

A Single-Step Process for Preparing Supercapacitor Electrodes from Carbon Nanotubes

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ABSTRACT

This paper introduces an easy single-step process for preparing the supercapacitor electrode from carbon nanotubes (CNTs) which were directly grown on a nickel foam framework by a chemical vapor deposition (CVD) technique. Due to the hierarchical porous structure and robust CNT-metal contacts, the present electrode exhibits better capacitive performance and lower internal resistance than the regular transfer method. Its specific capacitance of 25 F/g is 127% higher than the value of the electrode prepared by the transfer method. This work proposes a simple, effective and economical approach for preparing a supercapacitor electrode without additional catalysts and pre- or post-treatments.

Keywords: Carbon Nanotubes, Supercapacitor, Energy Storage and Conversion

1. Introduction

Supercapacitors, also known as electric double layer capacitors or electrochemical capacitors, store energy using reversible ion adsorption on the surface of highly porous materials. They are very promising to be applied as longlife and high-power energy storage devices [1]. Carbon nanotubes (CNTs) have been recognized as a kind of attractive electrode materials for supercapacitors because of its high accessible surface area, low electric resistance and interconnected network structure [2,3]. Instead of attaching CNTs to the current collectors mechanically (the transfer method) [2,4,5], the method of directly growing CNTs on current collectors has received considerable attentions due to the robust CNT-metal contacts and one-step process [6]. Recently, Many efforts have been made for directly growing CNTs on bulk substrates, such as Al [7], Ni [8], Ta [9], stainless steel [10] and Inconel [6,11], achieving desired electrochemical capacitive properties. However, these direct growth processes generally use a chemical vapor deposition (CVD) technique through catalyst based methods [6,7]. Therefore, the complex substrate pretreatments are required including coating an Al₂O₃ film using electron beam evaporation [9,11] and multistage processes using plasma enhanced techniques [8,10]. Furthermore, by using simple two-dimensional bulk substrates, only a lim-

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ited exterior surface of the metal substrate is utilized. Recently, J. R. McDonough *et al.* [12] showed that supercapacitors based on carbon nanofibers grown on porous nickel foam substrates had large per-area capacitances. However, in this approach, the substrate pretreatment with atomic layer deposition was required to help the nucleation and growth of carbon nanofibers.

In this letter, we present a direct growth approach of CNTs on a three-dimensional porous nickel (Ni) foam for fabricating supercapacitor electrodes by using an easy single-step CVD technique without any additional catalysts and pre-or post-treatments. Owing to the hierarchical porous structure and robust CNT-metal contacts, we demonstrate that the present electrode exhibits higher electrochemical capacitance, lower internal resistance and superior capacitive performance at very high current densities than the regular transfer method.

2. Materials and Methods

The CNTs were synthesized directly on a Ni foam in a quartz tube furnace CVD system using an easy singlestep approach. The Ni foam on a quartz boat was placed in a quartz tube and 200 sccm Ar was continuously introduced into the quartz tube to exhaust air. The CNTs were grown on the porous Ni foam substrate by blowing 30 sccm C_2H_2 in the atmosphere of Ar at 610°C inside the quartz tube for 10 min. The grown CNTs on Ni foam framework (Ni foam-CNTs) were directly used as an electrode for making supercapacitors. For comparison, the transfer method was also used to fabricate the CNT electrodes (transfer CNTs). That is, the CNTs were removed from the Ni foam, and were prepared as pellets by pressing a mixture of 91 wt% of as-grown CNTs and 9 wt% of polytetrafluoroethene (PTFE) binder. The total mass of active electrode materials was used to calculate specific capacitance.

The microstructure was observed by using a transmission electron microscope (TEM) (JEOL JEM 2010, Japan). The electrodes were characterized by a scanning electron microscope (SEM) (SIRION, FEI, the Netherlands). Electrochemical characterizations were carried out by an electrochemical workstation (CHI 660C, China) in 6 M KOH aqueous solution at room temperature in a three electrode configuration using a saturated calomel electrode (SCE) as the reference electrode. Different electrochemical techniques were used to characterize the capacitive performance, and internal resistance of supercapacitor electrodes. Cyclic voltammetry (CV) were performed between -0.9 and -0.1 V at different scan rates. A galvanostatic charge-discharge experiment was carried out between -0.9 and -0.1 V at the current density range from 0.25 to 5 A/g. The electrochemical impedance spectroscopy at a dc bias of -0.9 V was performed with a 10 mV AC signal amplitude in the frequency range from 100 kHz to 10 mHz.

3. Results and Discussion

Figure 1 and Figure 2 show the SEM and TEM images of CNTs directly grown on porous Ni foam, respectively. Obviously, the entangled CNTs were well grown on the surface of Ni foam framework. Comparing with the CNTs grown on bulk substrates, the present CNTs on the porous substrate make full use of the large internal surface area of the Ni foam. A hierarchical porous structure is therefore formed. That is, the Ni foam framework provides macropores which act as transport channels into the interior pores of the electrode. The interconnected CNTs network produces mesopores and micropores at the interstices, surface and central canals of the intertwined CNTs which contributes dominantly to the capacitance of supercapacitor electrodes [13]. The CNT-Ni foam interface of electrode fabricated by using the transfer method is present in Figure 3. The CNTs were aggregated and indirectly mechanically attached to the Ni foam framework mechaniccally via binder. There are obvious spatial gaps between CNTs and Ni foam framework which block the electronic transportation at this interface and therefore increase the internal resistance of supercapacitor electrodes.



Figure 1. SEM morphologies of CNTs grown on the Ni foam. (a) Low magnification; (b) High magnification.



Figure 2. TEM image of CNTs grown on the Ni foam.

Figure 4(a) illustrates the CV curves of the Ni foam-CNTs and the transfer CNTs. Both electrodes present regular, rectangular shapes at a relatively high scan rate (200 mV/s), indicating well-developed capacitive properties [2]. However, the electrode with Ni foam-CNTs ex-

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Figure 3. SEM morphologies of CNT-Ni foam interface of electrode fabricated by using the transfer method. (a) low magnification; (b) high magnification.

hibits much higher specific capacitance, because its CV curve covers a much larger area. **Figure 4(b)** is the ratio of retained capacitance calculated from galvanostatic charge-discharge experiments of both electrodes as a function of charge-discharge current density. As the current density rises to a very high value of 5 A/g, the specific capacitance of Ni foam-CNTs retains a value of 72% at the current density of 0.25 A/g, which exhibits an impressive capacitive performance at high current densities. However, a CNT electrode prepared by the transfer method loses 67% of capacitance at the same current density.

The Nyquist plots (**Figure 4(c**)) show an inconspicuous semi-circle at high frequencies and a steeper slope at low frequencies for the electrode with Ni foam-CNTs, which indicates that the Ni foam-CNTs electrode exhibits much lower internal resistance than that of the CNT electrode prepared by the transfer method [14,15]. In order to confirm these results, the comparison of the IR drop for both electrodes at different charge-discharge currents was preceded, as shown in **Figure 4(d**). Each branch of the galvanostatic charge-discharge curves starts with a volt-



Figure 4. (a) CV curves taken at 200 mV/s; (b) The ratio of retained capacitance values as a function of charge-discharge current density; (c) Nyquist plots (0.01~100000 Hz). The upper inset is a local amplified image at high frequencies; (d) IR drop plots of the Ni foam-CNTs and transfer CNTs electrodes.

age drop (IR drop) due to the internal resistance of electrode [16], and the slope of the linear relationship between the IR drop and charge-discharge current can be used to estimate the over-all resistance [15]. From **Figure** 4(d), the slope k is 1.64 and 4.19 for the Ni foam-CNTs and the transfer CNTs respectively, which reveals a better conductivity of electrode built from the CNTs grown on

electrode material	cyclic voltammetry (F/g)	galvanostatic charge-discharge (F/g)	electrochemical impedance spectroscopy (F/g)	average specific capacitance (F/g)
Ni foam-CNTs	25	24	25	25
Transfer CNTs	9	13	10	11

Table 1. Specific capacitances of the supercapacitor electrodes prepared using different techniques.

Ni foam.

The above experimental results indicate that the electrode with the Ni foam-CNTs has a significant higher specific capacitance than that of the CNT electrode prepared by using the transfer method. This can be mainly attributed to the hierarchical porous structure of Ni foam-CNTs electrode, which facilitates the charging and discharging of electric double-layer by profiting the penetration of ion at the electrode/electrolyte interface through a proper pore-size distribution [13]. The average specific capacitance of the Ni foam-CNTs electrode measured using three electrochemical techniques is 25 F/g as given in **Table 1**, which is 127% higher than the value of the CNT electrode prepared by the transfer method at the same conditions. At high charge-discharge current densities, easily permeable hierarchical porous structure also contributes to the impressive capacitive performance of the Ni foam-CNTs electrode by making better use of electrochemically accessible pores. The better conductivity of the Ni foam-CNTs electrode owes to the robust CNT-metal contacts-another important factor for determining the excellent capacitive performance at high current densities. For the CNTs electrode prepared by the transfer method, the high internal resistance can increase the voltage drop and resistive energy loss, which will restrain the capacitive performance of electrode and this effect, is enhanced remarkably at high charge-discharge current densities.

4. Conclusions

In conclusion, the present process provides an effective approach for directly preparing a supercapacitor electrode from CNTs without any additional catalysts and pre- or post-treatments. It presents much higher electrochemical capacitance, lower internal resistance and better capacitive performance at very high current densities than the CNT electrode prepared by the conventional transfer method. It is expected to have wide applications in various fields such as chemical and biological sensors and catalyst supports.

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