$), Huan Rice parameter (S = 8 - 10), energy of thermal quenching ($\Delta E = 0.02 - 0.06$ eV) were determined from the temperature dependences of the full width at half maximum (FWHM = $\Gamma(T)$).

Keywords: Ternary Chalcopyrite; Photoluminescence; Rare-Earth Elements

1. Introduction

ZnGa₂Se₄ belongs to the group $A^2 B_2^3 C_4^6$ compounds (where A-divalent cations of Zn, Cd; B-trivalent cations of Ga, In, Al, but C-chalcogens of S, Se, Te). Presence of birefringence, optical activity, major values of non-linear susceptibility coefficient, wide band gap, bright luminescence, high sensitivity is characteristic of these compounds. Above-mentioned properties propose these compounds to be prospective materials for semiconductive optoelectronics. ZnGa₂Se₄ has been first synthesized by author [1], X-ray analysis has been carried out. It is shown that ZnGa₂Se₄ crystallizes in tetragonal structure (space group S_4^2), lattice parameters a = 5.496 Å, c =10.99 Å, c/a = 2.

Presence of ordered vacancy in cations sublattice is characteristic of the given structure. Optical and photoelectric properties of ZnGa₂Se₄ have been investigated in [2,3]. From reflection spectra at 300 K there has been first evaluated band gap of ZnGa₂Se₄ (2.58 eV). Later from optical absorption measurements close to the edge in polarized radiation at 300 K it is established that minimum optical transitions $\Gamma_3 + \Gamma_4 \rightarrow \Gamma_1$ localized in the centre of Brillouin zone, are equal to 2.43 eV and resolved in polarization $E \perp C$. Physical properties also

luminescent, optical, photoelectric and other properties of ZnGa₂Se₄ comparing with other compounds of group $A^{2}B_{2}^{3}C_{4}^{6}$, for instance CdGa₂S₄, CdGa₂Se₄ and others have not been studied yet [4]. It is probably related to the difficulties of ZnGa₂Se₄ synthesis. This paper first deals with the investigation results of photoluminescence (PL) in ZnGa₂Se₄ and ZnGa₂Se₄:Eu polycrystals.

2. Experimental

 $ZnGa_2Se_4$ was synthesized from high purity Zn (99.999%), Ga (99.999%), and Se (99.9999%). A stoichiometric elemental mixture was loaded into an ampule, which was then pumped down to 1.3×10^{-2} Pa and placed in a furnace preheated to $950^{\circ}C \pm 10^{\circ}C$ so that its tip was outside the furnace. When the body of the ampule reached the furnace temperature, vigorous reaction began. In the course of the reaction, the furnace temperature was gradually raised to 1150°C and then maintained constant for 3 h to homogenize the melt. Next, the ampule was cooled to $500^{\circ}C \pm 5^{\circ}C$ and held there for 24 h, followed by furnace cooling. The dopant was introduced into ZnGa₂Se₄ in the form of europium fluoride, EuF₃. The samples were excited by the radiation from a PRK 4 mercury lamp at temperatures from 77 to 230 K. The sample temperature was monitored using a copper-constantan thermocouple and electronic potentiometer. PL spectra were measured on an SDL 1 spectrometer.

3. Results and Discussion

PL spectra of ZnGa₂Se₄ and ZnGa₂Se₄:Eu²⁺ in the temperature range of \sim (120 - 220 K) and \sim (110 - 230 K) are presented in **Figures 1** and **2**, respectively. No difference was detected when exciting the spectra with two different wavelengths at 365 and 375 nm.

Characteristic features of the temperature dependences of the PL spectra are as follows:

1) ZnGa₂Se₄:

a) No PL was observed at the temperatures higher than ~ 220 K.

b) One PL line appeared at ~220 K (591 nm, 2.10 eV) (Figure 1).

c) The intensity of PL line is linearly increased with decreasing the temperature as it is shown in **Figure 3** (curve 1). The energy position of the PL line did not revealed noticeable temperature dependence up to ~ 120 K.



Figure 1. PL spectra of ZnGa₂Se₄ at various temperatures *T* (K): 1-120, 2-130, 3-148, 4-220.



Figure 2. PL spectra of $ZnGa_2Se_4:Eu^{2+}$ at various temperatures T (K): 1-110, 2-157, 3-181, 4-198, 5-208, 6-215, 7-230.

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d) The temperature dependence of the $\Gamma(T)$ of PL line is shown **Figure 4** (curve 1).

2) ZnGa₂Se₄:Eu²⁺:

a) No PL was observed at the temperatures higher than 230 K.

b) Three lines were detected at ~ 110 K with the maxima at 566 nm (2.19 eV), 591 nm (2.10 eV) and 646 nm (1.92 eV) (Figure 2).

c) The lines at 591 and 646 nm disappeared when the temperature of sample increased higher than \sim 120 K.

d) The temperature dependence of the intensity of 566 nm line revealed two linear regions with different slope (**Figure 3**, curve 2). It confirms existence of two energy levels in the forbidden gap of doped compound.

e) The energy position of the PL line at 591 nm did not revealed noticeable temperature dependence up to \sim 230 K.

f) The temperature dependence of the FWHM of PL line (566 nm at 110 K) is shown **Figure 4** (curve 2).



Figure 3. Temperature dependences of the PL intensity for $ZnGa_2Se_4$ (1) and $ZnGa_2Se_4$:Eu²⁺ (2) crystals.



Figure 4. Temperature dependences of the FWHM of the most intense PL line (591 nm) for $ZnGa_2Se_4$ (1) and $ZnGa_2Se_4$:Eu²⁺ (566 nm at 110 K) (2).

It is seen that both samples revealed general characteristic features of the temperature dependences of the PL intensity. For both samples the intensity of PL lines decreased with increasing temperature. For ZnGa₂Se₄ effective PL was observed in the spectral range of 530 -670 nm and for ZnGa₂Se₄:Eu²⁺ in the range of 530 - 630 nm. $\Gamma(T)$ of PL observed also increased with increasing temperature.

Analysis of obtained data showed that the PL maxima at 566 nm for $ZnGa_2Se_4:Eu^{2+}$ is caused by the $4f^65d \rightarrow 4f^7(^8S_{7/2})$ transitions of Eu^{2+} ions, whereas maxima at 591 and 646 nm are due of the donor-acceptor pairs.

Temperature dependences of the intensity of PL lines for pure (591 nm) and doped compound (566 nm) were analyzed on the base of Mott theory and formula given in [4] and modified by the authors of [5]:

$$I = I_0 \frac{1}{1 + \tau A e^{\frac{\Delta E}{kT}}} \tag{1}$$

where I_0 —intensity at the temperature T at which no quenching takes place, τ —average life time of Eu²⁺ ions in an excited states, A—probability of non-radioactive transfer, k—Boltsmann constant, ΔE is for the energy of quenching. Value of A is poorly temperature dependent, whereas τ should not depend on temperature. In case of forbidden transitions increasing temperature leads to decrease τ [5]. In this case formula (1) can be written as follows [5]:

$$\lg\left(\frac{I_0}{I} - 1\right) = \lg\left(\tau A\right) - \frac{0.43\Delta E}{kT}$$
(2)

where notations are the same as in (1).

It follows from (2) that there is inverse linear dependence of $\lg(I_0/I - 1)$ on *T*. For both compounds linear dependences of $\lg(I_0/I - 1)$ on *T* is shown in **Figure 3**. Extrapolation of these dependences to $1/T \rightarrow 0$ gave the product of τA equal ~(500 - 1000) for both compounds. By using value $\tau = 10^{-6}$ s given for Eu²⁺ in [6,7] a probability of non-radioactive transfer $A \approx 10^8 \div 10^9$ s⁻¹ was estimated.

Analysis of the temperature dependences of $\Gamma(T)$ give an information about the mechanism of radiation transfer and a number of parameters characterizing it. Theoretical analysis of the experimental data of the absorption and emission spectra of solids allowed the author to derive the expressions for the temperature dependences of the Stokes shift (ΔS), Huan Rice factor (*S*), phonon energy ($\hbar \omega$) and the $\Gamma(T)$ [8-10]:

$$\Delta S = (2S - 1)\hbar\omega \tag{4}$$

$$\Gamma(T) = 2.36\sqrt{s}\sqrt{2kT\hbar\omega} \tag{5}$$

Comparison of the experimental data obtained in the present work (Figure 3) with those estimated with the

formulaes (3) and (4) results to the next values of the Stokes shift $\Delta S \approx 0.5$ eV, energy of the optical phonons $\hbar \omega \approx (25 - 30 \text{ meV})$, $\Gamma(T) \approx (0.14 - 0.18 \text{ eV})$ and Huan Rice factor $S \approx (8 - 10)$. It is worth to mention that the energy of phonons estimated in the present work are in good agreement with those obtained for ZnGa₂Se₄ from Raman scattering measurements [10]. Analysis of existing experimental results on electric and photoelectric properties of pure and doped ZnGa₂Se₄ and results obtained in the present research allowed to suggest possible energy diagram of the optical transitions which is shown in **Figure 5**.

It should be noted that Huan Rice parameter reflects the most probable number of phonons which take part in the absorption and emission processes [11-13]. According to [11] in case of strong localization of principle wave function of the excited state and its blurring upper theoretical limit of Huan Rice factor $S \approx 55$ eV. In dependence on the degree of the wave function localization the S parameter varies in the range of $S \approx 2 - 55$ eV for different materials [11,12]. At present stage a reason for lower values of S obtained in the present work is not clear. One probably reason may be that in the ZnGa₂Se₄ and ZnGa₂Se₄:Eu²⁺ crystals degree of the wave function localization and its blurring is the lowest one when comparing with other materials. Further research is necessary to clarify this suggestion.

Analysis of our data show that the product of the PL maxima intensity I_{max} $\cdot \Gamma(T)$ does not depend on temperature $B = I_{\text{max}} \cdot \Gamma(T) = \text{const}$ (Figure 6).

Increase of the $I_{\text{max}} \cdot \Gamma(T)$ for both compounds may be qualitatively explained by the model suggested in [13] and presuming the formation of PL mechanism with the process in which phonons are playing an important role.



Figure 5. Energy diagram of the optical transitions for Eu²⁺ doped ZnGa₂Se₄ at 110 K.



Figure 6. Temperature dependence of the product of the PL maxima intensity $B = I_{max} \cdot \Gamma(T)$ for Eu²⁺ doped ZnGa₂Se₄.

4. Conclusion

There has been synthesized ZnGa₂Se₄ compound, lattice parameters (a = 5.496 Å, c = 10.99 Å, c/a = 2) are determined, during the synthesis this compound is activated by Eu²⁺. In PL spectrum of ZnGa₂Se₄ maximum at 591 nm has been found out, in PL spectrum of ZnGa₂Se₄:Eu maxima at 566, 591 and 646 nm have been found out at 77 - 230 K in ZnGa₂Se₄ and ZnGa₂Se₄:Eu compounds. From analysis of temperature dependence $\Gamma(T)$ probability of non-radiative transfers ($A = 10^8 - 10^9 \text{ s}^{-1}$), energy of optical phonons ($\hbar \omega = 25 - 30 \text{ meV}$), Huan Rice parameter (S = 8 - 10), energy of thermal quenching ($\Delta E =$ 0.02 - 0.06 eV) have been determined. PL spectrum maximum of ZnGa₂Se₄:Eu at 566 nm is due to the transition 4f⁶5d \rightarrow 4f⁷(⁸S_{7/2}) of Eu²⁺ ions, maxima at 591 and 646 nm are due donor-acceptor pair.

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